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SUPERPLASTICITY AND JOINING OF ZIRCONIA-BASED CERAMICS

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Abstract

Steady-state creep and joining of alumina/zirconia composites containing alumina volume fractions of 20, 60, and 85% have been investigated between 1250 and 1350°C. Superplasticity of these compounds is controlled by grain-boundary sliding and the creep rate is a function of alumina volume fraction, not grain size. Using the principles of superplasticity, pieces of the composite have been joined by applying the stress required to achieve 5 to 10% strain to form a strong interface at temperatures as low as 1200°C

Introduction

The high-temperature plastic deformation of alumina ceramics has been extensively reported in the literature [1-3]. Achieving high ductility in pure alumina is difficult because of rapid static and dynamic grain growth that is accompanied by large strain hardening and severe cavitation. The relatively high grain-boundary energy (GBE) in pure alumina [1] provides a high driving force for grain growth. The high GBE is also responsible for the relatively low cohesive strength that results in severe cavitation [1].

Suppression of grain growth can be achieved through low temperature sintering or/and the use of additives. Although the solubility of a second phase in alumina is very limited, the use of additives such as MgO, ZrO₂ or Y₂O₃ inhibit the grain growth by segregation of these cations to the alumina grain boundaries [1,4].

In the case of alumina/zirconia, microstructure and mechanical properties has been widely studied [1, 4-7] and it has been shown that the equiaxed fine microstructure is preserved during deformation improving the ductility of this compounds [1,6,7].

In this paper we will characterize the mechanism controlling the plasticity as a function of the alumina volume fraction and of the phase distribution.

It is well known that grain boundary sliding (GBS) controls the macroscopic deformation of these compounds [6]. When two pieces of ceramics are compressed in the superplastic regime, the grains in the interface rotated with respect to each other producing interpenetration and a strong junction of both pieces [8-10]. Therefore, superplastic flow has been used to join alumina/zirconia ceramics and the conditions and strength of the junctions will be also discussed in this paper.

Experimental conditions

Alumina/zirconia composites with alumina volume fractions of 20, 60 and 85 %, and with different phase distribution, were investigated. The compositions are designated as ZT20A, ZT60A and ZT85A. Fabrication processes of the ZT60A and ZT85A were described in detail in references 11 and 12, respectively.

The ZT20A was fabricated as following: nanocrystalline ZrO₂-4 wt% Y₂O₃ (Y-TZP) powder with mean particle size of 10 nm was prepared by chemical precipitation from the

nitrate salt solutions, as previously described [13]. The commercial nanocrystalline alumina powder (Neomat) was composed of transition alumina (θ and δ) with a mean particle size of 10 nm. The Y-TZP and alumina powders were separately dispersed ultrasonically in ethanol; the pH of the dispersions were adjusted to 1.5 and 7 respectively. The two slurries were ball milled for 24 h before mixing; the resultant pH was between 1.5 and 2. At this stage two different mixing and sedimentation routes were adopted. In the case of the sedimentation (S designated samples), ammonia solution was added to the final mixed slurry to change its pH to 10.5. Then the slurry was left to sediment overnight; the latter was collected and washed with ethanol and dried at 200°C for 1 day. In the case of the milled (M designated samples), the mixed slurry was dried using a heating plate with continuous mixing to avoid sedimentation. Once the slurry was dried, the wet powder was dried at 200°C for 1 day. Cylindrical pellets of 15 mm in diameter and length were uniaxially cold pressed followed by cold isostatic pressing at 200 MPa. Finally, the pellets were sintered at 1600°C for 2 min using heating and cooling rates of 5 °C/min and 20 °C/min, respectively.

The average grain size of zirconia and alumina were determined by electron microscopy and the final density was determined by the Archimedes method. The phase content was determined by X-ray diffraction.

The samples were cut into parallelepiped shape having dimensions of 6x3x3 mm and compressed in air at temperatures between 1200 and 1350°C at constant load (creep experiments) or at constant cross-head speed in a Instron machine. In these latter experiments a steady-state stress, as defined by a zero work-hardening rate, was obtained after 1-2 % of deformation. The stress of the plateau is the flow stress for Instron experiments. The stress ranged between 5 and 500 MPa and the initial strain rate varied between 10^{-7} and 10^{-4} s⁻¹.

For the junction, pieces of ZT20AM and ZT60A were compressed in an Instron machine to a strain of 5 and 10 % at crosshead speed of 5 μ m/min that corresponded to an initial strain rate, $\dot{\epsilon} \sim 10^{-5}$ s⁻¹, for the dimensions of the pieces joined. The temperatures employed were between 1200 and 1300°C.

The junction strength was measured from cracks produced by a 100 N Vickers indenter whose diagonal was along the interface.

Experimental results and discussion

The ZT20AS specimens were characterized by a more homogeneous distribution of

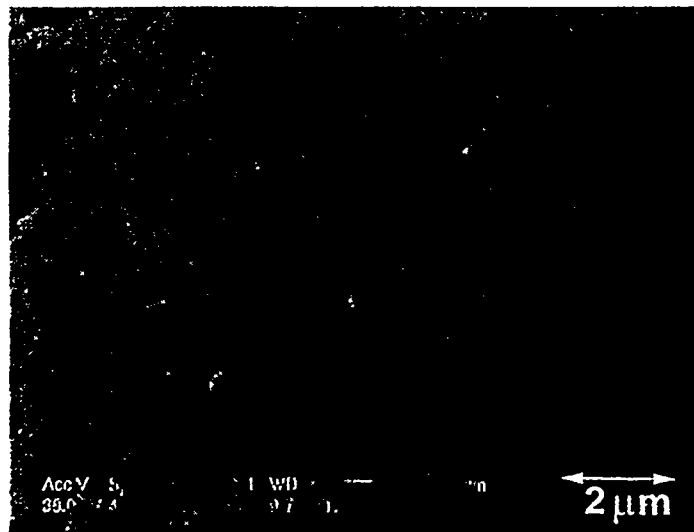


Fig 1. Microstructure of the as-received ZT20AS

both alumina and zirconia grains compared with the ZT20AM specimens. Fig. 1 is a typical micrograph of the microstructure of a ZT20AS ceramics. Scanning electron microscopy (SEM) [11,12] revealed that the ZT60A and ZT85A consisted of a very homogeneous distribution of both phases. The characteristics of these ceramics are listed in Table I. No differences in the grain size of both phases were found in the two types of ZT20A.

The superplastic behavior of alumina/zirconia composites has been analyzed using the equation:

$$\dot{\epsilon} = A\sigma^n d^{-p} \exp\left(\frac{-Q}{RT}\right) \quad (1)$$

where $\dot{\epsilon}$ is the strain rate, σ is the applied stress, d the grain size, R and T have their usual meaning, and n , p and Q are the creep parameters characterizing the mechanism controlling the creep behavior.

TABLE I. Grain size of both phases of the materials used in this work.

Material	Al ₂ O ₃ grains (μm)	3Y grains (μm)
ZT20A	0.8	0.6
ZT60A	0.7	0.4
ZT85A	1.3	0.5

Figs. 2a and 2b are creep curves typical of all materials tested. From stress and temperature changes it is possible to determine n and Q as shown. There are no

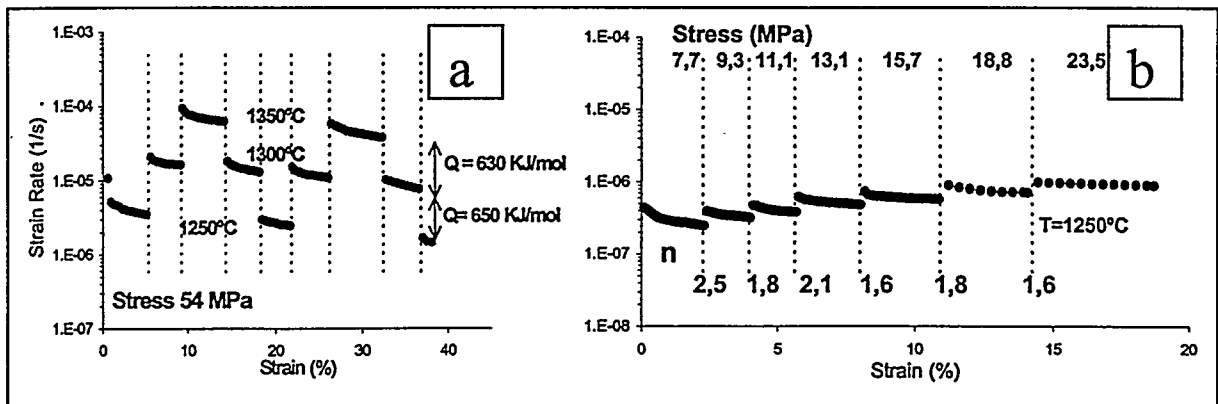


Fig 2. Typical creep curves to determine the activation energy Q (a) and the parameter n (b).

discontinuities in the extrapolation of the creep data for a given temperature, despite the fact that T was changed several times. Therefore, the creep data indicate that the microstructure remained constant during the test without grain growth. No grain-size exponent was determined because the grain size was not varied.

The effect of T and σ is shown in Fig. 3, which is a plot of $\log \dot{\epsilon}$ vs $\log \sigma$ for three different temperatures. The excellent agreement between the creep and Instron data is further evidence that microstructure was unchanged and that the tests were performed in steady state. That is, the final state is path independent. The slopes of these curves are close to 2 decreasing at the higher stresses. Q and n were determined from curves as in fig. 2. Values of n at low and intermediate stresses of 1.9 ± 0.1 and of 1.4 ± 0.1 at the higher stresses were measured. The values of n and σ are independent of the composition and phase distribution. Fig. 4 shows that there is no influence of the phase distribution. As shown in fig. 4, the value of 2 at stresses as low as 5 MPa indicates that the threshold stress is zero in these compounds. The same behavior has been reported in Y-TZP with low purity [14,15].

A n value of 2 has been also measured by Wakai et al. [6] for stresses below 80 MPa and by Clarisse et al [7]. However, these last authors found that n tends to 1 when the stress increases.

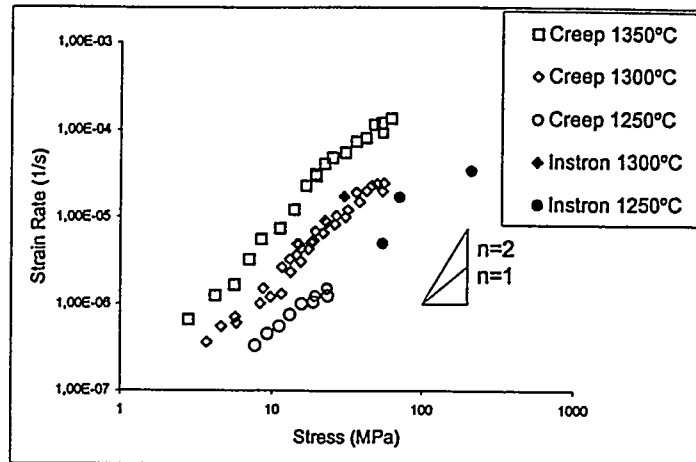


Fig. 3: Plot of $\log \dot{\epsilon}$ versus $\log \sigma$ for different temperatures.

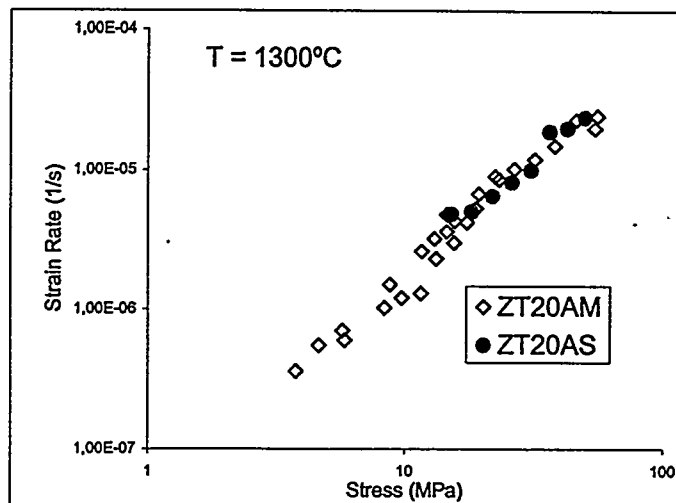


Fig.4: Influence of phase distribution in the mechanical behavior.

The Q value was weakly dependent on stress. A Q -value of 680 ± 20 kJ/mol at the lower stresses and of 640 ± 10 kJ/mol for stresses higher than 50 MPa were measured. Although the activation energy for non-doped alumina [7] and pure Y-TZP [14] ranged between 450-500 kJ/mol, a higher value of Q has been systematically found for the different alumina/zirconia composites [6,7,16]. The higher Q has been attributed to the interdiffusion of Zr to the Al_2O_3 phase resulting in the modification of the defect formation enthalpy [7].

As already mentioned, the suppression of grain growth in alumina can be achieved through the use of additives as zirconia. The zirconium, in alumina, precipitates to the interface [1,6,7] lowering the GBE and improving the stability of the microstructure and the superplastic formability of these compounds. At the same time, the segregation of Zr^{4+} at the grain boundaries reduces the aluminum diffusivity. For instance, the Al^{3+} in these compounds changes its diffusivity with respect to pure alumina, increasing the activation energy from 500 kJ/mol in pure alumina to 700 kJ/mol in alumina/zirconia composites [5,6]. No variation in Q with the zirconia/alumina composition has been detected. This result is compatible with stability of the microstructure of these compounds when the second phase is present at

relatively low volume fractions [1,7], as a consequence of the precipitation of zirconium at the grain boundaries and the change of the diffusivity.

In table II the creep resistant for the three compositions used in this work are presented together with the results in the literature, for a stress of 50 MPa and two different temperatures of 1300 and 1350°C. It is obvious that the mechanical behavior of these compounds depend on the alumina volume fraction and not of the grain size. For instance, in our work, ZT20A and ZT60A have almost the same grain size, however the strain rate is one order of magnitude lower for ZT60A.

TABLE II. Strain Rates at a stress of 50 MPa for different compositions and two temperatures.

Material	Authors	Al ₂ O ₃ grains (μm)	3Y grains (μm)	Strain Rate (1/s) 1300°C	Strain Rate (1/s) 1350°C
ZT20A	Present work	0.8	0.6	3×10^{-5}	1×10^{-4}
ZT20A	[7]	0.6	0.6	4×10^{-5}	2×10^{-4}
ZT50A	[7]	1.1	0.8	6×10^{-6}	3×10^{-5}
ZT50A	[6]	0.6	0.5	6×10^{-6}	2.5×10^{-5}
ZT60A	Present work	0.7	0.4	3×10^{-6}	-
ZT69A	[6]	1.0	0.6	1×10^{-6}	7×10^{-6}
ZT80A	[7]	1.4	0.7	-	7×10^{-6}
ZT85A	Present work	1.3	0.5	Broken	1×10^{-6}
ZT86A	[6]	1.0	0.5	1×10^{-6}	7×10^{-6}

After deformation, the microstructure remained unchanged (not shown here) with the same grain size and shape as measured by the form factor. These characteristics are typical when GBS is the main mechanism controlling plasticity.

Based on the GBS mechanism, two types of junction have been produced. In one case, two pieces of TZ20A were joined and in the other case one piece of TZ20A was joined with another of TZ60A. Fig. 5 shows the interface of the junction. The interface has almost

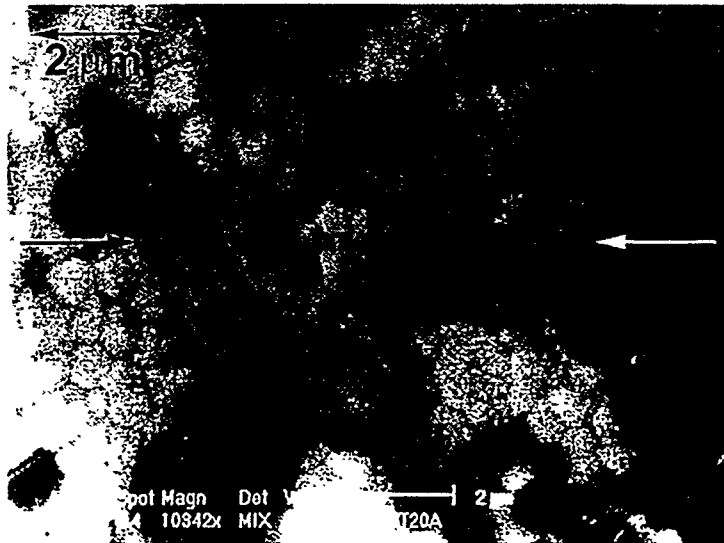


Fig. 5: Junction of two pieces of ZT20A. The arrows indicate the location of the interface.

disappeared, showing that the grains of one part interpenetrate the other. The junction is formed by rotation of the grains of both pieces. The crack lengths of the Vickers indentations developed along and perpendicular to the interface in the case of ZT20A/ZT20A are

280 ± 30 μm and 270 ± 20 μm. In the case of ZT20A/ZT60A, the cracks did not propagate along the interface but deviate into the brittle material (ZT60A). These results indicate that the pieces were perfectly joined.

Conclusion

It is shown that the ZTA composites due to the refinement of the microstructure of the alumina and to the presence of YTZP became more damage tolerant than pure alumina although the strength is controlled by this phase.

These compounds can be deformed superplastically at temperatures as low as 1250°C. The main mechanism of the plastic deformation at high temperature of these compounds is GBS accommodated by diffusion process, which are controlled at their turn by the precipitation of Zr at the grain boundaries.

Based on the GBS mechanism, different pieces of ZTA have been joined at temperatures as low as 1200°C for a period of time as long as 30 min. The cracks developed along the interface after Vickers indentation show that the pieces were perfectly joined.

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