NON-SHRINKING SOL-GEL TYPE POLYMERS BY RING OPENING POLYMERIZATION

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Introduction

Hydrolysis and condensation of alkoxysilanes to form highly crosslinked siloxane gels has become synonymous with sol-gel polymerization. However, application of these sol-gel systems has been limited by shrinkage associated with evaporation of the solvent needed for monomer/water miscibility and the resulting condensation products formed during polymerization. An attractive strategy for reducing shrinkage is to replace step growth polymerization in sol-gel processing of alkoxysilanes with a chain growth process that has proven to be an effective means for reducing or, as with the polymerization of spiroorbocarbonates, completely eliminating shrinkage in linear, hydrocarbon polymers. Here, we report ring-opening polymerization (ROP) of arylene and alkylene bridged disilaoxacyclopentanes (scheme 1), as well as their copolymerization with the simple 2,2,5,5-tetramethyl-2,5-acyclopentane (4), to make sol-gel type materials and their application in encapsulation of microelectronics.

Scheme 1

<table>
<thead>
<tr>
<th>R = Ph</th>
<th>R = Et</th>
<th>R = n-Bu</th>
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<tbody>
<tr>
<td>Me-O-Si-Me</td>
<td>Me-O-Si-Me</td>
<td>Me-O-Si-Me</td>
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<tr>
<td>Me-Si-Me</td>
<td>Me-Si-Me</td>
<td>Me-Si-Me</td>
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<tr>
<td>TBAH (0.1 mol%)</td>
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Scheme 2

Experimental Section

Monomer syntheses. Scheme 2 shows the synthetic scheme for preparation of the phenylene-bridged disilaoxacyclopentane, 1, which also applies to the preparation of the ethylene (2) and butylene (3) bridged precursors. All compounds have been spectroscopically characterized and experimental details will be published in future publications.

Polymer syntheses. Polymerization of 1, a white solid, can be achieved either neat as a melt polymerization using octadecylamine as catalyst, or as a solution in THF (1M) using Bu₄NOH (TBAH) as catalyst. Rapid polymerization of all monomers can be achieved when polymerized as a copolymer system with 4, using TBAH as catalyst.

Results and Discussion

Monomer 1 can readily be polymerized either in THF of when dissolved in monomer 4 (used as a solvent/co-monomer) to give transparent, yellow-tinted gels which show no visible shrinkage during polymerization. Furthermore, because the copolymer is prepared without solvent, total shrinkage is kept to less than 5% compared with 50-90% for traditional sol-gels. Because the resulting gels are insoluble, solid state NMR and infrared spectroscopy are used in their characterization.

Scheme 2

Figure 1. ²⁹Si NMR spectra for monomer 1 and its homopolymer. NMR resonances for 1 (left spectrum) are 21.9 and 20.9 ppm and those for the polymer (right spectrum) are 9.8 and 5.5 ppm.

Figure 2. IR spectra of monomer 1 (bottom spectrum) and its homopolymer (top spectrum).

By using 4 as both a solvent and a reactive monomer in the polymerization, the entire solution is converted into a network polymer and shrinkage can be further reduced. Exothermic polymerization of a mixture of 4 and 1 (80:20 by weight) with TBAH (0.2 mol% based on 4) gives rise to a transparent, crack-free gel within seconds of mixing. Higher catalyst concentration (1 mol% based on 4) leads to generation of enough heat during polymerization to cause bubbles to form (presumably due to monomer volatility).
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volatilization) and become entrapped within the resulting gel. Solid state $^{13}$C and $^{29}$Si NMR reveal a polymer composition representative of starting comonomer ratio (80:20) and no detectable unreacted monomer in the final gel.

Thermal gravimetric analysis of the polymers (figure 3) revealed relatively robust materials in homopolymer of 1 and its copolymer with 4 compared to the linear polymer of 4, which begins to degrade at 250 °C and is completely depolymerized by 500 °C. In contrast, the homopolymer of 1 begins to decompose at 375 °C and leaves a ceramic residue (17.1%). The copolymer shows an initial weight loss of about 4% followed by complete degradation starting at 500 °C. The addition of the phenylene-bridged crosslinker to the polymer of 4 increases the thermal stability by over 150 °C.

Nitrogen sorption porosimetry and scanning electron microscopy of the homopolymer of 1, prepared in THF, and the copolymer of 1 and 4 revealed no significant micro- or mesoporosity (figure 4). This lack of porosity suggests that the gels are composed of network polymers that are compliant enough to permit collapse of the pores.60

Figure 3. TGA plot comparing the weight loss of homopolymers of 1, 4, and 80:20 copolymer of 4 and 1.

A microelectronic test chip was successfully encapsulated using the ROP of the 80/20 mixture of 4 and 1. The copolymer formed quickly around the test chip as a transparent, slightly yellow resin with only few bubbles forming due to the heat of polymerization (figure 5). No visible cracks or shrinkage of the encapsulant were observed.

Figure 5. SEM photograph of 80:20 copolymer of 4 and 1.

Conclusions

We have designed a new class of cyclic siloxane compounds that behave as sol-gel systems when ring open polymerized using a hydroxide base. These monomers polymerize through chain growth polymerization, unlike conventional alkoxy silane sol-gel precursors, to form sol-gel polymers. They do not require solvent or water for polymerization, show no visible shrinkage or cracking during polymerization and are thermally stable. We have successfully utilized these materials in encapsulation of microelectronics. Current efforts are focused toward expanding this family of ROP monomers and optimization of their mechanical properties.

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References