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Xe-IMPLANTED ALUMINUM***

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HRTEM ANALYSIS OF SOLID PRECIPITATES IN Xe-IMPLANTED ALUMINUM

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Abstract

High-resolution transmission electron microscopy (HRTEM) was carried out to determine the shape and atomic arrangement of solid Xe precipitates in Al. Polycrystalline Al TEM specimens were implanted with 30 keV Xe⁺ at room temperature to a dose of 3×10^{20} ions/m² and were subsequently annealed at 523 K. Below a size of approximately 4 nm in diameter, Xe precipitates are solid with an FCC crystal structure that are mesotactically aligned with the Al lattice. In HRTEM along [011] projection, the difference in the lattice parameters of solid Xe and Al produces a precipitate image dominated by a two-dimensional Moiré pattern that repeats in both the <001> and <111> directions every 3 Al (or 2 Xe) lattice spacings. Multi-slice image simulations, using a three-dimensional atomic model, demonstrates that the precipitates are tetradehedra with faces parallel to the dense {111} planes and the {100} planes. Off-Bragg illumination of the precipitates minimizes Al lattice fringes and generates precipitate images which are in good agreement with the model.

Key words: Solid Xe, implantation, precipitation, High-resolution electron microscopy, Moiré patterns, tetradehedral precipitates, Off-Bragg illumination

1. Introduction

The behavior of inert gases in materials has been studied extensively over the past decades because of material problems associated with the development of fusion and fission reactors [1-2]. The sputter deposition, cleaning and thinning of materials by ions, commonly used in semiconductor industries, have also raised scientific, as well as technical concerns related to the behavior of heavier inert gases implanted by these processes. All the inert gases are insoluble and precipitate. At room temperature, inert gas precipitates less than 4 nm in diameter are crystalline solids mesotactically aligned with the surrounding matrix. Birtcher et. al [3-5] have reported for the case of Kr and Xe in Ni and Al that the difference in lattice parameters between the precipitate and the matrix varies with precipitate size and reaches as much as 50 %, depending on the combination of gases and metal matrix. These observations were carried out using electron diffraction in the transmission electron microscope (TEM). With the advent of high-resolution electron microscopy (HRTEM) with image resolution approaching to 0.1 nm, attempts have been made to determine the shape and topotactical alignment of the inert gas and other insoluble precipitates [6-9]. Gråbæk et al. [10] and Xiao et al. [11] investigated Tl and Pb inclusions in Al, respectively, and showed that the shape of inclusions is an octahedron bounded by {111} truncated by a with small amount of {100}. HRTEM images of inert gas precipitates have not been sufficiently well defined so as to allow precise determination of shapes and atomic arrangements. This paper describes HRTEM results for Xe precipitates in Al and comparison to multi-slice image simulations based on a three-dimensional atomic model relating the precipitate and matrix. Furthermore, the experimental determination of the shape is shown using an off-Bragg illumination technique in HRTEM.

2. Experimental

Thin Al disks, 3 mm in diameter, were mechanically cut from 4-9's starting material and annealed at 673 K. TEM specimens were made by electro-polishing using an electrolyte of $\text{HNO}_3:\text{CH}_3\text{OH}=1:2$ at 253 K. Ion implantations were carried out with 30 keV Xe at room temperature to a dose of 3×10^{20} ions/m². Specimens were then annealed at 523 K for 0.5 h in vacuum to remove radiation damage in the Al matrix and to consolidate the Xe within precipitates. Templier [12] has reported that such annealing results in sharper precipitate images and sharper and more intense electron diffraction from the solid Xe in Al.

Prior to HRTEM, the distribution of Xe precipitates was examined by conventional electron microscopy with a JEM-200CX. HRTEM was performed with a JEM ARM-1000 high-voltage TEM, operated at a voltage of 1 MV with a spatial resolution of 0.13 nm. Most of HRTEM observations were performed with the incident electron beam parallel to a $\langle 110 \rangle_{\text{Al}}$ direction. Some observations were performed with a slightly tilted (less than 1 degree) electron beam condition to produce clear Xe images.

Xe precipitates in an Al lattice were modeled as a tetradecahedron with faces parallel to $\{111\}_{\text{Xe}}$ and $\{100\}_{\text{Xe}}$ embedded in an Al cavity faceted on the same crystal planes. Based on the electron diffraction results, the lattice parameter of the FCC Xe crystal was chosen to be 50 % larger than that of Al. The precipitate size was chosen to be three coincident lattice spacings along each of the $\{111\}$ planes; namely 6 atoms for Xe and 9 atoms for Al. The total thickness of the model is about 7 nm and the surface is normal to a $\{110\}_{\text{Al}}$. Atomic relaxation was not considered at the Al/Xe interfaces. This model was constructed using a crystallographic package [13] on a Macintosh computer. Multi-slice image simulations were carried out to simulate the JEM ARM-1000 at 1000 keV, where the spherical aberration coefficient C_s is 2.6 mm, the spread of defocus Δ is 10 nm and the beam semi-divergence α is 0.3 mrad. The defocus was varied from +10 to -200 nm. The Scherzer defocus for structure

imaging is about -55 nm.

3. Results and discussion

Fig. 1 shows a set of diffraction contrast images of Xe precipitates in the thin region near the edge of a TEM specimen. In a normal bright field (BF) image (Fig. 1a), the precipitates appear as an unclear dark contrast. The FCC crystal structure of the solid precipitates is deduced from the extra spots in the selected area diffraction (SAD) pattern (Fig. 1b). Fig. 1c and Fig. 1d show a comparison between the BF image in the square area in Fig. 1a and a dark field (DF) image taken with a $(111)_{\text{Xe}}$ spot A marked in Fig. 1b. Coincidence of the two types of images indicates that the size of solid precipitates is less than 4 nm.

HRTEM images of Xe precipitates are shown in Fig. 2. The incident electron beam is parallel to a $[011]$ direction. Atoms appear as black dots in this image taken near Scherzer defocus. Two dimensional Moiré patterns along two $\langle 111 \rangle$ and one $\langle 100 \rangle$ directions are seen. Both the Moiré patterns of HRTEM and the spacing of diffraction spots in the SAD pattern (Fig. 1b) indicate that the Xe precipitates have an FCC structure that is mesotactic with the Al matrix with a lattice spacing about 50 % larger than that of Al. An enlarged image in Fig. 2b displays the projection of the atomic structure of a single precipitate. The shape of the precipitate is not precisely determined due to a small variation in the contrast at the interface between the precipitate and the matrix. A similar observation was already reported by Donnelly et al. [9]. In general, the image of atomic positions at the coincident sites at the Al and Xe interface is clear.

There are several possible models [10-11] for insoluble inclusions I in an FCC matrix Mx. In most cases, the inclusion exhibits a cube-cube orientation relation with the matrix, with parallel close-packed planes and directions: $(111)_{\text{Mx}} \parallel (111)_{\text{I}}$ and $[110]_{\text{Mx}} \parallel [110]_{\text{I}}$. The inclusion's

shape is typically a tetradecehedron, with faces parallel to dense $\{111\}$ and to $\{100\}$. The relative surface area of each class of planes reflects their relative surface energies. Typical examples of the results of multi-slice image simulations using such a model are shown in Fig. 3 for the $[011]$ projection and objective defocus Df of -55, -100, -165 nm. The simulated image at Df of -55 nm (Scherzer defocus) is in qualitatively good agreement with the HRTEM image in Fig. 2b. The overlapping of Al and Xe atoms in the image is clearly demonstrated at the coincident atom positions. But white spots appear in the center portion of atomic images at the coincident positions. This may reflect the thickness of the model used in the simulation. Potter et al. [7] have reported the image simulations of a Al/Xe/Al layer structure and found a similar behavior for a partially coherent interface. The most interesting point of the simulation is the clarity of the interface of the precipitate, which was not observed in HRTEM. This is attributed to lack of ideal coincidence at the Al/Xe interfaces and possible shifts of atomic positions within the precipitate.

To experimentally determine the shape of Xe precipitates, HRTEM was performed in off-Bragg electron illumination at different defoci. One such image in $[011]$ projection is shown in Fig. 4 along with the corresponding SAD pattern. This technique is common for the observation of small embedded particles [14]. One must be careful in the interpretation of such images because the defocus value was not Scherzer. This condition of electron beam and defocus caused the disappearance of Al lattice fringes. Two-dimensional projection of the Xe precipitates is determined to be a hexagon defined by 4 $\{111\}$ planes and 2 $\{100\}$ planes. This result is in good agreement with the model proposed in the image simulation. It is important to note that most of the precipitates in Fig. 4 show a high degree of crystalline perfection. However, some of the precipitates in Fig. 4 are not symmetrical along a $(100)_{\text{Xe}}$ planes. The existence of defects in a Xe precipitate in Al was reported previously [9].

Based on the crystalline lattice images, the FCC lattice parameter a_{Xe} for the Xe precipitate is determined from the HRTEM images and the SAD patterns to be 0.595 nm which yields a

molar volume of $3.2 \times 10^{-5} \text{ m}^3$. The equilibrium pressure P in the precipitate is estimated to be about $4.65 \times 10^8 \text{ Pa}$ using the isotherms of Anderson and Swenson [15] extrapolated to 300 K. Assuming a spherical precipitate with a diameter D , the average free surface energy γ which is balanced by the precipitate pressure P is given by $\gamma = PD/4$. The experimental values range from 0.1 to 0.5 N/m which are, in all cases, smaller than the previously determined value 1.14 N/m for Al [16].

4. Conclusions

High-resolution transmission electron microscopy (HRTEM) was carried out on Al TEM thinned specimens implanted with 30 keV Xe^+ at room temperature to a dose of $3 \times 10^{20} \text{ ions/m}^2$ and subsequently annealed at 523 K. The following conclusions are drawn:

- (1) Below a size of approximately 4 nm in diameter, Xe precipitates are solid with an FCC crystal structure that is mesotactic with the Al lattice.
- (2) In HRTEM along a [110] projection, the difference in lattice parameters of solid Xe and Al lattices produces a precipitate image dominated by a two-dimensional Moiré pattern that repeats in both the $\langle 001 \rangle$ and $\langle 111 \rangle$ every 3 Al (or 2 Xe) lattice spacings.
- (3) Multi-slice image simulations using a three-dimensional atomic model indicate that the precipitates are tetradehedra with faces parallel to the dense $\{111\}$ planes and the $\{100\}$ planes.
- (4) Off-Bragg illumination of the precipitates causes a disappearance of Al lattice fringes and produces precipitate images which are in good agreements with the atomic model.

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Captions of figures

Fig. 1. TEM micrographs of Xe precipitates in Al. The specimen was ion-implanted by 30 keV Xe^+ to a dose of 3×10^{20} ions/m² at room temperature and subsequently annealed at 523 K for 0.5 h. (a) a bright field image near specimen edge, (b) a selected area diffraction (SAD) pattern, (c) an enlarged image of square area in Fig. 1a and (d) a corresponding dark field image of Fig. 1 c using a spot A in Fig. 1b.

Fig. 2. High-resolution TEM (HRTEM) of Xe precipitates in Al, observed near Scherzer defocus. The incident electron beam is parallel to the [011] direction. (a) a low magnification micrograph and (b) an enlarged micrograph of square area in Fig. 2a.

Fig. 3. The results of multi-slice image simulation of Xe precipitates in Al matrix at different defocuses Df. The solid Xe is modeled as a tetradecahedron having 8 {111} planes and 4 {100} planes. (a) Df = -55 nm (Scherzer defocus), (b) Df = -100 nm and (c) Df = -165 nm.

Fig. 4. TEM results of Xe precipitates in Al, observed at off-Bragg illumination and showing clear Xe lattice image. The incident electron beam is slightly tilted along the [011] direction. (a) a high-resolution TEM (HRTEM) image and (b) a SAD pattern.

Figure 1.

Figure 2.

Figure 3.

Figure 4.