CHARACTERIZATION OF MIXED-METAL OXIDES USING SYNCHROTRON-BASED TIME-RESOLVED X-RAY DIFFRACTION AND X-RAY ABSORPTION SPECTROSCOPY

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ABSTRACT

Experiments are described showing the utility of synchrotron-based time-resolved x-ray diffraction (TR-XRD) and x-ray absorption near-edge spectroscopy (XANES) for characterizing the physical and chemical properties of mixed-metal oxides that contain Mo and a second transition metal (Fe, Co or Ni). TR-XRD was used to study the transformations that occur during the heating of a $\text{FeMoO}_4/\text{Fe}_2(\text{MoO}_4)_3$ mixture and the $\alpha \rightarrow \beta$ phase transitions in CoMoO_4 and NiMoO_4 . The Mo L_{II} - and O K-edges in XANES are very useful for probing the local symmetry of Mo atoms in mixed-metal oxides. The results of XANES and density-functional calculations (DMol3, DFT-GGA) show large changes in the splitting of the empty Mo 4d levels when going from tetrahedral to octahedral coordinations. XANES is very useful for studying the reaction of H_2 , H_2S and SO_2 with the mixed-metal oxides. Measurements at the S K-edge allow a clear identification of S, SO_2 , SO_3 or SO_4 on the oxide surfaces. Changes in the oxidation state of molybdenum produce substantial shifts in the position of the Mo L_{II} - and M_{III} -edges.

INTRODUCTION

Mixed-metal oxides play a relevant role in many areas of materials science, physics, chemistry, and the electronic industry. In principle, the combination of two metals in an oxide matrix can lead to materials with novel properties and a superior performance in technological applications. In this article we illustrate the use of two synchrotron-based techniques, time-resolved x-ray diffraction (TR-XRD) and x-ray absorption near-edge spectroscopy (XANES), in the characterization of the physical and chemical properties of mixed-metal oxides. We focus on the behavior of compounds of the MeMoO_4 type (Me= Fe, Co or Ni), which result from adding oxides of the MeO type to MoO_3 [1] and constitute an interesting group of materials due to their structural, electronic and catalytic properties [1-5].

Investigations at Brookhaven National Laboratory has established the feasibility of conducting sub-minute, time-resolved x-ray diffraction experiments under a wide variety of temperature and pressure conditions ($-190^\circ\text{C} < T < 900^\circ\text{C}$; $P < 45\text{ atm}$) [6]. This important advance results from combining the high intensity of synchrotron radiation with new parallel data-collection devices [6]. Examples of problems studied to date with TR-XRD include the hydrothermal synthesis and thermal dehydration of zeolites, the binding of substrates and inhibitors in porous catalytic materials, and phase transformations in oxides [4,6,7]. X-ray absorption near-edge spectroscopy has emerged as a powerful tool for studying the electronic and structural properties of solids [8]. Recently, we have shown that the results of theoretical calculations based on density-functional theory (DFT) can be very useful for explaining the XANES spectra of oxides [7].

EXPERIMENTAL

The time-resolved powder diffraction patterns of the MeMoO_4 compounds were collected on beamline X7B of the National Synchrotron Light Source (NSLS) at Brookhaven National Laboratory. The diffraction patterns were accumulated on a flat image plate (IP) detector and retrieved using a Fuji BAS2000 scanner. The XANES spectra of the mixed-metal oxides were collected at the NSLS on beamlines U7A (Mo M-edges and O K-edge) and X19A (Mo L-edges and S K-edge). The measurements at X19 were performed in the “fluorescence-yield mode”, whereas the data at U7A were acquired in the “electron-yield mode”. The theoretical spectra for the Mo L-edges [7] were calculated using DFT and commercial versions of the CASTEP [9] and DMol3 [10] codes available from Molecular Simulations Inc.

The different isomorphs of the MeMoO_4 compounds were prepared following the methodology described in ref. [3]. In a set of experiments, the molybdates were exposed to H_2 or SO_2 in a RXM-100 instrument from Advanced Scientific Designs. The reduction with hydrogen was carried out in a flow-reactor with a 15% H_2 /85% N_2 mixture (flow rate = 50 cm^3/min) and the temperature was ramped from 40 to 650 or 800 $^\circ\text{C}$ (heating rate = 20 $^\circ\text{C}/\text{min}$). Pure oxide powders were exposed to SO_2 in a reaction cell (“batch reactor mode”) at 50 $^\circ\text{C}$ for 15 min with a constant SO_2 pressure of 10 Torr.

RESULTS

Phase transitions and studies using time-resolved x-ray diffraction. Under atmospheric pressure the MeMoO_4 compounds can adopt a low temperature α -phase in which Mo and the second metal (Fe, Co or Ni) are in an octahedral coordination, or a high temperature β -phase in which Mo is in a tetrahedral environment with the second metal remaining in octahedral coordination [1,4,7]. Using TR-XRD we monitored the $\alpha \rightarrow \beta$ phase transitions in CoMoO_4 and NiMoO_4 [4,7]. Figure 1 shows that in the case of the cobalt molybdate the transformation occurs

Fig 1 Time-resolved x-ray diffraction patterns for the $\alpha \rightarrow \beta$ transition in CoMoO_4 . Heating rate = 1.6 $^\circ\text{C}/\text{min}$

between 330 and 410 °C. For the nickel molybdate, the transformation takes place at much higher temperatures: 530-630 °C [4,7]. There are no intermediates in the $\alpha \rightarrow \beta$ transitions for these molybdates: the diffraction lines for α -MeMoO₄ disappear and simultaneously the lines for β -MeMoO₄ appear. First-principles density-functional calculations carried out with the CASTEP program (DFT-GGA level) show that α -CoMoO₄ is ~ 5 kcal/mol more stable than β -CoMoO₄ [7,11]. On the other hand, α -NiMoO₄ is ~ 9 kcal/mol more stable than β -NiMoO₄ [7]. For the $\alpha \rightarrow \beta$ transition in NiMoO₄, the DFT calculations predict an energy barrier of ~ 50 kcal/mol [7]. The results of TR-XRD indicate that the α -NiMoO₄ $\rightarrow$ β -NiMoO₄ transformation follows a kinetics of first order with an *apparent* activation energy of ~ 80 kcal/mol [4,7].

In the case of the iron molybdate, the procedure followed for the synthesis of the compound [3] in some situations can yield a mixture of β -FeMoO₄, Fe₂(MoO₄)₃ and an amorphous FeMo_xO_y phase due to the possibility of changes in the oxidation states of the second metal and molybdenum (i.e. Fe²⁺ $\rightleftharpoons$ Fe³⁺ and Mo⁶⁺ $\rightleftharpoons$ Mo⁵⁺). The evolution of this type of system as a function of temperature can be monitored using TR-XRD. In Figure 2 diffraction lines are observed for β -FeMoO₄ [1,3] and Fe₂(MoO₄)₃ [12] at low temperatures (< 300 °C). The relative intensity of these lines indicates that the Fe₂(MoO₄)₃/ β -FeMoO₄ ratio in the sample is ~ 0.2. Upon heating one sees the disappearance of the lines for Fe₂(MoO₄)₃ at temperatures between 350 and 430 °C. Above 430 °C, only diffraction lines for β -FeMoO₄ are observed but there is a continuous increase in the intensity of these lines. On the basis of the TR-XRD results, one can conclude that Fe₂(MoO₄)₃ and the amorphous FeMo_xO_y phase transform into β -FeMoO₄ at high temperatures (350-700 °C).

In addition of being a valuable technique for studying phase transitions, we have found that TR-XRD is also very useful for studying structural changes that occur during cycles of reduction (with H₂)/oxidation(with O₂) or during the reaction of H₂S with the molybdates [11,13].

Studies using x-ray absorption near-edge spectroscopy. The left-side panel in Figure 3 shows Mo L_{II}-edge XANES spectra for MoO₃, the β -FeMoO₄/Fe₂(MoO₄)₃/FeMo_xO_y mixture of Figure 2 (T= 25 °C), and β -FeMoO₄. These spectra involve transitions from the 2p core levels into the

Fig 2 Time-resolved x-ray diffraction patterns for the heating (1.8 °C/min) of a mixture of β -FeMoO₄, Fe₂(MoO₄)₃, and an amorphous FeMo_xO_y phase.

Fig 3 Left panel: Mo L_{II}-edge XANES spectra for MoO₃, a mixture of β-FeMoO₄/Fe₂(MoO₄)₃/FeMo_xO_y, and β-FeMoO₄. Right panel: Calculated DOS for the empty Mo 4d bands of β-FeMoO₄ (DMol3, DFT-GGA results).

empty 4d orbitals of Mo. The right-side panel in Figure 3 displays the calculated density of states (DMol3, DFT-GGA) for the empty 4d bands of Mo in β-FeMoO₄. In octahedral and tetrahedral environment, the Mo 4d orbitals split in two sets of bands with t and e symmetry as a result of the ligand field generated by the oxygens around the Mo atoms. In a tetrahedral coordination the set with e symmetry is more stable than the set with t symmetry (Fig 3). The opposite is valid for an octahedral coordination [7]. The energy separation between the two sets of bands is larger in an octahedral configuration. In Figure 3, the XANES spectra of the oxide mixture and β-FeMoO₄ show a peak-to-peak separation of ~ 1.7 eV which is typical of Mo in a tetrahedral environment (see Table I). On the other hand, in the XANES spectrum for MoO₃, the peak-to-peak separation is ~ 3.4 eV, a value representative of Mo in octahedral coordination (Table I). We have found that the peak-to-peak separations in the Mo L_{II}-edge and O K-edge of the XANES spectra for the molybdates give direct and reliable information about the coordination of Mo in these systems (see Table I). The case of the O K-edge spectra is particularly interesting since, due to orbital mixing, one can have electron transfer from the O 1s orbitals into the empty orbitals of both metals

Table I: Splitting of Mo L_{II}- and O K-edges in XANES

Compound	Mo L _{II} -edge (eV)	O K-edge (eV)
MoO ₃	3.2 (O _h)	3.4
α-NiMoO ₄	3.1 (O _h)	3.2
α-CoMoO ₄	2.8 (O _h)	3.2
β-FeMoO ₄	1.6 (T _d)	1.8
β-CoMoO ₄	1.5 (T _d)	1.9
β-NiMoO ₄	1.7 (T _d)	1.9
β-MgMoO ₄	1.6 (T _d)	1.9

present in a molybdate. The results of DFT calculations indicate that this transfer should occur mainly into the unoccupied Mo 4d bands [7,11], and indeed the XANES experiments show that the line-shapes in the O K-edge spectra track very well those observed in the corresponding Mo L_{II} -edge spectra [4,7,11].

XANES is very useful to study electronic perturbations induced on the molybdates by reaction with H_2 , H_2S or SO_2 . As an example, Figure 4 shows Mo L_{II} -edge XANES spectra recorded after partially reducing samples of β -FeMoO₄ by exposing them to a stream of 15%- H_2 /85%- N_2 . Spectra of temperature programmed reduction (TPR) indicate that the molybdate reacts only with a minor amount of H_2 at temperatures below 650 °C, and the consumption of hydrogen becomes larger at temperatures between 650 and 800 °C [3]. This is consistent with the trends seen in Figure 4. In this figure, the reaction with H_2 induces a shift of the Mo L_{II} -edge features towards lower photon energies which indicates a reduction in the oxidation state of Mo in the sample.

The MeMoO₄ compounds are precursors of hydrodesulfurization catalysts [3] and can be used as sorbents of SO_2 . Changes in the position of the Mo M_{III} -edge in XANES have proven to be quite useful for monitoring the degree of sulfidation of Mo during the reaction of nickel molybdate with H_2S [13,14]. Measurements at the S K-edge show the formation of metal sulfides and sulfates as a result of the reaction of hydrogen sulfide with iron, cobalt and nickel molybdates [11,13,14]. Figure 5 displays a S K-edge XANES spectrum collected after dosing SO_2 to β -FeMoO₄ at 50 °C. There are two clear peaks that correspond to SO_3 (~ 2478 eV [14]) and SO_4 (~ 2482 eV [14]) produced by reaction of SO_2 with O atoms from the lattice of the oxide substrate. A very weak feature is seen around 2474 eV that could be attributed to a very small amount of chemisorbed SO_2 . There was no dissociation of the adsorbed SO_2 and no peaks are seen for metal sulfides or pure sulfur in the XANES spectrum. Our experiments indicate that β -FeMoO₄ is much more reactive towards SO_2 than MoO₃, but less reactive than Fe₂O₃. The main product of the interaction of SO_2 with β -FeMoO₄ and Fe₂O₃ is SO_4 . This sulfate species is stable on the oxides up to temperatures well above 300 °C.

In summary, the experiments described above show that the combination of two synchrotron-based techniques like time-resolved x-ray diffraction and x-ray absorption spectroscopy can provide detailed information about the structure, phase transitions, electronic properties and

Fig 4 Mo L_{II} -edge XANES spectra acquired before and after exposing β -FeMoO₄ to a 15%- H_2 /85%- N_2 mixture at 40-650 °C or 40-800 °C

Fig 5 Adsorption of SO₂ on β -FeMoO₄ at 50 °C

chemical activity of mixed-metal oxides. Following this approach one can get a better understanding of the behavior of this important type of materials.

ACKNOWLEDGMENT

This work was supported by the US Department of Energy (DE-AC02-98CH10886), Basic Energy Sciences, Chemical Science Division. Travel grants from the American Chemical Society made possible visits of J.L. Brito to BNL and J.A. Rodriguez to IVIC. The facilities used at the NSLS (X7B, X19 and U7A beamlines) are supported by the Chemical and Materials Science Divisions of the US Department of Energy.

REFERENCES

1. A.W. Sleight and B.L. Chamberland, *Inorg. Chem.* **7**, p. 1672 (1968).
2. J. Zou and G.L. Schrader, *J. Catal.* **161**, p. 667 (1996).
3. J.L. Brito and A.L. Barbosa, *J. Catal.* **171**, p. 467 (1997).
4. J.A. Rodriguez, S. Chaturvedi, J.C. Hanson, A. Alborno and J.L. Brito, *J. Phys. Chem. B*, **102**, p. 1347 (1998).
5. C. Mazzochia, C. Aboumrad, C. Diagne, E. Tempesti, J.M. Herrmann and G. Thomas, *Catal. Lett.* **10**, p. 181 (1991).
6. P. Norby and J.C. Hanson, *Catal. Today*, **39**, p. 301 (1998).
7. J.A. Rodriguez, J.C. Hanson, S. Chaturvedi, A. Maiti and J.L. Brito, *J. Chem. Phys.* **112**, in press (2000).
8. J.G. Chen, *Surf. Sci. Rep.* **30**, p. 1 (1997).
9. M.C. Payne, D.C. Allan, T.A. Arias and J.D. Johannopoulos, *Rev. Mod. Phys.* **64**, p. 1045 (1992).
10. B. Delley, J. Schefer and T. Woike, *J. Chem. Phys.* **107**, p. 10067 (1997).
11. J.A. Rodriguez, J.C. Hanson and J.L. Brito, to be published.
12. A.W. Sleight and L.H. Brixner, *J. Solid State Chem.* **7**, p. 172 (1973).
13. J.A. Rodriguez, S. Chaturvedi, J.C. Hanson and J.L. Brito, *J. Phys. Chem. B*, **103**, p. 770 (1999).
14. S. Chaturvedi, J.A. Rodriguez and J.L. Brito, *Catal. Lett.* **51**, p. 85 (1998).