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IN-PILE MEASUREMENT OF THE ELECTRICAL
RESISTIVITY AND THERMOELECTRIC
POWER OF SINTERED UO_2

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OAK RIDGE NATIONAL LABORATORY
operated by
UNION CARBIDE CORPORATION
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U.S. ATOMIC ENERGY COMMISSION

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IN-PILE MEASUREMENT OF THE ELECTRICAL RESISTIVITY
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ABSTRACT

Two thin plates of high-density UO₂ were sandwiched together with thermocouples between them and pressed against the outer surfaces. The thermocouples were used as electrodes and the electrical resistivity of the UO₂ measured by the potential drop method using corresponding arms of the thermocouples as current lead-in wires and potential probes. The UO₂ was heated by its own fission heat and the electrical resistivity measured at temperatures from 50 to 972°C. Preirradiation measurements were made using an electric furnace to obtain temperatures to 750°C.

Electrical resistivity did not change with irradiation up to about 10^{18} nvt, at which point the resistivity decreased by a factor of 3. The resistivity ranged from 8.75×10^3 ohm-cm at room temperature (before irradiation) to 0.4 ohm-cm at 773°C (after nvt $> 10^{18}$). The activation energy was 0.5 ev during irradiation as compared with 0.65 ev before irradiation.

The temperature between the plates was higher than the outside temperature during irradiation. This thermal gradient created a thermoelectric emf between the UO₂ and the platinum thermocouples. The thermoelectric emf for the Pt-UO₂-Pt system ranged from 130 μ v/°C at 162°C to 29 μ v/°C at 900°C. The UO₂ behaved like an n-type conductor during irradiation.

INTRODUCTION

The electrical conductivity of unirradiated UO₂ has been measured by several investigators, and a comprehensive survey of the literature is given by Willardson *et al.*¹ The need for good measurements of electrical conductivity, particularly at high temperatures, has been emphasized,²

¹R. K. Willardson, J. W. Moody, and H. L. Goering, The Electrical Properties of Uranium Oxides, BMI-1135 (September 1956).

²S. Aronson and J. Belle, J. Chem. Phys. 29, 151 (1958).

not only because of their intrinsic value in understanding the behavior of UO_2 as a semiconductor, but also because of their bearing on achieving an understanding of the oxidation of UO_2 and $\text{UO}_2\text{-ThO}_2$ solid solutions. It is of equal interest to study the effects of neutron irradiation on the electrical conductivity. It would be of considerable advantage to make measurements during irradiation, since ionization produced by fission fragments might affect the free carrier density. There have been no reports so far in the open literature on in-pile or postirradiation measurements of the electrical conductivity of UO_2 .

Willardson et al.¹ studied the thermoelectric power and electrical conductivity of UO_2 cylinders as a function of the O/U ratio in the temperature range from room temperature to about 750°C . From these measurements they deduced the phase equilibrium diagram in the system $\text{UO}_2\text{-O}_2$. The thermoelectric power was measured for the couple Alumel- UO_2 -Alumel. It was observed that the specimens in the as-prepared states were p-type semiconductors and became n-type as the O/U ratio was increased. Also, minima and maxima in the thermoelectric power were observed for different UO_{2+x} sintered pellets in the temperature range $180\text{--}340^\circ\text{C}$, depending upon the carrier type and O/U ratio.

Aronson and Belle² and Aronson and Clayton³ studied respectively the electrochemical systems $\text{Fe, FeO} \mid (\text{CaO} + \text{ZrO}_2) \mid \text{UO}_{2+x}$, Pt and $\text{Fe, FeO} \mid (\text{CaO} + \text{ZrO}_2) \mid (\text{U}_{1-y}\text{Th}_y\text{O}_{2+x})$, Pt and therefrom measured the emf of the half cells (UO_{2+x}) , Pt and $(\text{U}_{1-y}\text{Th}_y\text{O}_{2+x})$, Pt in the temperature range $1150\text{--}1350^\circ\text{K}$. From these data they deduced the thermodynamical properties of the system $\text{UO}_2\text{-O}_2$ and $(\text{UThO}_2)\text{-O}_2$.

In the present study, in-pile measurements of the electrical resistivity of UO_2 and of the thermoelectric power of the system Pt- UO_2 -Pt have been made in the range $150\text{--}1150^\circ\text{C}$ up to an integrated flux of about 0.2×10^{20} nvt. The results reported here are of a preliminary nature, as further work is in progress.

EXPERIMENTAL

The in-pile measurements were made in the C-1 facility of the Oak Ridge Research Reactor. This facility was designed primarily for studies

³S. Aronson and J. C. Clayton, J. Chem. Phys. 32, 749 (1960).

of fission gas release from bare fuel plates of UO_2 and has been described in detail by Carroll and Baumann.⁴ The facility permits an accurately controlled insertion of the fuel specimen into the reactor core so that studies can be made at different flux levels. The fuel specimen is heated by its own fission heat, and any desired temperature can be obtained. At a given flux position, a wide range of fuel temperatures (about 600°C) can be achieved by the independent variations of a cooling air flow and the nature of the sweep gas.

The UO_2 fuel specimens used in this facility are usually in the form of flat, very thin platelets (approaching the geometry of an infinite slab) with thermocouples positioned on either side. This fuel configuration has been used in attempts to determine the thermal conductivity of UO_2 by measuring the temperature gradients through the plates.⁴ It seemed desirable to take further advantage of the existing facility and fuel configuration to measure the electrical resistivity.

Because the prime purpose of the experiment is the study of continuous fission gas release, all experimental work on additional properties had to be such as would not interfere with the main aim. The need to make good contact between the UO_2 and the electrode was recognized and different methods were tried. The coating of a large area of the fuel plate, or the use of large contacts, could not be permitted since they might interfere with fission gas release. Attempts to utilize small vacuum-deposited coatings of platinum were not successful. The method of painting metallic paints and firing at high temperatures was not used because of the likely effects on the O/U ratio of the sintered plates. Tight-fitting metallic casings could not be used because the differential temperature expansion between the UO_2 and the casing would probably break the thin plates.

After considering all the various possibilities, it was decided to use spring-loaded thermocouples which would yield to the thermal expansion of the UO_2 and yet make good contact at all times. Of the methods

⁴R. M. Carroll and C. D. Baumann, Experiment on Continuous Release of Fission Gas During Irradiation, ORNL-3050 (February 1961).

available for measurement of electrical resistivity, the potential-drop method was chosen, as it would eliminate the need to know the precise variation of the resistivity of the lead wires. The corresponding arms of the thermocouples were used as the current lead-in wires and potential probes. The thermocouples were also used to measure the temperature of the UO_2 . This method, from the results obtained so far, seems to have worked reasonably well, but it cannot be regarded as completely satisfactory. The contacts are susceptible to possible mechanical vibrations. The possibility of increasing the number of springs as well as of having a spring-loaded knife-edge contact is being considered.

Figure 1 shows the arrangement of the specimens and thermocouples as fully assembled. Two sintered UO_2 plates $1 \times 0.5 \times 0.040$ in. were

Fig. 1. Thin-Plate Samples Positioned in Facility Tube.

placed between two sintered ThO_2 plates on the side and supported on another ThO_2 plate at the bottom. These were secured in an Inconel holder, which could slide into a larger Inconel capsule. Two thermocouples, which also served as electrical lead wires, were located on one plate as nearly opposite each other as was possible. The other center thermocouple, on the second UO_2 plate, was slightly displaced (Fig. 2a and b). All the thermocouples were, however, located at the same height. This was achieved

Fig. 2. Location of Thermocouples Between Plates.

by cutting open the ceramic¹ (Al_2O_3) thermocouple sheath at the side at equal distances from one end, pulling the thermocouple wire through the opening, and spot-welding the couples (Fig. 2c). The thermocouples were pressed tightly against the UO_2 surface by means of tungsten springs which acted on the ThO_2 plates and through these on the thermocouples. The ThO_2 plates were used to study certain aspects of fission gas release and were not pertinent to the measurement of electrical resistivity or thermoelectric power.

The circuit used for making the measurements is shown in Fig. 3. For the sake of simplicity, only one UO_2 plate is shown, and some of the switches used to convert from electrical to temperature measurements are not shown. The potential drop across the UO_2 plate and the thermoelectric emf were measured on a Leeds and Northrup K_2 potentiometer and in favorable ranges also directly recorded on a 10-mv-span Honeywell-Brown recorder. The current was read directly on a 0-100 μa Weston ammeter as well as recorded. Two DPDT switches enabled the reversal of the polarity of the superimposed emf and the measurement of the emf across the plate in both positions.

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Fig. 3. Circuit for Measurement of Electrical Resistivity and Thermoelectric Power.

RESULTS AND CONCLUSIONS

The fission heat and to some extent the gamma heat produced the elevated temperatures in the in-pile runs. For any given position in the neutron flux field and for any given flow rate of the cooling air stream, the temperatures remained constant within $\pm 5^\circ\text{C}$ over prolonged periods. However, because of the geometry of the two fuel plates, the temperatures between the plates were always higher than on the outside. This temperature gradient generated the thermoelectric emf.

The electrical resistivity of the sample was measured prior to installation in the reactor, after assembling and leak testing the entire experiment. A small Nichrome-wound furnace was placed around the capsule to obtain elevated temperatures. During these measurements, a helium purge was maintained to exclude oxygen from the system and prevent the oxidation of the UO_2 . The results of this out-of-pile measurement are tabulated in Table 1 and summarized in Fig. 4.

Table 1. Out-of-Pile Measurement of Electrical Resistivity

Temperature (°K)	Reciprocal Temperature [(°K) ⁻¹]	ρ , Resistivity (ohm-cm)
	$\times 10^{-3}$	
294	3.40	8750
463	2.16	28.7
473	2.11	26.7
663	1.51	19.34
678	1.47	16.77
708	1.41	13.31
723	1.38	13.92
813	1.23	6.74
958	1.04	2.81
1023	0.98	1.92

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ORNL-LR-DWG 55089 Fig. 4. Electrical Resistivity of UO_2 as a Function of Temperature.

It was found during in-pile testing that superimposition of the thermoelectric emf on the potential drop across the UO_2 plate made measurements more difficult, particularly at higher temperatures. At temperatures in excess of about 600–700°C, the thermoelectric emf was equal to or greater than the potential drop, and hence the results obtained by reversing the current flow were highly divergent. Attempts were made to utilize this thermoelectric emf to measure the resistance of the UO_2 plates, using the classical method for measuring the internal resistance of a battery. The results of the in-pile resistivity measurements and thermoelectric power are tabulated in Tables 2 and 3. The in-pile resistivity data are also plotted in Fig. 4. An apparent discrepancy in the temperature measurements from one of the thermocouples was detected, which has introduced some uncertainty in the ΔT measurements.

A major difficulty encountered in the in-pile measurements, not encountered in

Table 2. In-Pile Measurement of Resistivity

Temperature (°K)	Reciprocal Temperature [(°K) ⁻¹]	ρ , Resistivity (ohm-cm)
$\times 10^{-3}$		
nvt = 0		
323	3.09	2080
nvt < 10 ¹⁸		
473	2.11	58.5
525	1.9	29.5
608	1.65	24.3
693	1.44	7.3
778	1.29	4.4
803	1.25	3.1
856	1.17	7.7
976	1.03	3.6
993	1.01	1.1
1113	0.90	7.15
1123	0.88	5.9
1123	0.88	1.1
1151	0.87	4.9
1155	0.86	3.3
1175	0.85	2.0
1245	0.80	5.1
nvt $\geq$ 10 ¹⁸		
403	2.48	55.0
438	2.28	31.2
438	2.28	30.2
513	1.95	9.9
513	1.95	10.3
568	1.76	5.0
569	1.76	4.9
983	1.02	0.6
993	1.01	0.8
1046	0.96	0.4
1143	0.87	0.7
Internal Resistance Battery Method		
983	1.02	1.0

out-of-pile measurement, was the extreme and rapid fluctuations of both the potential and the current. The continuous fluttering of the meters made accurate readings of emf and current extremely difficult. At times these extremely rapid fluctuations would die down and measurements could be made with reasonable accuracy. Also, at times, the K₂ potentiometer would become completely insensitive, as if a very large resistance had been introduced in the system either due to bad contacts, high resistance of UO₂, or formation of high-resistance coating on the UO₂ surface.

An analysis of traces obtained by continuous monitoring of the current and voltage indicated that the system seemed to attain an equilibrium if maintained under constant temperature and flux conditions. The recording of the thermoelectric emf under such conditions was relatively smooth. The superimposition

of an external emf and the resultant flow of current through the UO₂ seemed to throw the system off balance, and a higher noise-to-signal ratio was obtained. Immediately after a flux or temperature change, the fluctuations

Table 3. Thermoelectric Power of System
Pt-UO₂ During Irradiation

Temperature (°K)	Thermoelectric Power ($\mu\text{V}/^\circ\text{C}$)
435	130
438	300
513	219
517	85
559	78
564	78
568	227
778	72
823	47
851	150
873	42
901	55
913	63
983	42
983	41
1003	34
1021	49
1023	43
1073	34
1073	63
1155	202
1158	35
1168	159
1173	29

of the emf and current became very pronounced, the emf fluctuations covering as wide a range as 5-7 mv from a mean of 5 mv. With passage of time the fluctuations subsided, so that normal measurements could be made.

From fission gas release data, it has been noted that the bulk of xenon measured has been released as iodine, and a considerable iodine plate-out has been observed.⁴ Also, following a withdrawal and reinsertion of the fuel, the iodine buildup reaches equilibrium after the lapse of a few days. A deposition of such an iodine or iodide layer between the UO₂ and the electrical contact and

its subsequent removal, or the formation and breakdown of some oxide or other layer on the surface, could also lead to the behavior pattern observed. It does not seem very probable that the thermocouples physically shift due to vibrations, because, given sufficient time, the fluctuations decrease considerably. Moreover, superimposing an external emf leads to pronounced unidirectional voltage fluctuations (the direction depending on the direction of the superimposed emf) with, however, an almost steady current. These facts favor a variation of the surface characteristics, as would result from formation and decomposition of mono- or multimolecular layers with consequent shifts and motion of grain boundaries. However, the effects of mechanical vibrations cannot be ruled out completely, since fluctuations resembling bad contacts have been observed on changing cooling air flow outside the capsule.

It is interesting that the in-pile measurements of resistivity fall into two more or less distinct groups. Measurements made up to irradiation of about 10^{18} nvt fall very close to the values obtained on the unirradiated material. Measurements made during irradiations greater than 10^{18} fall on an almost parallel line of lower resistivity. The curves obey the general equation $\rho = A \exp (-\Delta E/2kT)$. The "activation energies" calculated for the unirradiated and irradiated material are:

	Activation Energy (ev)
Unirradiated, room temp to 200°C	0.83
Unirradiated, 400 to 750°C	0.65
Irradiated up to 10^{18} nvt (150-700°C)	0.5
Irradiated beyond 10^{18} nvt (150-800°C)	0.5

The measurements of the thermoelectric power have been somewhat erratic and the reasons for this behavior are not clear, yet they have shown that the material behaved like an n-type conductor.

The values of the activation energies calculated are in good agreement with values obtained¹ for n-type material. Willardson et al.¹ have interpreted the activation energy of 0.9 ev obtained on n-type material (metal excess) as due to both donors and acceptors (UO_{2+x} and U_3O_{7-z}) and thereby having the effect of reducing the free carrier density. It is not unreasonable to suppose that the high-density sintered UO_2 used in the present experiment also contained phases like UO_{2+x} and U_3O_{7-z} .

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