Corrosion Behavior of Copper-Base Materials in a Gamma-Irradiated Environment

Final Report

W. H. Yunker

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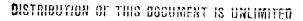


Westinghouse P.O. Box 1970
Hanford Company Richland, Washington 99352

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CORROSION BEHAVIOR OF COPPER-BASE MATERIALS IN A GAMMA-IRRADIATED ENVIRONMENT: FINAL REPORT

Wayne H. Yunker

ABSTRACT

Specimens of three copper-base materials were corrosion tested with gamma radiation exposure dose rates in the range of 1.9 x 10³ k/h to 4.9 x 10⁵ R/h. Materials used were pure copper, 7% aluminum bronze and 30% copper-nickel. Exposures were performed in moist air at 95 °C and 150 °C and liquid Well J-13 water at 95 °C, for periods of up to 16 mo. Specimens were monitored for uniform weight loss, stress-induced corrosion and crevice corrosion. Specimen surfaces were examined visually at 10% magnification as well as by Auger Electron Spectroscopy, x-ray diffraction and metallography. Corrosion was not severe in any of the cases. In general, the pure copper was corroded most uniformly while the copper-nickel was the least reproducibly corroded.

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CORROSION BEHAVIOR OF COPPER-BASE MATERIALS IN A GAMMA-IRRADIATED ENVIRONMENT: FINAL REPORT

1.0 SUMMARY

Over 400 specimens, made from three copper-base materials, were exposed to three corrosion environments in a gamma radiation field. The environments approximated those expected during the early, mid, and late periods of storage of nuclear waste in the tuff formation inside of Yucca Mountain near the Nevada Test Site (NTS). The work was performed in support of the Nevada Nuclear Waste Storage Investigations (NNWSI) Project under the direction of the Lawrence Livermore National Laboratory (LLNL). The three materials used were pure copper, 7 wt% aluminum copper alloy, and 30 wt% copper-nickel alloy. The object of the tests was to observe the corrosion behaviors of these materials under a variety of conditions and specimen configurations and in a radiation field that was at least 10 times stronger than expected under storage conditions in order to enhance any radiation effects. Specimen types used were weight loss, crevice, tear drop, and electrochemically prepitted. Environments consisted of moist air at 150 °C, air that was nearly saturated with water vapor at 95 °C, liquid Well J-13 water at 95 °C, and the liquid-gas interface at 95 °C. A data base was created that specifies all conditions and observations for each specimen. Analyses were also made of the yas and liquid phase compositions. Exposure times ranged from 1 mo to 16 mo and gamma exposure dose rates from 1.9 x 10^3 R/h to 4.9 x 10^5 R/h.

Corrosion damage was relatively mild during these time intervals. No evidence of stress-induced corrosion was found. Underfilm corrosion features were seen with the aluminum bronze and the copper-nickel alloy. The defects in the electrochemically prepitted specimens tended to disappear with general corrosion, rather than grow. In general, the most severe corrosion occurred on specimens in the gas phase at 95 °C that was very nearly saturated with water vapor. Metallographic examination showed some pitting of the aluminum bronze and the copper-nickel alloy. Surface examination of the pure copper specimens

showed isolated pitting only. General corrosion rates varied from 0.01 mil/yr to 0.15 mil/yr at 10,000 h exposure. Both linear and parabolic corrosion rates were seen. X-ray diffraction of the oxide formed on pure copper showed both CuO and Cu20. The compound (Cu0.2 Ni0.8)0 was identified on one nickel alloy specimen. Auger electron spectroscopy of several very thin corrosion films showed oxide with layers having different compositions. There was no excessive corrosion at the gas-liquid interface. However, where isolated drops of water may have formed during part of a test, more severe local corrosion appeared. The corrosion of the copper-nickel alloy was the most severe in appearance and the least reproducible.

2.0 INTRODUCTION

The U.S. Department of Energy (DOE) is evaluating the feasibility of using copper-base materials in the construction of containers for high-level radioactive waste for permanent, geologic disposal. The disposal site being considered is at Yucca Mountain, adjacent to the NTS. For this evaluation, the corrosion behaviors of several candidate materials need to be determined in chemical, physical, and radiological environments similar to those in the proposed repository site. The experiments described here were conducted during fiscal years (FY) 1985 and 1986 at Westinghouse Hanford Company (Westinghouse Hanford) located in Richland, Washington. They are part of the NNWSI Project under the direction of LLNL.

The primary objective of this work is to measure the corrosion rates and behaviors of selected copper-base materials in gamma radiation fields under conditions that are similar to those expected in the proposed repository. These experiments were designed to produce as much quantitative and qualitative corrosion data as possible, in a short time. Resources were focused on exploratory experiments that produced severe corrosion rather than on tests that used large numbers of specimens (for statistical purposes) or on parallel experiments without gamma radiation.

The environmental conditions expected for nuclear waste containers buried in the proposed repository have been described. (1) Briefly, the site is in a layer of tuff rock (a welded, devitrified volcanic ash) permeated by air, water vapor, and some condensed water. The reference horizon at the proposed site is about 200 m above the static water table. Water used for these experiments is from Well J-13 near Yucca Mountain on the NTS. The composition of this water is presented in Table 1. The radiation doses expected initially at the burial container surfaces range from about 5 x 10^3 rd/h for defense high-level waste to about 2 x 10^4 rd/h for commercial spent fuel. For the early years of burial, the temperatures at the surfaces of many of the packages are expected to exceed the boiling point of water (95 °C) at the

TABLE 1
WELL J-13 WATER COMPOSITION*

Impurity	Concentration (mg/l)
Silicon	27.0
Sodium	43.9
Calcium	12.5
Potassium	5.1
Magnesium	1.9
Lithium	0.042
Strontium	0.035
Iron	0.006
Aluminum	0.012
Bicarbonate	125.0
Sulfate	18.7
Chloride	6.9
Nitrate	9.6
Fluoride	2.2

^{*}Reference groundwater composition for tuff repositories (based on composition of Jackass Flats Well J-13 at the Nevada Test Site). The pH is slightly alkaline (7.6).

Source: J. Delaney, "Reaction of Topopah Spring Tuff with J-13 Water: A Geochemical Modeling Approach Using the EQ3/6 Reaction Path Code," UCRL-53631, Lawrence Livermore National Laboratory, Livermore, California, March 1985.

elevation of the reference horizon. The radiation field and temperature will decrease with time. When the temperature of the waste package reaches 95 °C, the radiation dose rates in many of the packages will have decreased to about 2×10^2 rd/h.

The interaction of gamma radiation with aqueous environments produces a host of transient radicals, ions, and stable molecular species. These have been thoroughly described in Reference 2. Radiolysis of air-saturated Well J-13 water is expected to result in an oxidizing environment with O_2 (from air) and H_2O_2 as the dominant species. Much smaller concentrations of O_2^- , HO_2 , OH, H_2 , H_1 , and e_{aq}^- are also generated. In the moist air phase, nitric acid and various oxides of nitrogen are formed and can dissolve in any liquid water phase that is present. These radiation-induced changes in the chemical environment can be expected to alter the normal rates or mechanisms of corrosion.

Early results and a description of this work have been published. (3,4) Most of that information has been repeated here to provide a more complete view of the program results. The few details not repeated are referenced in the text. Some of the preliminary interpretations previously offered have been revised in the light of this more complete data set.

3.0 EXPERIMENTAL

3.1 <u>ENVIRONMENT</u>

For these experiments, gamma radiation fields in the range of 1.9×10^3 to 4.9×10^5 R/h were used. This was supplied by a 60 Co gamma radiation source (Figure 1). The flux calibration values are shown in Figure 2. These values had been obtained using an ion chamber and thermoluminescent dosimeters and are traceable to a National Bureau of Standards (NBS) certified source. No additional calibrations were made for these tests. The corrosion specimens were located in the lowest 30 in. of each tube. The other environmental conditions used were: 1) an air-water vapor mixture at 150 °C; 2) a water-vapor-saturated air phase at 95 °C, and 3) liquid Well J-13 water at 95 °C. The two gas phases had similar dew points. Condition 1 represents early life repository conditions and conditions 2 and 3 represent the conditions after several hundred years.

3.2 CORROSION CELLS

3.2.1 Specimens

The compositions of the three materials selected for study are shown in Table 2. Over 400 weight loss, crevice, and tear drop (stressed) specimens were exposed for periods of from 1 mo to 16 mo. The specimen configurations are shown in Figure 3. The mating surfaces of the crevice specimens were polished with 600-grit abrasive to within ± 0.05 mm (0.002 in.) of flatness. The tear drop specimens were prepared by the single stressing method⁽⁵⁾ and fusion-welded with a tungsten inert gas (TIG) electrode.

3.2.2 Support Cages

All specimens were mounted on high-purity aluminum-oxide spacers in specimen support cages as shown in Figures 4, 5, and 6. The cages were 127 mm (5 in.) high and just fit into the 95-mm (3 3/4-in.) diameter space of a containment

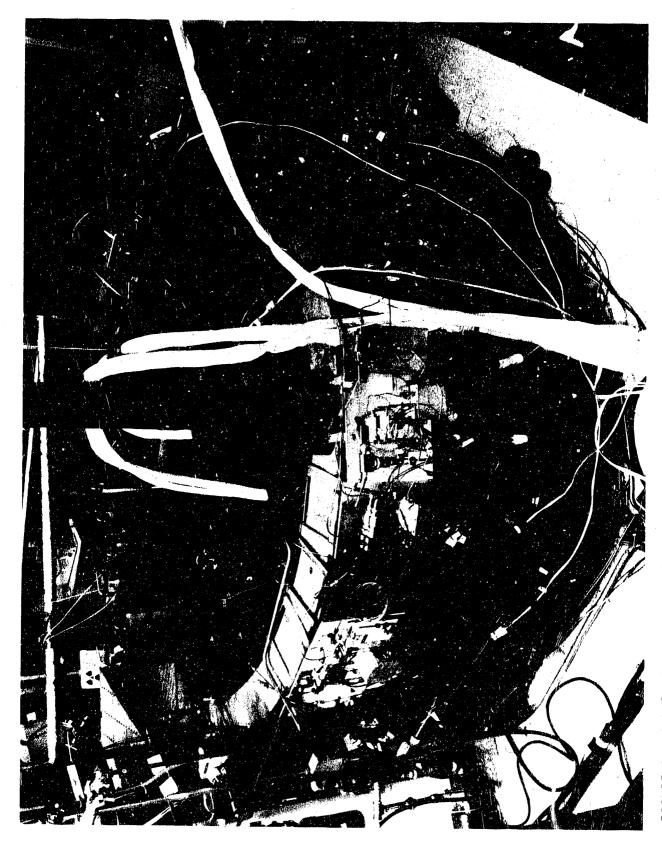


FIGURE 1. Cobalt-60 Gamma Irradiation Facility in the 3730 Building at Westinghouse Hanford Company.

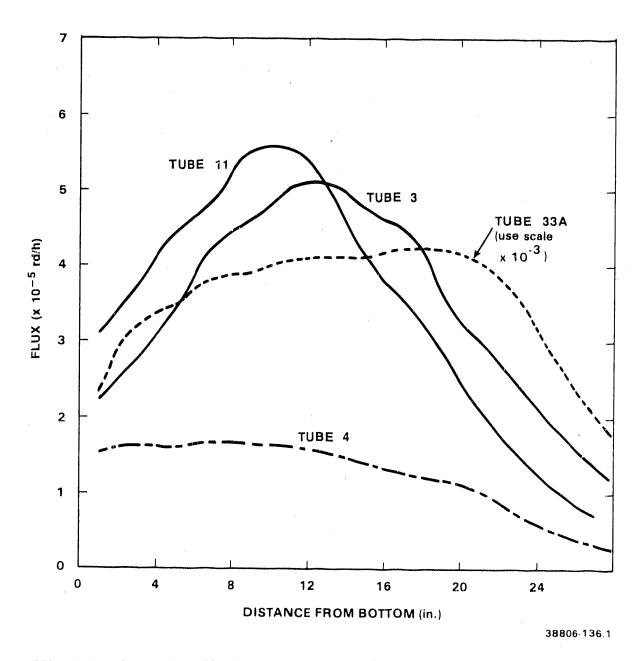


FIGURE 2. Gamma Irradiation Flux versus Distance from the Bottom of the Irradiation Tubes.

TABLE 2 NOMINAL COMPOSITION OF CORROSION SPECIMENS

<u>Element</u>	Allo CDA* 101/102	y Composition (wt%) CDA 613/614	CDA 715
Aluminum			CUA 715
Cadmium	0.001 max	6.75	
Carbon	0.001 max		
Cobalt			0.04
		0.01	
Copper	99.99 min**	90.82	69.18
Iron		2.46	0.53
Lead	0.001 max	0.01	0.01
Manganese		0.16	0.51
Mercury	0.0001 max		••••
Nickel		0.05	29.6
Phosphorus	0.0003 max		0.002
Sulfur	0.0018 max		0.002
Tin		0.2***	0.01
Zinc	0.0001 max	0.01	0.07

^{*}Copper Development Association (CDA) identification number. **Equals 99.95 for CDA 102. ***Only in CDA 613.

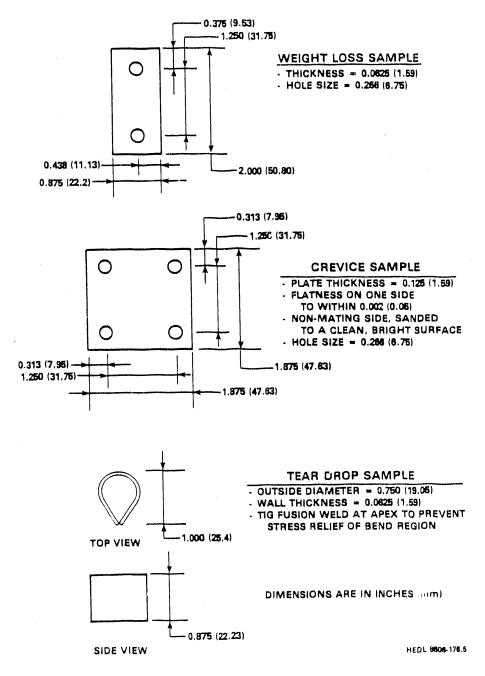


FIGURE 3. Sketches of the Weight Loss, Crevice, and Tear Drop Corrosion Specimens.

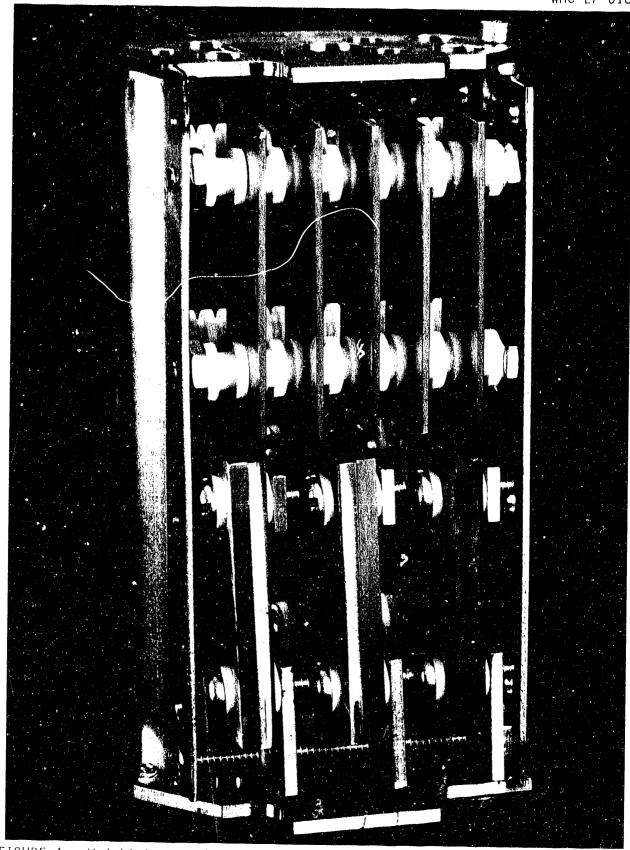


FIGURE 4. Weight Loss and Crevice Specimens Assembled in a Support Cage. (Spacers are made of high purity aluminum oxide.)

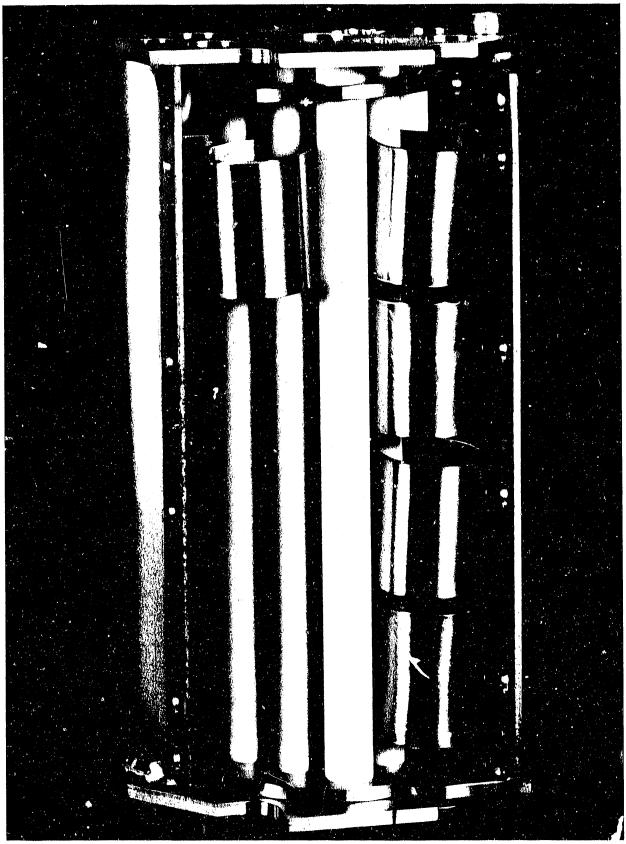


FIGURE 5. Tear Drop Specimens Mounted in a Support Cage. (Spacers are made of high-purity aluminum oxide.)

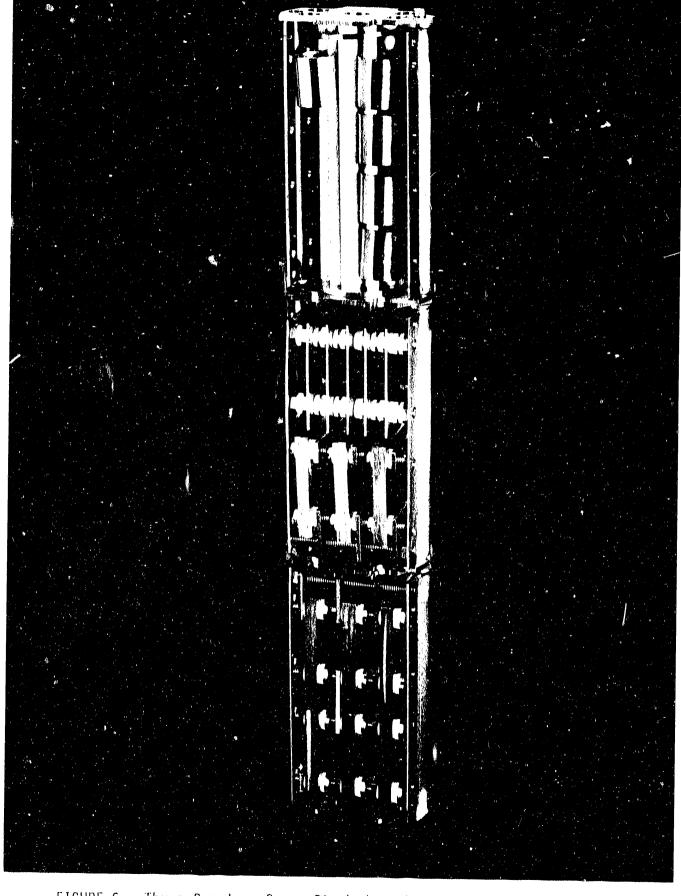


FIGURE 6. Three Specimen Cages Stacked as in a Corrosion Vessel. Neg 8503864-lcn

vessel (see below). All metal parts were made of the same alloy as the enclosure vessel. The six cages stacked into a vessel could be placed in any vertical order to provide flexibility in the exposures given to the specimens. The cages had a standard design and could hold any of the three types of specimens. The cage capacities were: 16 tear drop specimens, 6 crevice specimens, 50 weight loss specimens, or 3 crevice with 28 weight loss specimens.

As the cages were lowered into a vessel, a keying arrangement aligned and held each cage in position in the vessel to ensure that the specimens were not in contact with the thermowell tube and gas ventilation tube. Because the operation of transferring a vessel into and out of the gamma irradiation tubes required considerable handling and movement of the vessel, all specimens were physically restrained in their mountings to prevent any metal-to-metal contact.

3.2.3 <u>Vessels</u>

Environmental control for the corrosion specimens was maintained in the sealed pressure vessels that were lowered into the exposure tubes of the Cobalt-60 Gamma Irradiation Facility. (6) Permanent mounting of the stainless steel exposure tubes ensured that the flux calibrations were reproducible. The stainless steel tubes contained air at about 24 °C.

A schematic of the vessels is shown in Figure 7 and a photograph is shown in Figure 8. Metals used were titanium and Monel* 400. The titanium vessels were fabricated with full-penetration electron beam and TIG welds to eliminate internal crevices. The outer protective containers for the vessels and their electrical heaters were made of 304 stainless steel. The two 3.2 mm (1/8-in.) outside diameter gas lines, made of Inconel** 600, connected the vessels to

^{*}Monel is a trademark of Huntington Alloy Products, Inc., International Nickel Company, Inc.

^{**}Inconel is a trademark of Huntington Alloy Products, Inc., International Nickel Company, Inc.

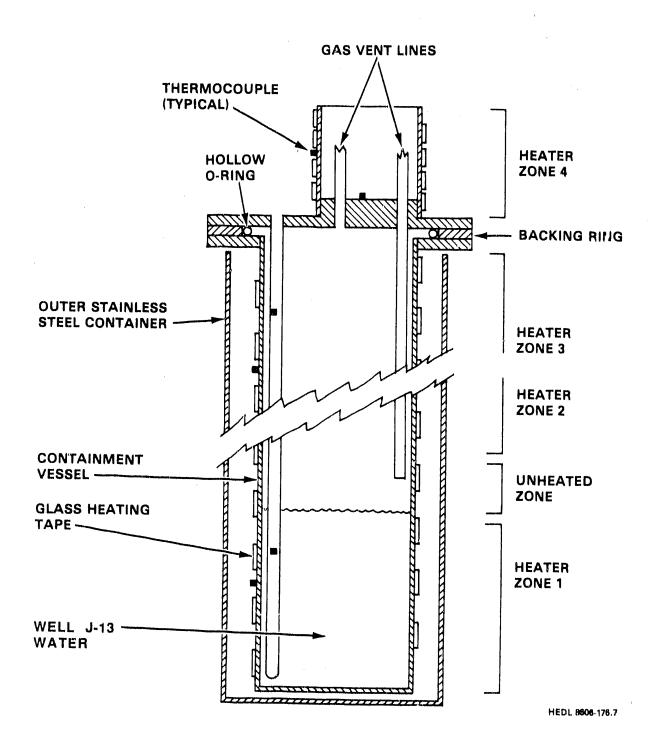


FIGURE 7. Schematic of a Corrosion Vessel.

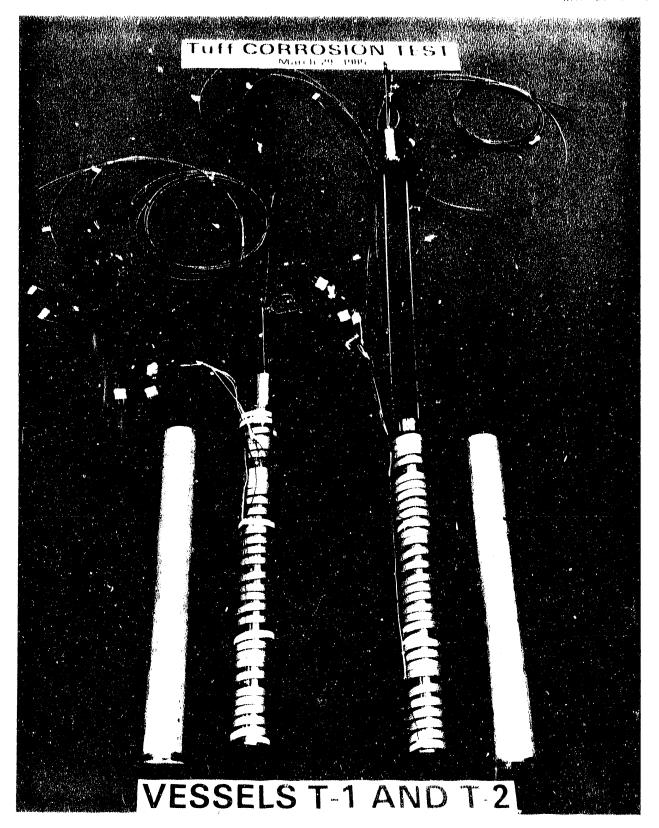


FIGURE 8. Two Corrosion Vessels with the Stainless Steel Guard Vessel Removed. (The 1id of the right-hand vessel is also removed.)

the pneumatic control panels on the operating floor of the laboratory. The vessels were sealed by hollow metal O rings made of 321 stainless steel and plated with a 0.05-mm (2-mil) thick layer of nickel.

To prevent refluxing of water on the underside of the vessel lid and in the gas line connections just above the lid, heat was supplied to the top of the vessel by a glass-insulated heating tape wrapped around a brass cylinder. The 127-mm (5-in.) high cylinder formed a heated enclosure for the gas line connections and conducted heat to the vessel lid. A thermocouple mounted on the upper surface of the lid was used to determine the set point for the heater controller.

For vessels containing a liquid water phase, particular care was needed to prevent condensation of liquid on the surfaces of specimens that were mounted in the gas phase of the vessel. Such condensation can have a marked effect on the observed corrosion behavior. The tall, narrow gas chamber in these vessels normally promotes the formation of a vertical temperature gradient and stagnation of the gas in this space. The design and operation techniques used in the experiment to prevent this were: 1) a 64-mm (2-1/2 in.) gap between the top of the water, the water heater zone, and the lowest specimens in the gas space, the lower end of the gas phase heater, and 2) an independent, two-zone heating of the gas space. By careful adjustment of the three heater controllers, thermal convection was created in the gas space to mix the gas, maintain a more nearly uniform temperature distribution, and maintain a gas temperature slightly greater (3 °C-estimated) than the dew point.

3.3 SUPPORT EQUIPMENT

The experiment support equipment was mounted on three racks as shown in Figure 9.

3.3.1 Gas System

Gas circulation from the pneumatic control panel and through a vessel was ensured since one gas vent was on the vessel lid and the other was located

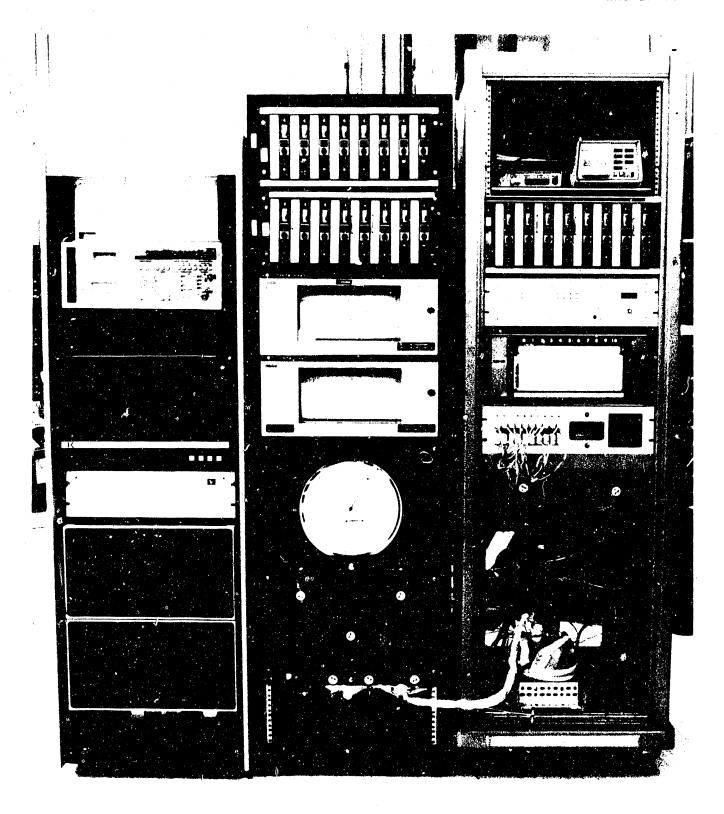


FIGURE 9. Experiment Support Equipment. (The data logger is in the left-hand cabinet. The other cabinets contain the temperature controllers, recorders, and gas manifolds.)

about 25 mm (1 in.) above the water level, or bottom of the vessel. The two 3.2-mm (1/8-in.) diameter gas lines from each vessel were attached to separate gas control manifolds. A schematic of two manifolds is shown in Figure 10. This arrangement allowed for gas sampling, purging of the gas space and calibration of the pressure transducers under operating conditions. Each vessel was protected from overpressure with a Nupro* series CPA adjustable pressure relief valve.

3.3.2 <u>Heaters and Controllers</u>

All six heaters on each vessel installation were individually controlled by Research, Inc., Model 639B three-function controllers**. Control thermocouples were banded to the outside of the vessel, or gas lines, between wraps of the heating tape. The controller set points were adjusted to produce the desired temperature values on the immersion thermocouples inside of the vessel. The immersion thermocouples were spring loaded against the inside walls of their immersion tube to assure minimum temperature error. The calibrations of the immersion thermocouples were verified before an experiment by operation of the vessel when it was filled with water at the normal boiling point. The 3.2-mm (1/8-in.) tubes, 7.6 to 12.2 m (25 to 40 ft) in length, were trace heated with metal-sheathed resistance heating cables. All parts of a gas manifold that were directly connected to the vessel were trace heated with glass-covered resistance heating tape. The manifolds and gas lines were maintained at about 115 °C to prevent moisture condensation. The extra thermocouples installed on the equipment to check the trace heating were monitored on a multistation thermocouple readout panel.

^{*}Nupro series CPA adjustable pressure relief valve is a trademark of the Nupro Company, Willoughby, OH.

^{**}Model 639B three-function controller is a trademark of Research, Inc.

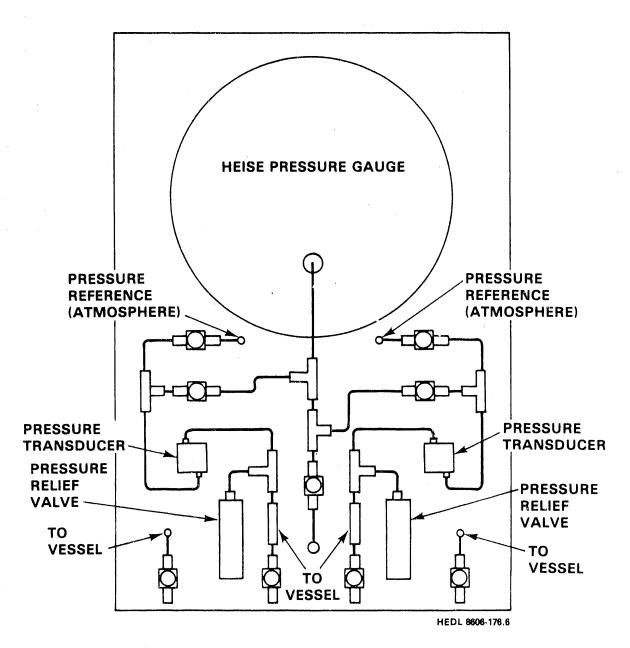


FIGURE 10. Schematic of Two Gas Manifold Systems.

3.3.3 <u>Data Systems</u>

System condition values were measured hourly by a Fluke Model 2240A data logger^a and recorded on nine-track magnetic tape by a Kennedy Model 1600 recorder^b. Periodically, the data tapes were read by a Kennedy Model 9832 tape deck^c and the data transmitted to a Sperry^d 1100/40 computer for processing. Details of the data channels and output records are given in Reference 3. To provide backup records for the vessel temperatures and to detect temperature transients that might occur between recordings by the data logger, the midlevel immersion thermocouple voltage from each vessel was recorded on an Esterline Model L1102S strip chart recorder^e. The pressure in each vessel was measured with a Validyne Model CD280 Multichannel Carrier Demodulator^f with Model DP-7 and DP-15 variable reluctance, differential pressure transducers that used the air atmosphere as a reference. Calibration of these units was made under operating conditions with a calibrated, 690 kPa $(100-1b/in^2)$ Heise⁹ gauge, readable to less than 0.69 kPa $(0.1 lb/in^2)$. Finally, a telephone callout system was installed that would automatically call one of several key personnel if any abnormal value of temperature or pressure occurred in any vessel or if electrical power was interrupted.

^aFluke Model 2240A data logger is a trademark of John Fluke Manufacturing Company, Inc.

^bKennedy Model 1600 recorder is a trademark of an Allegheny International Corporation

^CKennedy Model 9832 tape deck is a trademark of an Allegheny International Corporation

dSperry is a trademark of UNISYS

^eEsterline Model [11025 strip chart recorder is a trademark of Scientific Columbus, Inc.

fValidyne Model CD280 Multichannel Carrier Demodulator is a trademark of Validyne Engineering Corporation

⁹Heise gauge is a trademark of Dresser Industries, Newtown, CT.

3.4 PROCEDURES AND TECHNIQUES

3.4.1 Specimens

3.4.1.1 General Treatments

Each new specimen was engraved with a unique serial number that identifies origin and composition throughout all phases of exposure and examination. A record was kept of the specific location of each specimen in each cage, vessel, and experiment. The serial number code is XYZZZ where:

X = Alloy

 $1 = CDA \ 101/102 - pure copper$

2 = CDA 613/614 - aluminum bronze

3 = CDA 715 - 30% nickel-copper

Y = Specimen type

W = Weight loss

B = Bend (tear drop)

C = Crevice

ZZZ = Serial number

Specimens were prepared for use by ultrasonic cleaning in xylene, then hand polished with 600-grit silicon carbide paper and rinsed several times in absolute ethyl alcohol; one rinse was in the ultrasonic cleaner. All specimen cages, hardware, and vessels were cleaned in a similar manner. The ceramic hardware was etched in 30% nitric acid at approximately 80 °C, rinsed in demineralized water several times, and baked in a vacuum oven overnight at approximately 200 °C. When specimens were loaded into their respective cages, the spatial arrangement of each specimen type assured that adjacent specimens were of different alloys.

Specimen cages removed from an experiment were isolated in closed containers and taken to the laboratory for disassembly and specimen replacement. New specimens were mounted with clean ceramic spacers. To minimize bias, speci-

mens of each type and alloy were examined in as large a group as possible and without regard for exposure history.

Oxide films were removed according to ASTM G1-81:(7) 1 to 3 min in a deaerated, hydrochloric acid solution (1:1) at room temperature. Separate solutions were used for each alloy. After the films were removed, the specimens were rinsed in demineralized water, rubbed hard by hand with a lint-free cotton cloth, rinsed again in absolute ethyl alcohol, and air dried.

3.4.1.2 <u>Crevice Specimens</u>

Crevice specimen pairs were not washed or rinsed after removal from the vessel. The pairs were kept tightly bolted together until ready for examination. After the pairs were opened, a careful, visual examination was made of all inner surfaces, using a 10X magnifier. A color photograph was made of one inside and one outside surface of each specimen. Specimens were cut by a hand-operated hacksaw for analysis and examination.

3.4.1.3 <u>Tear Drop Specimens</u>

After the oxide films were removed, each specimen was given a careful, visual examination with the aid of a 10X magnifier. Color photographs were made (up to 10X magnification) of any features of particular interest. The specimens were examined for microcracks by fluorescent dye penetrant per ASME code. (8) The specimens were then cut with a hand-operated hacksaw for analysis and examination.

3.4.1.4 Weight-Loss Specimens

Specimens were lightly rinsed with demineralized water and dried overnight in air. Weighings were made before the corrosion exposure (after final cleaning and polishing), before removal of the oxide film (after rinsing and drying), after the oxide film was removed, and after a second acid treatment of equal length to the first treatment (to correct for the metal loss by the acid). A color photograph was made of one side of each specimen before oxide removal.

After the oxide film was removed, a close visual inspection was made of all surfaces, using a 10% magnifier. Color photographs were taken following the same procedure used on the tear drop specimens.

In each 95 °C experiment, one set of weight-loss specimens was mounted so that half of the metal was in Well J-13 water and the other half was in the gas phase. Twenty-seven specimens were exposed at approximately 4 x 10^5 R/h for 1, 3, and 9 mo. Nine others were exposed at approximately 4 x 10^3 R/h for 3 1/2 mo. After their exposures to the corrosion environments, the specimens were cut in half lengthwise with a hand-powered hacksaw. The oxide film was removed from one piece and the metal surface examined visually using a 10X magnifier. Some of the pieces were sectioned and polished for metallographic examination.

3.4.1.5 <u>Electrochemically Prepitted Specimens</u>

A set of weight loss and tear drop specimens of each of the three materials was electrochemically prepitted at LLNL and then placed in the gamma corrosion experiments at Westinghouse Hanford Company. The prepitting was performed in a synthetic Well J-13 type water solution prepared as shown in Table 3. The pitting was accomplished by raising the electrochemical potential of a specimen by steps to several hundred millivolts above the measured corrosion potential. The ranges of process times and voltages used are summarized in Table 4. These specimens were not included in the Westinghouse Hanford marking system described above or included in the data base in the Appendix. The variations in the pitting conditions used produced a variety of pitting structures and was consistent with the program objectives, i.e., to determine the corrosion effects on a variety of specimens. At Westinghouse Hanford, the specimens were mounted in the all-titanium vessel T-3 and exposed to conditions similar to those that existed in vessel T-1 (i.e. 95 °C, liquid and gas phases). The oxide film removal process mentioned above was modified for these specimens. The hydrochloric acid etch was a 30-s exposure in a highenergy ultrasonic bath. This was followed by a 30-s rinse in the ultrasonic bath, followed by the normal rinse sequence. After the oxide films were

TABLE 3

COMPOSITION OF SYNTHETIC WELL J-13 TYPE WATER (Used only for electrochemical prepitting)

<u>Chemical</u>	Quantity*
Sodium bicarbonate	17.0
Potassium nitrate	1.5
Sodium chloride	1.2
Potassium sulfate	3.36
Sodium fluoride	0.508
pH range:	8.17 to 8.64

^{*} grams/liter of solution (molal)

removed and visual observations were made, the tear drop specimens that had received the 6-mo exposures were sectioned, polished, and given metallographic examinations.

3.4.1.6 <u>Metallographic Examinations</u>

Metallographic sectioning and examinations were made to determine whether the surface corrosion features were related to any breakdown of the structure of the underlying metal. The metallography also complements the other types of examinations performed. This sampling provided a limited view only of possible metal structures and should not be considered a comprehensive examination of the specimens.

The 37 specimens selected for examination included examples from each subgroup of alloy and environmental condition. Specimens were mounted, sectioned, and polished using standard metallographic techniques. After the specimens were carefully examined, unusual features were photographed (magnification range 5X to 400X). The specimen surfaces were then etched with a mixture of about

TABLE 4

RANGES OF MEASURED CORROSION AND PITTING POTENTIALS USED BY LAWRENCE LIVERMORE NATIONAL LABORATORY FOR PREPARATION OF PREPITTED SPECIMENS

	Time At Final Value (min)	10 - 25	8 - 18	5 - 20
nence	Final Potential	+400	+1,200	+1,100
Pitting Potential Sequence (mV vs. SCE)	Size of Step Increase	50 mV/5 min to 100 mV/5 min	50 mV/5 min to 150 mV/5 min	250 mV/5 to 18 min
	Starting Potential	-100	+500	+750
Corrosion Potentials (mV vs. SCE)	Tear Drop Specimens	-77 to -100	-119 to -147	-125 to -168
Corrosion (mV vs	Weight Loss Specimens	-63 to -76	-135 to -172	-129 to -150 -125 to
	Material (CDA No.)	102	613	715

19 vol% concentrated nitric acid and concentrated formic acid to reveal the grain structures. After the specimens were reexamined, additional photographs were made, as appropriate.

3.4.1.7 <u>Visual Examinations</u>

All specimen surfaces were carefully examined under good lighting, using a 10X magnifier. Descriptions of specimen conditions and special features were recorded. A series of two-letter codes or descriptors was devised to condense this information into a usable form so that comparisons, groupings, and correlations could be made using the data base. A detailed description of the list is in Appendix A along with the data base. The descriptors are somewhat suggestive of their meaning, so that they can be strung together to make descriptive phrases. For example, "FWDKSPPT" reads "a few dark spots and pits."

While every effort was made to make the descriptors as explicit as possible, the system is qualitative, and there are overlaps in the definitions. However, the value of being able to easily analyze and compare specimen conditions with other experimental values in the data base was felt to outweigh the brief inconvenience of learning the codes. In particular, the notation of "pits" was used in cases where it was not clear whether the features were pits (with recognizable depth) or spots. Thus this entry is on the conservative side.

3.4.1.8 Air Water-Vapor at 150 °C

To establish the vessel T-2 atmospheric composition at the beginning of an experiment, the vessel was placed into the gamma field and all systems were brought to the operating temperatures. Compressed breathing air was humidified by sparging at 0.5 L/min through water at 75 °C to 80 °C, and then through a heated delivery system to the vessel. After three to five vessel volumes of gas had passed through the system, the vessel exit valve was closed, the system was pressurized to 1 to 2 $1b^f/in^2$ (gage), and the gas system was sealed. The small pressurization was used to detect leaks that

might develop in the system during the experiment. The gas was not changed during the 1-mo and 3-mo experiments. During the 6-mo exposure period, the vessel was repurged every 7 to 10 d (as above) with one vessel volume of humidified air.

3.4.1.9 Water-Saturated Air and Water at 95 °C

The specimen cages to be located in water were loaded into vessel T-1, T-3, or T-4. Well J-13 water was added to a predetermined level that would be across the middle of the interface specimens during the experiment. The remainder of the cages were loaded and the vessel was then sealed and lowered into the gamma tube. All heaters, except the water heater, were then energized and brought to their operating temperatures.

After the metal specimens had a chance to reach the system temperature (usually overnight), the water temperature was carefully increased without allowing it to reach the temperature of the gas space. Temperature values used for this were based on an estimated overall measurement uncertainty of £1 C°. When the vessel was depressurized or gas samples were taken during an experiment, the water temperature was first cooled to about 30 °C to prevent a violent expansion of the water (from the pressure release) that would splash onto the specimens mounted in the gas phase.

As soon as a vessel was opened for specimen exchange, the gas phase cages were quickly removed and a pH meter electrode was lowered into the vessel. Several readings of the pH were made with reference to standard buffer solutions. Several samples of the water were then removed from the vessel pool with pipettes. Finally, the cages under water were removed, and the remainder of the water was poured into storage containers. The vessel was then swabbed several times with clean, lint-free cotton cloths and rinsed with deionized water several times, drained, and air-dried. Fresh Well J-13 water was used when the vessel was reloaded.

The seal surfaces for the O ring were hand polished with 600-grit silicon carbide paper before each experiment. The seal of a loaded vessel was given a

final leak check by pressurizing the vessel to 340 to 550 kPa $[50-80\ lb^f/in^2]$ (gage)], depending on the setting of the pressure relief valve, and by covering all the seal flange areas with a commercial bubble-forming liquid.

3.5 <u>QUALITY ASSURANCE</u>

The level of quality assurance for this work is level three as defined in Reference 9. The basic elements of this level include traceability of specimens from the foundry heat number to the experimental conditions and the final analyses of each specimen, calibrations of all primary measurement instruments to the NBS references, written laboratory procedures, and complete, indexed, laboratory records.

4.0 <u>RESULTS</u>

4.1 DATA BASE

All information regarding the specimens and their examinations was entered into a Symphony data base on an IBM Personal Computer (PC), for ease of data verification and analysis. The data base listing is shown in Appendix A along with an explanation of the codes and formats used. Elsewhere in this report, a shorthand notation is used for the exposure conditions for specimens:

MMM/P/N

where:

MMM = Nominal temperature in °C

P = Phase (G = gas; L = liquid)

N = Number of mo of exposure.

Table 5 summarizes the numbers of each type and alloy of the specimens that were processed.

4.2 <u>VESSEL ENVIRONMENT</u>

Since gamma irradiation alters the chemical composition of an air-water environment, measurements of the changes that occurred under these experimental conditions are essential for the characterization of the observed corrosion and the development of an understanding of the corrosion processes. In situ measurements of many of the expected species (2) were beyond the scope of these experiments. Values recorded were 1) vessel pressure versus time, measured hourly, 2) water and gas temperature, measured hourly, 3) gas composition, measured every few weeks, when excess vessel pressure was released [operating maximum was 340 kPa $(50\ 1b^f/in^2\ (gage)]$, and 4) liquid water composition, measured each time an experiment was interrupted for the exchange of corrosion specimens.

TABLE 5
NUMBERS OF EXPOSED SPECIMENS

Alloy

Specimen Type

	Weight Loss	Inter- face	Tear <u>Drop</u>	Crevice	Prepit Weight Loss	tted Tear Drop	<u>Totals</u>
Copper	56	6	41	30	6	4	143
Al-Bronze	56	6	39	30	6	4	141
N1-Cu	<u>56</u>	_6	40	<u>30</u>	<u>6</u>	_4	142
TOTALS	168	18	120	90	18	12	426

 $^{^{1}}$ Symphony, V1.1 is a trademark of Lotus Development Corporation.

4.2.1 Air-Water Vapor at 150 °C

There was no noticeable change in the gas pressure in vessel T-2 during the experiment beyond the daily fluctuations in the barometric (reference) pressure. To make reliable measurements of changes in the amount of gas versus time would have required more extensive instrumentation and thermostating of the system. Mass spectrometric analyses of the gas samples are shown in Table 6. After the first 1-mo and 3-mo experiments were completed, the gas was replenished every 7 to 10 d. (Note that water vapor was not determined by the sampling and analysis methods used.)

4.2.2 <u>Air-Water Vapor At 95 °C</u>

Examples of the rates and amount of pressure buildup in vessel T-1 are shown in Figure 11 for the titanium vessel used in the second part of a 7-mo experiment. The composition of the gas in the vessel was measured with a mass spectrometer. Samples were drawn into evacuated stainless steel bulbs after the connecting lines and manifold were purged with gas from the vessel. As mentioned above, the water vapor content cannot be reliably measured by this method. However, liquid water was always present in the vessel, hence its

TABLE 6

GAS COMPOSITION IN VESSEL T-2 AT 150°C

	Gas	Composition (vol	<u>%)</u>	
Gas	Air	After 3-Month Experiment	After Last 4.5 Mo* of 6-Mo Experiment	15 Days After Previous Purge/ Replenishment <u>of Vessel</u>
Argon	0.95	0.79	0.04	0.08
Carbon dioxide	0.05	0.53	0.06	0.07
Carbon monoxide	<0.1	<0.1	<0.1	<0.1
Hydrogen	<0.01	<0.01	<0.01	<0.01
Methane	<0.01	<0.01	<0.01	<0.01
Nitrogen	78.0	80.1	80.2	80.1
Oxygen	21.0	18.6	19.7	19.7
Water vapor (not measured)		·		

*Eight days after previous gas replenishment NOTE: Analytical sensitivity for N₂O, NO₂, and NO was 0.02 vol% Ammonia (NH₃) was not present with water vapor.

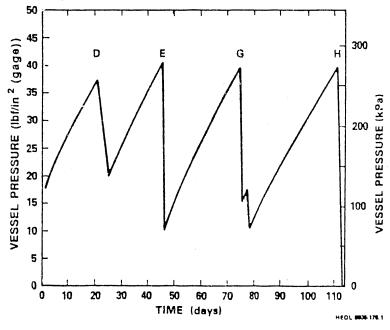


FIGURE 11. Gas Pressure Variation in Vessel T-1 (95°C) During the Last 3-3/4 Mo of the 7-Mo Experiment. The average rate of pressure increase is $(1.0\ lb^f/in^2/d)$ 6.9 kPa/d.

partial pressure was calculated from its temperature. The partial pressure of each gas in the vessel was calculated from the gas pressure and water temperature at the time of sampling and the measured composition of the sample. Results are shown in Table 7.

4.2.3 <u>Liquid Water at 95 °C</u>

Table 8 shows the chemical composition of the water solutions relative to the Well J-13. Note that the 30-d and 95-d values were from a Monel 400 vessel while the others were from titanium vessels. Cation analyses were made by an inductively coupled plasma (ICP) emission spectrometer and the anion analyses by anion chromatography. Samples were filtered before being analyzed. The analyses of some precipitates found in the early experiments have been discussed. (3)

4.3 <u>METAL SPECIMENS</u>

4.3.1 <u>Stress-Assisted Cracking</u>

The dye penetrant examination performed on the 121 corroded tear drop specimens and on three archive specimens revealed no evidence of surface cracking. Twenty of these specimens, regular and prepitted, were then cross sectioned, polished, and metallographically examined. No evidence of subsurface corrosion or cracking was seen. Details of these examinations are discussed below.

4.3.2 <u>Crevice Corrosion</u>

Ten specimens of each alloy were exposed to each of the three corrosion environments. The results of the visual observations of the local corrosion features seen on the crevice specimens are listed in the data base (Appendix A). These are also summarized in Tables 9 through 11. The width of the crevice corrosion penetration zone around the outer edge of the specimens is designated by the code "nM", where n is the corrosion zone width in milli

TABLE 7 GAS COMPOSITION IN VESSELS DURING CORROSION EXPERIMENTS

17	Total . Press	4 1 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2	0.1
16 nts	CH4 (xE3)	(0.33 (0.26 (0.27 (0.13 (0.13 (0.13 (0.11 (0.11 (0.09	(0.1 (0.1
15 Experiment	He (xE3)	 <0.33 <0.26 <0.27 <0.13 <0.02 <0.13 <0.03 <0.01 <0.01 <0.01 	(0.1) (0.1) (0.1)
14 'ing	CO (xE3)	(3.33 (2.56 (2.75 (2.75 (1.33 (2.84 (2.56 (1.24 (3.37 (1.07 (1.00 (1.00 (1.00 (1.00 (1.00 (1.00 (1.00 (1.00	(1.0
	osphere sures Ar (xE3)	1.33 0.77 2.75 0.44 0.13 (0.28 1.02 0.12 6.06 0.64 3.48	0.20 9.50 9.34
0 11 12 13 Composition in Vessels	(absolute atmospheres) Partial Pressures H2 CO2 Ar	8.00 0.51 19.27 8.65 2.27 2.27 6.14 2.86 17.85 1.71 19.06	0.20
11 positio	(absolu Parti H2	1.65 1.67 1.16 0.88 0.55 1.21 1.21 1.51 1.51 0.00 0.00	0.00
10 Gas Comp	N 2	0.84 0.26 0.26 0.71 0.37 0.63 0.11 0.93 0.15 0.15	0.80 0.78 0.78
6	02	0.84 0.62 0.76 0.37 0.71 0.71 0.90 0.090 0.090	0.21 0.21 0.21 0.21
ω	н20	0.90 0.91 0.82 0.83 0.83 0.80 0.79 0.79 0.78	
7	Vessel Net Press (psia)	449.0 37.6 4C.5 32.6 119.6 41.7 37.6 118.3 15.7	
9	Vessel Total Press (psig)	3.47.3 3.0.3 3.0.3 3.0.3 3.0.3 3.4.2 7.4.1 7.6.5 6.5 6.5 7.6 8.6 8.6 8.6 8.6 8.6 8.6 8.6 8.6 8.6 8	
Ŋ	Vessel Water Press (psia)	2.5.2.2.2.2.2.2.2.2.2.2.2.2.2.2.2.2.2.2	
đ.	Vessel Water Temp (C)	97.0 94.5 94.9 94.9 93.7 93.2	
м	1136	10:00 08:00 10:00 08:00 09:00 08:35 08:40 13:40 10:23	09:15
2	Date	09/05/85 111/12/85 01/08/86 01/31/86 03/04/86 07/15/86 10/10/86 10/14/86 10/14/86 10/14/86	10/09/86 01/10/86 03/05/86
	Sample Number	11-C 11-C 11-E 11-B 11-A 11-A 11-A 11-A 11-A 11-A	12-D CY-1 Poom Book*

Explanation of table, by column number:

Temperature of water in vessel, under experimental conditions, before the gas sample was taken. Water vapor pressure in the vessel, from temperature in col 4, calculated with the equation; From the data records. 1-3,6

Intal pressure of permanent gases in the vessel, under experimental conditions; col 6 - col 5 + 14.7. Partial pressures of the constituent gases, under experimental conditions. Just prior to sampling; log P(torr) = 8.689743 - 2167.71/T(K).

Total of partial pressures under experiment conditions; sum of col B thru 16. Col 8 is from col 5. 1.

*Handbook of Chemistry and Physics, 57th ed., CRC Press, Cleveland, Onio 1976.

TABLE 8

WATER COMPOSITIONS AT 95 °C

9+95e	Titanium 6.4	1.80 6.8 0.62 13.1 3.0	23.7 16.2 <0.08 0.033	4.5 0.76 0.015 6.1 0.039	0.51 0.023 44.
36+63e	Titanium 6.5	0.68 2.4 17.5 5.4 2.8	6.6 6.1 <0.08 0.009	4.6 0.59 0.045 3.0 0.036	0.38 0.007 18.
25+128 ^e	Titanium 8.0		11.4 11.0 <0.08 0.042	6.8 0.16 0.914 2.1 0.029	0.35 0.007 18.
115.d	Titanium 7.6	0.78 3.6 18.8 4.4	9.7 9.3 0.18	3.9 0.03 0.04 4.8 0.13	0.45 <0.005 2.8
95.	Monel ^C 400 9.0	0.38 2.5 2.5 1.8 1.8	<1.0 8.4 8.1 <0.15 0.04	2.7 <0.01 <0.02 2.4 <0.03	0.06 <0.005 17.
30.	Monel ^C 400 8.0	1.4 5.8 10. 6.2 3.8	<1.0 21. 21. <0.15 0.03	9.8 0.09 <0.02 3.3 0.06	1.1 <0.005 36.
0. J-13 ^b (HEDL)	- 8.9	2.1 7.0 8.7 19.	23. 23. <0.15	12. <0.01 <0.02 6.4 0.07	1.8 <0.005 41.
0. J-13a (LLNL)	<u>al</u> 7.6	2.2 6.9 9.6 18.7	125.f 0.012	12.5 0.006 5.11 0.042	1.92 43.9
Exposure (Days)	Container Material <u>pH</u> Snoring ()	F (WDDIII) F (C1) NO3 SO4 NO2 (est)	Total C Inorganic C Al Ba	Z L K	Mn Na

(CONTINUED) TABLE 8

, +6	0.34 <0.10 2.1 <0.15 0.017	2.0
36+63 ^e	0.25 <0.10 3.6 <0.15 0.015	1.1
25+128 ^e	0.036 <0.10 0.92 <0.15 0.021	1.5
115. ^d	0.23 <0.10 1.45 <0.15 0.008	1.3
95.	0.06 <0.10 0.49 <0.15	0.88
30.	2.1 <0.10 0.54 0.15	2.4
	<0.03 <0.10 30.0 0.15 0.04	2.8
0.	27.0	wt./L) 2.9 2.8
Exposure (Days)	Species (wppm) Ni Pb Si Sn Sr	Anion/Cation Balance (milliequivalent wt./L) Total anions 2.9

alawrence Livermore National Laboratory
bwestinghouse Hanford Company
CMonel is a trademark of Huntington Alloy Products, Inc., International Nickel Company, Inc.
dEnd of exposure for 7-mo experiment
eVessel opened to air once for specimen changes, but water not sampled

Chromatographic column used did not provide adequate results for standard method. Used one Estimated accuracy of results is ±15%. point calibration only. NH[‡] was not determined. NOTE:

fas bicarbonate

TABLE 9

CREVICE SPECIMEN CORROSION: CDA 101/102

Visual* Observations	ARLCIM ARLCIM ARLCIM ARLCIM	LCIM ARSMPTLCIM ARLCIM ARVSSPPT?/ARLCIM? ARLCIM ARLCIM ARLCIM ARLCIM
Gamma Flux (R/hr x E-5)	2.5 2.5 0.67 2.5	4.1 4.1 4.1 0.037 0.037 4.9
Exposure Time (mo)	3.2 5.1 7.4 8.9	0.0.5.2.2.6.0.0.0.0.0.0.0.0.0.0.0.0.0.0.0.0.0
Temp- erature (°C)	95 95 95	95 95 95 95 95 95
Reaction Phase		
Alloy Type (CDA No.)	102 101 101 101	152 102 102 101 101 101 101
Specimen Number	1 C 24 1 C 39 1 C 33 1 C 35	1 C 12 1 C 26 1 C 27 1 C 27 1 C 43 1 C 40 1 C 31 1 C 35

*Visual observation codes are defined in Appendix A and discussed in the text.

TABLE 10

CREVICE SPECIMEN CORROSION: CDA 613/614

Visual* Observations	ISLC/NF FWPTFL3M ISLC	LCFL ARLC GELCFLPT GELCPT GESMPT?SPLC GELCFL GELCFL GELCPTFL GELCPTFL	ARLCIM/ISFL?/CUFM ISLCIM ARLCIM?/CUFM ARLCIM?/NF/CUFM ARLCIM/CUFM ARPTLCIM/CUFM LCIM?/CUFM
Gamma Flux (R/h x E-5)	1.6 0.31 1.5	2.5 2.5 0.033 0.033 2.5 0.67 2.5 2.5	4.1 4.1 4.1 0.037 0.037 4.1
Exposure Time (mo)	1.5 2.1 5.6	3.2 3.5 3.5 5.1 7.4 8.0 8.9	1.0 3.2 3.2 3.5 5.1 8.9
Temp- erature (°C)	150 150 150	0 0 0 0 0 0 0 0 0 0 0	00000000000000000000000000000000000000
Reaction	១១១		
Alloy Type (CDA No.)	614 614 614	614 613 614 614 614 614 614	614 614 613 613 614 614 614
Specimen Number		2 C 6 2 C 23 2 C 35 2 C 36 2 C 26 2 C 24 2 C 28 2 C 28 2 C 28	00000000

*Visual observation codes are defined in Appendix A and discussed in the text.

TABLE 11

CREVICE SPECIMEN CORROSION: CDA 715

Visual* Observations	ISLC/CLNF GEVSPTIM/CLNF	FWLGLC/GEDKFM ISLC/CLNF FWXLLCIM/GEDKFM FWFL5M ARLCFL ARPTLC/ARDKFM CL/DKFMISLCIM ARLCFL5M/GEDKFM	GEVSPTIM/CLNF GESMPTLCZM GESMMDLTSP/ARVSPTIM GEVSPTIM ARVSPTIM GEVSPTIM GEVSPTIM GEVSPTIM/CLNF
Gamma Flux (R/h_x E-5)	4.6 0.75	2.5 0.033 0.033 2.5 0.67 0.67 2.5	4.1 4.1 4.1 4.1 0.037 4.1 4.9
Exposure Time (mo)	1.2	3.5 3.5 3.5 4.7 8.0 8.9	1.0 3.2 3.2 3.3 5.1 8.9 9.9
Temp- erature (°C)	150 150	95 95 95 95 95	95 95 95 95 95
Reaction Phase	വര		
Alloy Type (CDA No.)	715 715	715 715 715 715 715 715 715	715 715 715 715 715 715 715
Specimen Number	3 C 10 3 C 4	3 C 15 3 C 28 3 C 37 3 C 27 3 C 26 3 C 26 3 C 26	3 C 33 3 C 23 3 C 23 3 C 23 3 C 35 3 C 35 3 C 25

*Visual observation codes are defined in Appendix A and discussed in the text.

meters. Some specimens also showed corrosion of the closely mated surfaces at locations inward from the edges of the specimens.

Crevice corrosion was seen only around the outer edges on the CDA 101/102 specimens and only on those exposed at 95 °C. Thirteen of the thirty specimens were thus affected.

Only 1 of the 12 corroded gas phase specimens of CDA 613/614 showed crevice corrosion around the outside edges; it had been exposed at 150 °C. Eight of the liquid phase specimens showed similar attack. The local corrosion found on the other gas phase specimens was in the midsurface regions of the mated surfaces, away from the edges. Included in these features were the filiform-like corrosion patterns that are also seen under the oxide films on the weight loss and tear drop specimens.

Ten of the nickel alloy (CDA 715) specimens showed local corrosion. Half of the 95/G specimens had local corrosion in the inner areas away from the edges of the paired plates. All of the local corrosion of the liquid phase specimens was in the crevice near their outer edges of the specimens.

In summary, crevice corrosion at the edge of the specimens occurred only on the CDA 101/102 specimens exposed in the gas phase, not on those from the liquid phase. The results from the aluminum bronze CDA 613/614 specimens were just the reverse. The CDA 715 nickel alloy showed crevice corrosion on both liquid and gas phase specimens but more regularly on those from the liquid phase. In the regions between the plates, but well away from the specimen edges, no local corrosion features were seen on pure copper. Some features were seen on the nickel alloy specimens, and local corrosion features were common in this region on the aluminum bronze specimens.

4.3.3 Prepitted Specimens

Three sets of prepitted specimens were visually examined:

- o Those in an as-received condition from LLNL.
- o Those that received 3-mo exposure
- o Those that received a 6-mo exposure.

In general, the pits found were rather broad, having a ratio of pit depth to pit width value (aspect) of one or less. The distribution, density, and size of the pits varied considerably across the surfaces of the specimens; this likely is due to the variations in electrochemical conditions used in their preparation.

The exposure of the specimens to the irradiation corrosion environment in the Westinghouse Hanford Company gamma pit seemed to result generally in larger (broader) pits and fewer small pits. This suggests that the pitting initiation process did not continue under the irradiation corrosion environment. Under these conditions, the smaller pits would disappear as the metal near the pits was consumed. A dense population of larger pits would combine, but would eventually disappear by the general corrosion process. The metallographic examinations described above showed no evidence of intergranular penetration or corrosion and there was no evidence of stress-assisted cracking. The only cracks that were seen were related to mechanical deformations of the metal surface that apparently occurred before the corrosion experiments (i.e., in the manufacture of the specimens). There were differences in the behaviors of the different alloys, however.

On the CDA 101/102 specimens the pits were quite broad. Some of the larger pits (about 20 μm - 300 μm deep) showed roughness on their surfaces that was about 1/10 the scale of the pit size. This small-scale roughness may have been smoothed somewhat by the corrosion, but not substantially. The unpitted

surfaces were relatively smooth. Examples of these specimens are shown in Figures 12 and 13 (87520C and 87329C).

Some of the pits in the aluminum bronze (CDA 613/614) surfaces had a higher than average aspect, e.g., 3-5 (depth/width) and up to about 25 μm in depth. In general, the unpitted surfaces did not have the roughness found on the CDA 715 specimen. Significant differences were seen between the three types of specimens: as received, 3-mo, and 6-mo exposures. Examples are shown in Figures 14 and 15 (87521E and 87330E).

On the nickel alloy specimens (CDA 715), the original pits (as received) and those remaining after corrosion in the gas phase were quite rough and irregular. See Figures 16, 17, and 18 (87522D, 87328C, and 87328F). Undercutting of the specimen surfaces was common. However, pitted and unpitted surfaces on liquid phase specimens were considerably smoother. Pit depths ranged up to 50 μ m. Other depressions, found up to 180 μ m deep, likely were related to other mechanisms (e.g., mechanical damage to the metal or electrode clamps).

4.3.4 <u>Interface Specimens</u>

The location of the waterline was easily recognizable on these specimens, both with and without the oxide layers. In most cases, there was only a faint line across the bare metal surface, showing that there was very little difference in the general corrosion in this region. Local corrosion was quite evident on the specimens that received the most exposure (9 mo).

On pure copper, CDA 101/102, isolated areas of dark spots and pits were found just above and below the interface region. The section in Figure 19 (87302A) shows these pits to be about 5 μ m deep. This is comparable with corrosion seen on some of the 95/G phase specimens. Also, it illustrates that some of the dark oxide deposits not removed by the HCl solution can actually cover pits or other forms of local corrosion. The corrosion of the CDA 613/614 specimens was about the same as for the CDA 101/102 specimens. Pits found were comparable, but there was more coloration of the metal surface under the

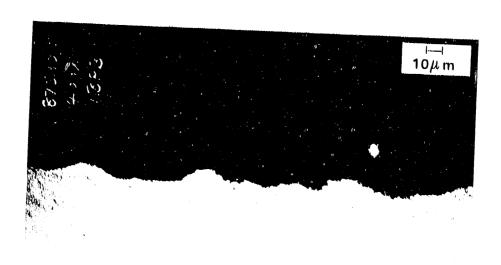


FIGURE 12. Surface Roughness After Electrochemical Prepitting of a CDA 102 Tear Drop Specimen and No Subsequent Corrosion Exposure (400X). Photo No. 87520C.

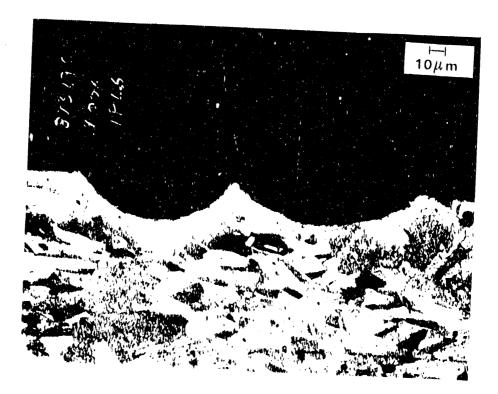


FIGURE 13. Surface Pitting on Electrochemically Prepitted CDA 102 Tear Drop Specimen After 6-Mo Exposure at 95 °C in the Liquid Water (Well J-13) Phase (400 X). Photo No. 87329C.

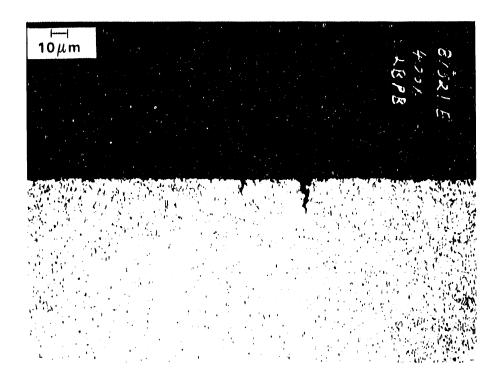


FIGURE 14. Surface Pitting After Electrochemical Prepitting of a CDA 613
Tear Drop Specimens and No Subsequent Corrosion Exposure (400X).
Photo No. 87521E.

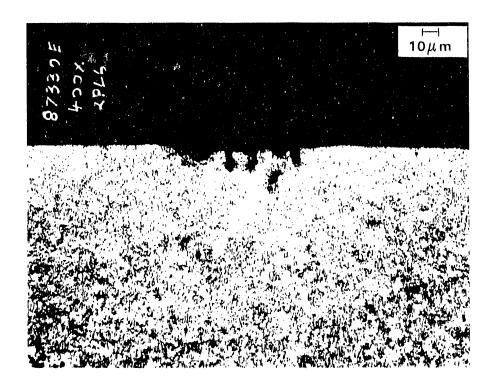


FIGURE 15. Surface Pitting on Electrochemically Prepitted CDA 613 Tear Drop Specimen After 6-Mo Exposure at 95 °C in the Liquid Water (Well J-13) Phase (400X). Photo No. 87330E.

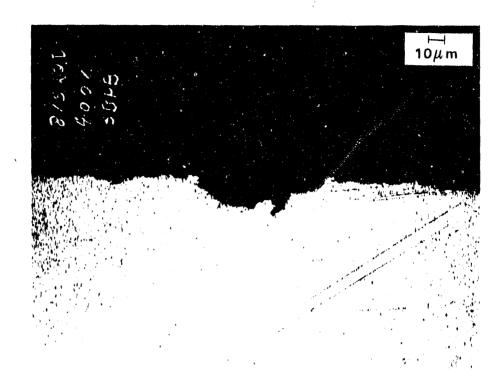


FIGURE 16. Surface Pitting After Electrochemical Prepitting of CDA 715 Tear Drop Specimens and Without Subsequent Corrosion Exposure (400X). Photo No. 87522D.

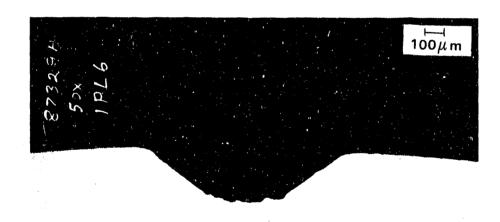


FIGURE 17. Surface Corrosion of Electrochemically Prepitted CDA 715 Tear Drop Specimen After 6-Mo Exposure at 95 °C the Gas Phase (50X). Photo No. 87329A.



FIGURE 19. Surface Roughening on Specimen 1W032 Exposed for 9 Mo in the Liquid-Gas Interface at 95 °C. Photo No. 87302A.

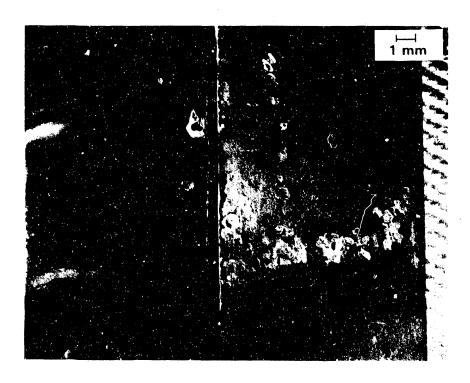


FIGURE 20. The Front Surface of CDA 715 Specimen 3W00B at the Liquid-Gas Interface Position (5X). (The gas phase is at the top. The oxide film was removed from the piece on the right side.)

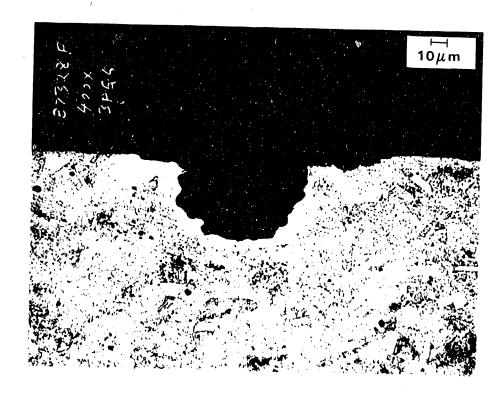


FIGURE 18. Surface Pitting of Electrochemically Prepitted CDA 715 Tear Drop Specimen After 6-Mo Exposure at 95 °C in the Gas Phase (400X). Photo No. 87328F.

oxide layer. There was extensive local corrosion of the interface region of the CDA 715 specimens that received 9-mo exposures. Figure 20 (page 127) shows an extensive crater formation. Figure 21 (87312E) shows the cross section of one crater and the grain structure of the underlying metal. The vertical dimension of the crater is about 25 μ m. These corrosion features are comparable to those seen elsewhere on this alloy, reported below.

In summary, there is no enhanced local corrosion (e.g., metal thinning) at the interface of the gas and bulk water phases (i.e., beyond that observed on specimens totally immersed in either of the phases). Apparently the bulk water is efficient for buffering any acid picked up from the adjacent air phase as far as promoting extra corrosion is concerned.

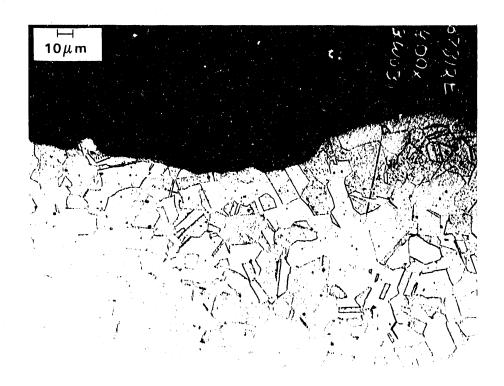


FIGURE 21. Section of Local Corrosion on Specimen 3W031 After 9-Mo Exposure in the Liquid-Gas Interface at 95 °C (400X). Photo No. 87312E.

4.3.5 <u>Metallographic Examinations</u>

From their metallographic examinations, none of the other specimens (not prepitted or interface) showed evidence of subsurface corrosion or cracking. However, there were differences between the alloys in general surface roughness and in the nature of the pits or the local corrosion features.

On the CDA 101/102 specimens, the 150/G specimens showed areas of general surface roughening to a depth of about 5 μ m (Figure 22) (87301B). On the 95/G specimens, both narrow pits (Figure 23) (87303A) and broad pits (80 μ m wide by 12 μ m deep) were seen. These showed adherent corrosion products over the pits. Specimens from the 95/L environment showed surface roughening of about 5 μ m in depth.

For the CDA 613/614 specimens, isolated areas of roughness were seen (depths <5 μ m) on the 150/G specimens. The 95/G specimens, showed the most severe

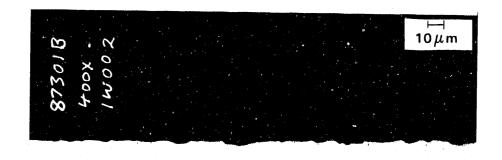


FIGURE 22. General Surface Roughening on Copper Specimen 1W002 After 14-Mo Exposure in the Gas Phase at 450 °C (400X). Photo No. 87301B.

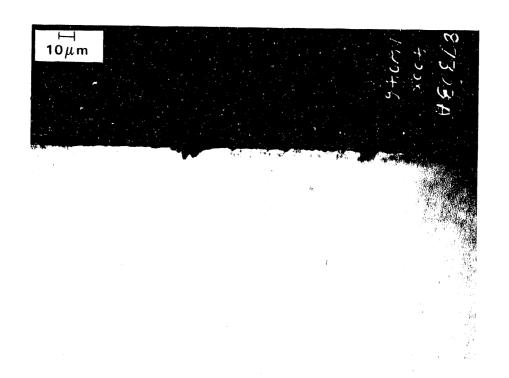


FIGURE 23. Pits in Copper Specimen 1W046 From 4-Mo Exposure in the Gas Phase at 95 °C (400X). Photo No. 37303A.

attack for this alloy. Both roughened surfaces (up to 100 μ m in depth) (Figure 24) (87306A) and adherent deposits of corrosion product over a roughened surface were seen. The 95/L exposed specimens had surface roughening of about 10 μ m and some areas with adherent corrosion products (Figure 25) (87308C) (about 375 μ m wide and 25 μ m deep).

On the CDA 715 specimens, the 150/G pieces showed surface roughness of about 13 μm and some isolated, broad pits. On the 95/G specimens, the surface roughness was about 40 μm and considerable undercutting of the surface was seen (Figure 26) (87323D). From 95/L specimens, roughness of about 15 μm was seen and undercutting was also present. Broad pits also were seen (Figure 27) (87314B).

A broad, smooth and symmetrical deformation was observed on most of the specimens (Figure 28) (87301A), including uncorroded archive specimens. The metal grain structure under these features was also compressed and oriented in layers concentric to the curved surface. These apparently resulted from the specimen fabrication process and were not created in these experiments. In some cases, cracks were seen within the area of compressed metal (Figure 29) (87310E). These features would eventually disappear as the metal was corroded to deeper regions that are uniform. However, this suggests that future testing should include articles that have been prototypically rolled, welded, extruded, or otherwise deformed for a major fraction of the wall thickness. Corrosion behavior in these areas may well be different from that on uniform material specimens.

4.3.6 <u>Visual Observations</u>

Results of all of the visual observations are shown in the data base (Appendix A), along with an explanation of the descriptors used. Detailed descriptions of the specimens that were examined early in the program are in the appendix of Reference 3. As an aid for comparing the corrosion behaviors of the three alloys in the three corrosion environments, the data base was sorted for the incidence of the descriptors for visually observed: 1) pits

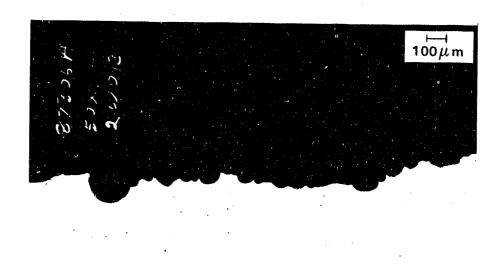


FIGURE 24. Corrosion Surface of CDA 613 Specimen 2W013 From 13-Mo Gas Phase Exposure at 95 °C (50X). Photo No. 87306A.

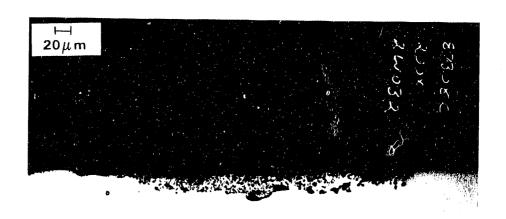


FIGURE 25. Adherent Corrosion Product on CDA 613 Specimen 2W032 After 7-Mo Exposure in Liquid Water at 95 °C (200X). Photo No. 87308C.

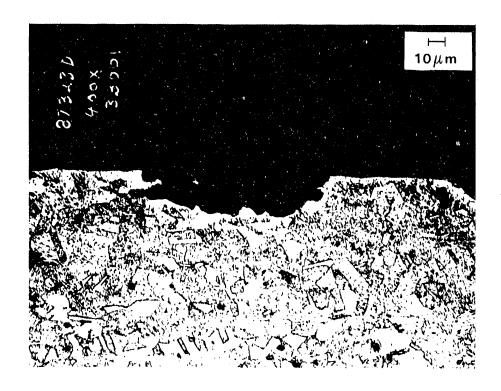


Figure 26. Pit Formed on CDA 715 Specimen 3B001 During 13-Mo Exposure in the Gas Phase at 95 °C. [Note undercut at left side of pit (400X)]. Photo No. 87323D.

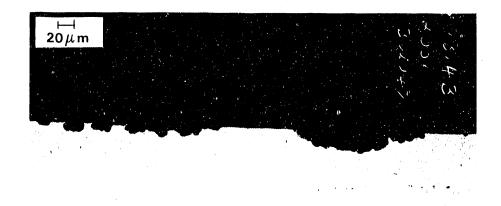


FIGURE 27. Broad Pit and Surface Roughening of Specimen 3W047 After 9-Mo Exposure in Liquid Water at 95 °C (200X). Photo No. 87314B.

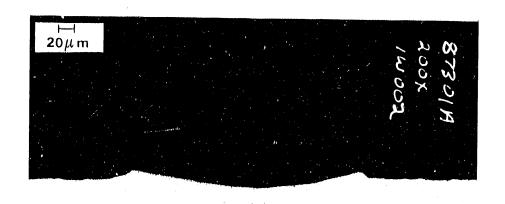


FIGURE 28. Broad Surface Deformations Seen on Most Specimens With and Without Corrosion Exposures (200X). Photo No. 87301A.

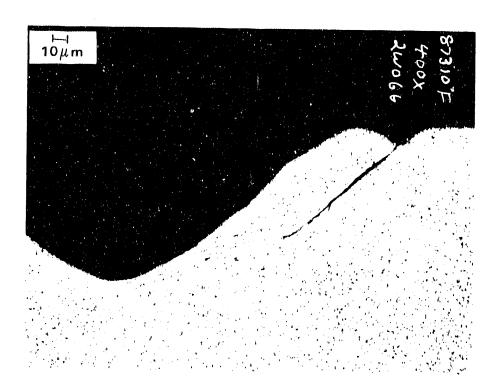


FIGURE 29. Surface Crack Associated with Compression Deformation of Specimen that Occurred Prior to Corrosion Experiments (400X). Photo No. 87310E.

(PT); 2) local corrosion (LC); 3) filiform-like features (FL); 4) no features (NF) or clear surface (CL); and 5) mosaic pattern (MP). Crevice, interface, and prepitted specimens were not included. Full listings of the results are shown in Appendix B. The numbers of occurrences of each feature are summarized in Table 12 as the percentages of the number of specimens in each category in each specimen/exposure-environment subgroup. The various feature types are assumed to occur as independent events. This comparison suggests that the CDA 101/102 material may have a more favorable corrosion behavior than the other materials from the standpoint of local corrosion.

While filiform-like corrosion was seen in the metal surface of the aluminum bronze, only a similar etched pattern was seen on the surface of the nickel alloy. The severe metal surface corrosion, seen on the nickel alloy and designated as MP, mosaic pattern, appears similar to an engraved relief pattern of parallel, but staggered, closely packed dash marks. This occurred mostly on the 95/G specimens and may be related to the greater spread that was found in the general corrosion rate data, discussed below.

Also, a review was made of the results from the very low-flux, (~3 x 10³ R/h) specimens that had received 3.5 mo of exposure. On the CDA 101/102 specimens, the only corrosion feature seen was crevice corrosion starting around the edges of the 95/L exposed crevice specimens. There was a significant difference in the metal-oxygen ratio in the oxide layer, however. The metal oxide ratio of the three 95/G specimens averaged 1.34, while the three 95/L specimens averaged 0.86. For the CDA 613/614 specimens, all of the local corrosion features seen at higher flux levels and longer exposure times were seen on these specimens. The metal-oxygen ratios in the corrosion layer were nearly the same, 0.81 for the 95/L exposed specimens, and 0.98 for those exposed at 95/G. On the CDA 715 specimens, a variety of local corrosion features resulted from the 95/G environment. Crevice corrosion was found on the crevice specimens from the 95/L environment. Again the metal-oxygen ratios were nearly the same for the 95/G exposures (0.91) as for the 95/L (0.84).

TABLE 12

PERCENTAGE OF VISUALLY OBSERVED CORROSION FEATURES

Feature	Material CDA No.	150/G	nvironment 95/G	95/L
CL	101/102	69.	21.	38
or	613/614	34.	0.	5,
NF	715	42.	12.	14,
LC	101/102	11.	24.	10.
	613/614	9.	53.	33.
	715	3.	79.	62.
PT	101/102	23.	55.	29.
	613/614	46.	59.	86.
	715	25.	42.	19.
FL	101/102 613/614 715	0.	none 34. 6.	5. 0.
MP	101/102 613/614 715	3,	none none 24.	5.

Descriptors:

CL = Clear surface; NF = No features; LC = Local corrosion PT = Pitting; FL = Filiform-like; MP = Mosaic pattern.

4.3.7 <u>General Corrosion (Weight Loss)</u>

The amount and average rates of metal corrosion and oxide growth were determined from the measured weight changes of the flat, rectangular (weight loss) specimens. These values are given in the data base (Appendix A) and are plotted in Figures 30 through 38. The visual examination of these specimens, before the oxide film was removed, showed that the corrosion was not at all uniform over the surface of the specimens. Also, crevice corrosion found around some of the support washers likely biased the results.

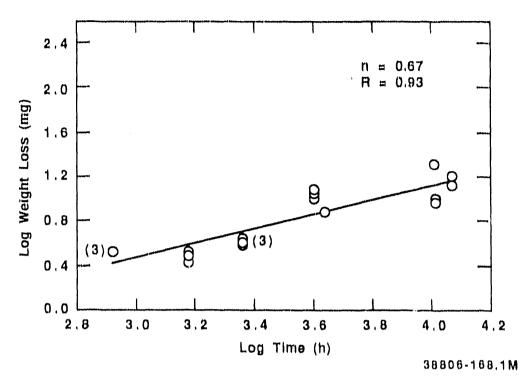


FIGURE 30. CDA 101/102 Corrosion at 150 °C in the Gas Phase.

NOTE: The following information applies to Figures 30 through 38.

- (n) = multiple data points for one symbol or closely grouped symbols.
- 4.0 = nominal gamma radiation flux x_{10}^{-5} R/h in Figures 31, 34 and 37.
 - n = slope of curve Equation (1) in text.
 - R = Correlation coefficient from linear regression.
 - + = Data point not used in linear regression in Figures 32, 35 and 38.

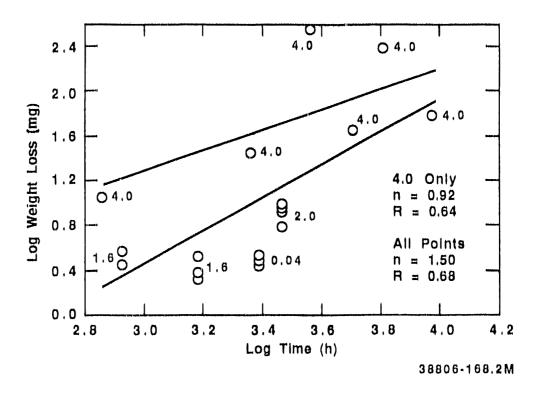


FIGURE 31. CDA 101/102 Corrosion at 95 °C in the Gas Phase. (See Note with Figure 30.)

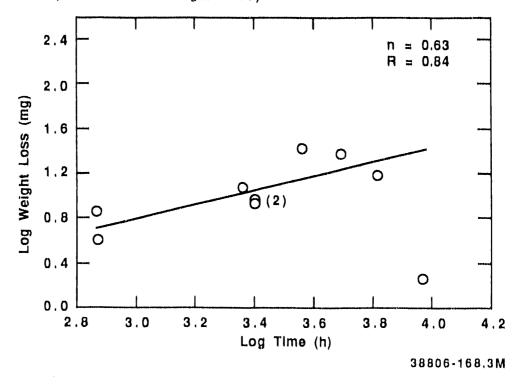


FIGURE 32. CDA 101/102 Corrosion at 95 °C in the Liquid Phase. (See Note with Figure 30.)

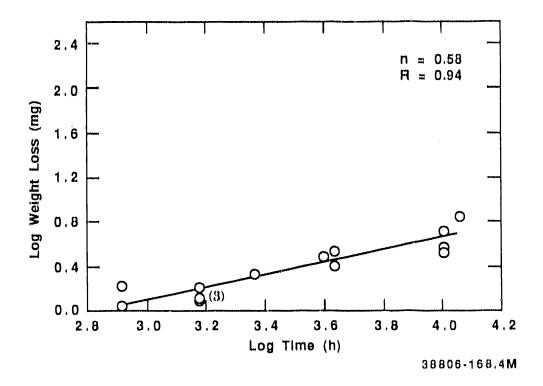


FIGURE 33. CDA 613/614 Corrosion at 150 °C in the Gas Phase. (See Note with Figure 30.)

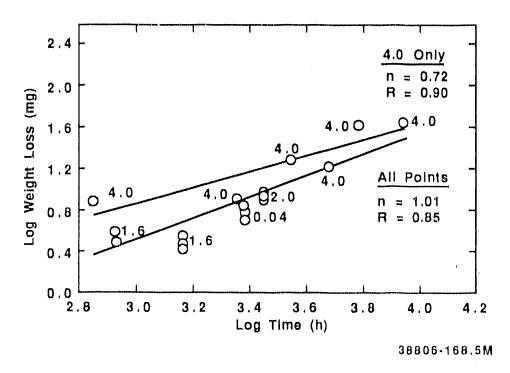


FIGURE 34. CDA 613/614 Corroston at 95 °C in the Gas Phase. (See Note with Figure 30.)

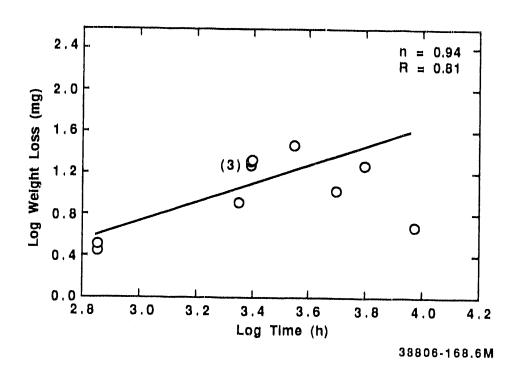


FIGURE 35. CDA 613/614 Corrosion at 95 °C in the Liquid Phase. (See Note with Figure 30.)

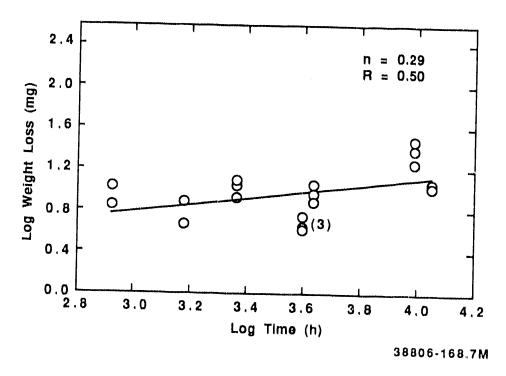


FIGURE 36. CDA 715 Corrosion at 150 °C in the Gas Phase. (See Note with Figure 30.)

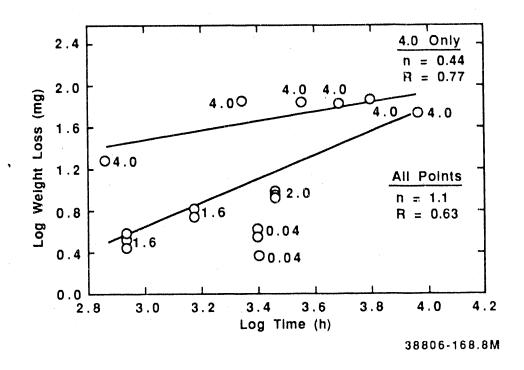


FIGURE 37. CDA 715 Corrosion at 95 °C in the Gas Phase. (See Note with Figure 30.)

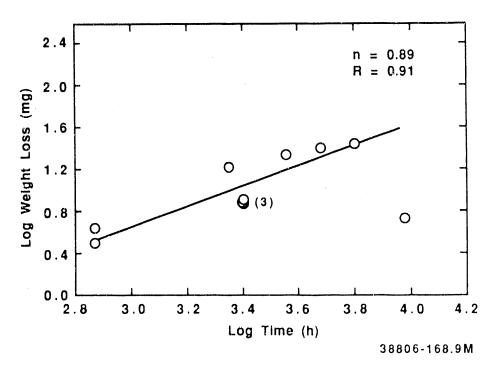


FIGURE 38. CDA 715 Corrosion at 95 °C in the Liquid Phase. (See Note with Figure 30.)

From the equation:

$$W = k t^n + C \tag{1}$$

where:

W = Weight lost by a specimen per unit area

t = Time

k and C are constants.

the order of the time dependency of the corrosion process (n) can be examined from a plot of log W vs. log t. A value of n=1 indicates a linear process while n=1/2 indicates a parabolic process. A linear regression analysis was used to estimate the value of "n" for each of the data subsets. These values and the correlation coefficients "R" are also shown in the figures. In general, the agreement of replicate values within each subset is considerably better than between adjacent subsets. Thus, the data scatter appears to be related to the corrosion process or subtle environment differences rather than to the methods of measurement.

No effect of radiation level could be seen in the data subsets from the 150/G and 95/L environments. However, in the data subsets from the 95/G environments, a general trend appears of increased corrosion with gamma radiation flux. The general data scatter and lack of data at intermediate exposures preclude assignment of a quantitative value to this effect. For these cases, a second linear regression analysis was made of the maximum flux data only.

4.4 OXIDE LAYERS

The third area of examination is the nature and composition of the oxide layers that formed on the various materials and under the different conditions. This information supplements the metal damage results by providing insight into how the corrosion process occurs and what changes might occur in the process with time and changes in the environment. The methods used were visual examination, calculation of the metal-oxygen ratio, Auger electron spectroscopy, and X-ray diffraction.

4.4.1 <u>Visual</u>

The oxide layers on the CDA 101/102 and CDA 613/614 specimens were generally tight and adherent. However, on some CDA 101/102 specimens, there were areas where the oxide dissolved much more slowly in the hydrochloric acid solution than was normal, indicating that the oxide structure was not uniform. The layer of oxide on CDA 715 appeared much heavier than on the other materials and had a flat black surface. The oxide on the 95/G exposed specimens was often covered with many small craters. Apparently small pieces of the oxide layer break loose from time to time. This may be related to the unusual mosaic pattern also seen on these specimens, as previously discussed.

4.4.2 <u>Metal-Oxygen Ratios</u>

Values of weight-of-metal-lost divided by net-weight-gained and multiplied by the atomic weight ratio give the metal-oxygen ratio (M/O). These were calculated for the general corrosion (weight loss) specimens. Values are given in Appendix A and summarized in Table 13. These values can indicate the internal consistency of the data and differences in compositions of the oxide layers in the different environments.

Since the layers were too thin to quantitatively separate from the base metal, the metal compositions of the CDA 613/614 and CDA 715 oxides were not determined and were assumed to contain only copper and oxygen. The ratios for the pure copper CDA 101/102, however, show the average stoichiometry of the oxide layer and can be compared with those estimated from X-ray diffraction and Auger electron spectroscopy results.

The oxide formed on CDA 101/102 exposed at 150/G appears to be $\rm Cu_2O$ compared with the CuO formed on the ones exposed in the 95/L phase. The oxide on the specimens exposed to 95/G may be mixed phases. The oxides formed on the other materials appear to be about the same, regardless of the reaction phase.

TABLE 13

SUMMARY OF METAL-OXYGEN RATIOS IN CORROSION LAYERS

Corrosion			
Conditions	CDA 101/102	CDA 613/614	<u>CDA 715</u>
150/G	2.02 ± 0.31* 2.3**	1.06 ± 0.28	0.99 <u>+</u> 0.18
95/G	1.63 ± 0.28 2.3**	0.94 <u>+</u> 0.20	0.80 ± 0.13
95/L	0.94 <u>+</u> 0.46	0.74 ± 0.34	1.06 ± 0.56
Overall	1.74 ± 0.71	0.99 ± 0.44	0.93 ± 0.31

^{*}Standard deviation.

NOTE: In all calculations, copper was assumed to be the only metal present.

4.4.3 X-ray Diffraction

X-ray diffraction was used for specific identification of the oxide phases in the corrosion layers. Lawrence Livermore National Laboratory examined 22 thin films from 1-mo and 3-mo exposed specimens. (3) Westinghouse Hanford Company examined 18 oxides from specimens that had 5-mo to 16-mo exposures. Most of the Westinghouse Hanford Company work was on powdered samples scraped from the metal specimens in order to eliminate the spectra from the base metal and to remove possible crystal orientation effects. Spectra of the base metals were also made to aid in resolving lines due to the base alloys. Most of the X-ray diffraction samples were taken from the outside surfaces of crevice specimens. Results are summarized in Table 14.

Both Cu_2O and CuO phases were identified on all samples except the following. Only CuO was found on the CDA 614 specimens exposed in a 95/L environment and on CDA 715 from the 95/G environment. The identification of Cu_2O on CDA 614

^{**}From Auger electron spectrometry data.

TABLE 14

SUMMARY OF X-RAY DIFFRACTION RESULTS:
OXIDE PHASES IDENTIFIED

<u>Materials</u>

Corrosion Conditions	CDA 101/102	CDA 613/614	CDA 715
150/G	16:Cu ₂ O,CuO	16:Cu ₂ O(?),CuO 16P:Cu ₂ O,CuO	16:Cu ₂ O,CuO 16P:Cu ₂ O,CuO, (Cu _{0.2} Ni _{0.8})O
95/G	8P:Cu ₂ O,CuO 5P:Cu ₂ O	8:Cu ₂ O(?),CuO 8P:CuO	8:CuO
95/L	5P:Cu ₂ O,CuO	5:CuO	5:Cu ₂ O,CuO

NOTE: The number starting each entry is the number of months of corrosion for that specimen. The "P" designates that the X-ray diffraction sample was on the metal plate. Other samples were powders, scraped from the metal plate.

from the 95/G environment is in question. Of particular interest was the identification of $(Cu_{0.2}Ni_{0.8})0$ phase on the CDA 715 specimen that had received an 150/G/16 exposure.

4.4.4 Auger Electron Spectroscopy

All of the Auger electron spectroscopy results have been reported. (3) Some of these are repeated here for emphasis and to update some of the previous interpretations. While the profiles of atomic composition vs. depth in the oxide layers differed between specimens, the most interesting results are shown in Figures 39 and 40. The very thin corrosion films show a layered oxide structure that has different compositions. The profile in a thin film on an exposed CDA 613 specimen (Figure 39) shows an aluminum-rich layer positioned between the copper oxide and the base metal. The profiles of an oxide film on exposed CDA 715 (Figure 40) show a layer high in nickel content over a layer relatively high in copper content. How these layers are

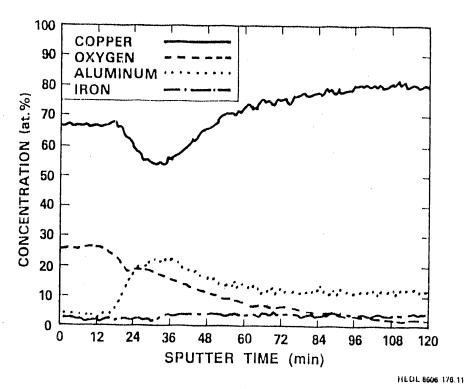


FIGURE 39. Concentration (atom percent) Profile of the Oxide Layer on a CDA 613 Specimen After 1 Mo in the Gas Phase at $150\,^{\circ}\text{C}$. Sputtering rate is the order of 10 nm/min.

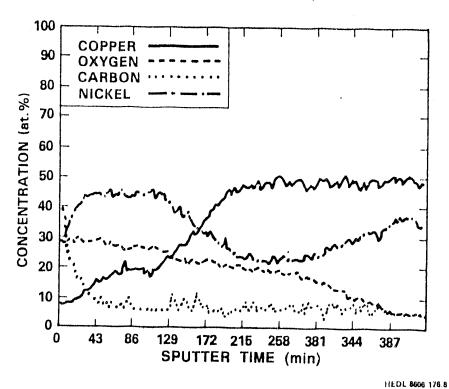


FIGURE 40. Concentration (atom percent) Profile of the Oxide Layer on a CDA 715 Specimen After 1 Mo in the Gas Phase at 95°C. Sputtering rate is the order of 10 nm/min.

distributed as the oxide layers grow was not determined. Since the Auger electron spectroscopy was tuned to the signal from each of the elements of interest as they occurred on the cleaned oxide surface, the data represent the elements in their chemically combined state. The nickel-copper oxide structure was seen by X-ray diffraction on a different specimen. However, no sign of aluminum oxide was seen in the X-ray diffraction data. The decline of the metal-oxygen ratio on the CDA 101/102 specimens was continuous and showed no bands or layers.

5.0 DISCUSSION

This program was basically a survey of the effects of gamma radiation on the corrosion behaviors of three copper-base materials. Since the data are to be used in a preliminary material selection process for nuclear waste barriers, the corrosion exposure conditions were selected to approximate the basic stages for a canister environment in tuff. These conditions were 1) air with a low relative humidity, 2) air at a lower temperature, near the water dew point and water boiling point, and 3) bulk water at its boiling point. The levels of radiation were selected to be much higher than expected in a repository, in order to emphasize possible radiation effects. In the relatively short time available for these studies, emphasis was given to observing as many behaviors as possible and to identifying the conditions that might be severe. These relatively brief exposure times may not have been sufficient for the formation of oxide layers that are characteristic of long-term steady state conditions. Nor was there time to sequence specimens through the succession of the three environments or to run parallel experiments without the presence of gamma radiation.

There is considerable literature on the oxidation of copper, especially at elevated temperatures (>200 °C). However, there is less known about the process at lower temperatures. The oxidation process becomes even more complex for alloys and in the presence of ionizing radiation. At higher temperatures, the normal oxidation of pure copper follows a parabolic time dependency [n = 1/2 in Equation (1)]. Copper ions migrate through the cation-deficient oxide lattice to react with oxygen near the surface of the oxide film. The oxide phase Cu₂O is thermodynamically stable under normal, atmospheric conditions. At lower temperatures, however, migration of the cations is controlled by the gradient of the electric field within the oxide lattice. The theory for this is reviewed in detail by Smeltzer and Young. (10) The oxidation is a logarithmically time-dependent process up to about 150 °C, has a cubic time dependence at increased temperatures, and follows a parabolic time dependence at 200 °C and above. (11)

In these experiments, it is possible that some of the ionized species produced in the moist air environment by the gamma radiation may migrate into the oxide lattice, whereas oxygen does not. Also, other alloy metals in the oxide layer will alter its electronic structure. Since the other metals in the copper alloys will not migrate through the copper oxide lattice at the same rate as copper, it is reasonable that some enrichments of component metals will occur in and under the oxide film. Hence, the oxidation properties expected in these experiments cannot be assumed to be the same as for normal air oxidation of pure copper.

By necessity, much of the fundamental data on the low-temperature oxidation that is likely to occur were taken on relatively thin films. While this information is important, it is the corrosion behaviors on materials with the thick films that are of greatest interest, since the majority of a waste container's service life is spent with a thick corrosion film. If it is assumed, conservatively, that the heavy film will break and allow the environmental constituents to penetrate to the underlying layers, then the process of healing such flaws is also of interest.

The three types of data recorded during these experiments are 1) corrosion environment, 2) metal degradation, and 3) oxide film composition. Of the three corrosion environments, 150/G, 95/G, and 95/L, the 95/G was the most difficult to control in terms of surface conditions on the specimens. This is primarily an experimental problem in temperature and humidity control in a large volume and under remote conditions. Accurate predictions of the sequence of atmospheric compositions in the proposed repository are essential for planning experiments that provide a close simulation.

A summary of observed corrosion behaviors is shown in Table 15. The corrosion rates are values at 10,000 h; the "P" and "L" refer to parabolic and linear time dependence of the corrosion processes (Equation 1), and the numbers are the observed stoichiometries of the oxides formed. In general, the overall corrosion rates are low. At 10,000 h, 0.15 mil/yr is the highest value. It is recognized that for log-log plots of the general corrosion data,

TABLE 15
SUMMARY OF CORROSION BEHAVIORS

6		<u>Materials</u>	
Corrosion Conditions	CDA 101/102	CDA 613/614	CDA 715
150/G	0.02 mil/yr	0.01 mil/yr	0.02 mil/yr
	n = 0.67(P)	n = 0.58(P)	n = 0.29(P?)
	M/O = 2.0 <u>+</u> 0.31	M/0 = 1.1 <u>+</u> 0.28	M/0 = 1.0 <u>+</u> 0.18
95/G	0.15 mil/yr	0.07 mil/yr	0.10 mil/yr
	n = 1.5(L)	n = 1.0(L)	n = 1.1(L?)
	M/O = 1.6 <u>+</u> 0.28	M/0 = 0.94 <u>+</u> 0.12	M/0 = 0.80±0.13
95/L	0.04 mil/yr	0.08 mil/yr	0.07 mil/yr
	n = 0.63(P)	n = 0.93(L)	n = 0.91(L)
	M/O = 0.94±0.48	M/O = 0.74 <u>+</u> 0.34	M/O = 0.91±0.38

NOTE: Copper is assumed to be the only metal present in the oxide layer.

some of the data subsets have considerable scatter (low values of R). Future studies may indeed alter some of these findings. From the present results, however, the following observations are made.

5.1 GENERAL OBSERVATIONS

From these experiments, the most severe corrosion appears to be related to the presence of small amounts of liquid water on the metal surface, either in films or drops. Such small quantities can become concentrated with the corrosive species that are produced by the gamma radiation at the air/water interface and exceed any buffering capacity of the small amount of liquid water. This accelerated corrosion is in keeping with the modeling predictions presented by Van Konynenburg. (2) In contrast, dissolution of these same species in a bulk-water phase will produce a much more dilute solution that is uniformly distributed over an entire specimen. The bicarbonate ion in the Well J-13 water also buffers the pH.

Except for the pure copper specimens (CDA 101/102) at 150 °C (low humidity), the predominant oxide phase appears to be CuO, not the Cu₂O that is predicted by thermodynamics. While Cu₂O appears in most of the X-ray diffraction spectra, it is apparently not present in sufficient quantities to change the overall metal/ oxide ratios. It is also assumed that the other metals are not present in sufficient quantity in the oxide layers to significantly alter the M/O values.

There was no evidence of stress-assisted cracking of the tear drop specimens either from the dye penetrant examinations or from the metallographic examinations. Likewise, there was no additional corrosion of specimens at the interface between the gas and bulk liquid phases. What corrosion was seen in the interface region was comparable to that seen on other specimens mounted completely within the bulk phases. On those specimens that were electrochemically prepitted by LLNL prior to their gamma field corrosion, it appeared that the pits disappeared through the general corrosion processes rather than continuing to propagate. However, there was spontaneous pit formation on all three materials on the specimens that were not prepitted. Some of these pits were captured in the metallographic cross sections. However, thorough examinations and measurements of the pitting were not attempted. Seldom did pits cover an entire specimen with a uniform pattern. They often occurred only in certain regions of a specimen. The pits were normally quite broad, at least as wide as they were deep. In the visual examinations, the distinction of pits versus spots was sometimes difficult. Borderline cases were listed as pits, so as not to allow possible occurrences to go unrecorded.

While crevice corrosion occurred around the edges of the closely spaced pairs of plates, local corrosion also occurred in the mated surfaces well inside of the edges for CDA 613/614 and CDA 715. The metallographic examinations showed uneven corrosion in the region of nonuniform grain structure (e.g., where some surface compressions had occurred during specimen manufacturer). There was evidence of metal component enrichment in the fusion welds of the CDA 613. Although weld study was not part of the scope, no particular corrosion problems were seen in the metallographic examination of the sectioned

welds. It is concluded, however, that any future corrosion studies should include test items that contain the welds, deformation, deep crevices, etc., of a proposed applications design, along with specimens for general corrosion.

5.2 <u>CDA 101/102</u>

Corrosion of pure copper is quite similar under the 150/G and 95/L environments, although the oxide layer appears to be mostly CuO in the latter case. Under 95/G conditions, the corrosion is different. It appears to be a linear process, rather than a parabolic one, and appears to have about five times the rate at 10,000 h. It may be dependent on the level of the gamma flux under these conditions. The makeup of the oxide is intermediate to that produced under the other conditions. In the crevice specimens, local corrosion occurred only near the specimen edges and not in the inner regions of the space between the two plates. The surfaces of the prepitted specimens remained relatively smooth. There was some tendency for this material to pit. but it was not repeatable. The pits were broad. If these pits were due to some undetected impurity in the corrosion environment, then the impurity can be assumed to be at a very low concentration. Control of a significant (pit causing), trace impurity in the operating environment of a repository could be difficult. A few large, isolated pits were found to be from particles of foreign materials imbedded into the surfaces during manufacture of the specimens and not removed by the cleaning processes.

In the visual inspection of the low flux specimens, crevice corrosion was the only type of local corrosion found. Relative to the other materials, then, CDA 101/102 showed much less local corrosion at the low flux.

5.3 <u>CDA 613/614</u>

This aluminum bronze appears to oxidize regularly and relatively slowly in the 150/G phase. The CuO is the predominant product. The parabolic process in the 150/G environment becomes a linear process in the 95/G and 95/L environments. The apparent gamma flux dependency is also noticeable in the 95/G phase.

Considerable underfilm and other types of local corrosion were observed. As with the CDA 715 alloy, the oxide layer is apparently not homogeneous. Hence, it will be more difficult to characterize it under all possible repository conditions and sequences of geological events.

5.4 CDA 715

While the data from the 95/L environment show a well-behaved, linear rate of corrosion with the formation of CuO, the corrosion behaviors in the 150/G and 95/G environments are more ambiguous. The wide scatter in the data from the 95/G environment may be due partly to the gamma radiation dependence and also to a process whereby the small pieces of oxide that break off allow accelerated corrosion in that area. Overall, this may lead to the mosaic pattern found in the metal surface. The scatter in the group of data from the 150/G environment suggests that there is a yet-to-be-determined variable affecting those corrosion results. This may well be a consequence of the structure or properties of the oxide layer. These results suggest that the corrosion behavior of this material is complex under the conditions of those experiments.

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APPENDIX A

DATA BASE

APPENDIX A

DATA BASE

The following tables contain the primary information on the identification, corrosion environment and experimental results for the specimens used in these experiments. Excluded are the specimens that were electrochemically prepitted by LLNL (see text). The information was entered into a Symphony* data base. It was then verified, examined, and extracted for presentation. A column-by-column explanation of the tables follows. The entries are grouped by material and listed alphanumerically by specimen serial numbers.

^{*}Symphony, V1.1 is a trademark of Lotus Development Corporation.

TABLE A1 EXPLANATION OF DATA BASE TABLES

<u>Column</u>	<u>Contents</u>
1	Alloy code: CDA $101/102 = 1$; CDA $613/614 = 2$; CDA $715 = 3$.
2	Specimen form code: W = weight loss; B = tear drop; C = crevice.
3	Serial number within each subset of alloy and form.
	Note: Columns 1, 2 and 3 combine to make the unique specimen identification numbers.
4	CDA alloy identification number, from the chemical compositions.
5	ID number of vessel in which specimens were exposed (T-1, T-2, etc.)
6	Identification of the cage(s) in which specimens were mounted.
7	Exposure periods (sequential) for specimens (1 through 5).
8	Exposure phase: L = liquid; G = gas; I = interface.
9	Nominal temperature of exposure in ^O C.
10	Total exposure time (hours).
11	Total exposure time (months).
12	Nominal gamma exposure dose rates X10 ⁻⁵ R/hr.
13	Measured weight loss (mg) - 'W' specimens only.
14	Measured corrosion film weight (mg) - 'W' specimen only.
15	Calculated corrosion rate mil/year - 'W' specimen only.
16	Metal-oxygen weight ratio - 'W' specimen only.
17	Analytical procedures used: A = Auger electron spectroscopy; X = X-ray diffraction; D = dye penetrant examination; M = metallographic examination; P = photograph(s). All specimens received a lox visual examination.
18	Results of visual examination (see Table A-2 for the description codes.)

TABLE A-2

DESCRIPTOR CODES USED TO CHARACTERIZE THE SPECIMEN SURFACES FROM VISUAL EXAMINATIONS

The codes are pairs of letters as defined below. Adjacent codes describe one type of feature. "/"s separate the descriptions of different features. These are qualitative descriptors; there is overlap in some of the meanings. The descriptors are somewhat suggestive of their meaning; they should be strung together to make phrases. For example, "FWDKSPPT" reads a few dark spots and pits."

<u>Abbreviation</u>	Meaning
Abbreviation AR BK CL CO CU DD DK FL FM FW GE IS LC LG LT MD MP NF PT RO RS RU SM SO	Limited area of coverage Black appearance Clear, clean surface Coloration/stain, not opaque Copper colored Dense Dark appearance Filiform-like (underfilm corrosion n) Film - not opaque, full coverage A few, more than 5 General coverage, most of specimen Isolated occurrence - fewer than 6 Local corrosion, not FL, PT, etc. Large - greater than 10 mil Light/white in appearance Medium - range of 6 to 10 mil Mosaic pattern of corrosion No corrosion features Pit corrosion Rows of corrosion features Residue Roughened surface Small - range of 1 to 5 mil Smooth surface
SP VS	Spots Very small - less than 1 mil
WD XL	On weld Extra large - greater than 100 mil
nM	Located in band 'n' mm wide around the edge of a crevice specimen.

INDEX

Abbreviation	Meaning
Color:	
BK CU DK LT	Black appearance Copper colored Dark appearance Light/white in appearance
Coverage:	
DD IS FW AR GE RO WD nM	Dense Isolated occurrence - fewer than 6 A few, more than 5 Limited area of coverage General coverage, most of specimen Rows of corrosion features On weld Located in band 'n' mm wide around the edge of a crevice specimen.
Size:	
VS SM MD LG XL	Very small - less than 1 mil Small - range of 1 to 5 mil Medium - range of 6 to 10 mil Large - greater than 10 mil Extra large - greater than 100 mil
Surface:	
CL NF CO FM SO RU	Clear, clean surface No corrosion features Coloration/stain, not opaque Film – not opaque, full coverage Smooth surface Roughened surface
Features:	
FL LC MP PT SP RS	Filiform-like (underfilm corrosion n) Local corrosion, not FL, PT, etc. Mosaic pattern of corrosion Pit corrosion Spots Residue

TABLE A-3

CDA 101/102 DATA BASE

Analy Viscal Examination		95 NO S V 30	9 CL/F#0KSPPT?		P ARDKSPP1	1.0 E S 3 E		P GEXLBISP/PI			148786748401.18838	ADP ARSHPT FMPT	CLNF	CL/FWSP	FMDKSPPT/FMLC	CL/ISMBSP	1951 0	נר	FWLGDKSP		D ISPI/CL	נו	DP CL/ARBKSP	ISBKSP //	GELTHP/FWMDDKSF	FWMDDXLTSP/PT?	FINLEDKSP	CLNF	PP Spacer; GESMPT/ARDKSP	Non-data; Spacer; 6EDKSPP;	Non-data; Spacer; GELCPT7/SEDKSP	ARSMSP/PT?	P GEMDOKSP	GEXLDKSP	CLNF	GESHXLDKSP	CLNF	PP Spacer; GELCP1/GEDKSP	GEMPLISP/ARDKSPF17	GEMPDKLTSP/PT?	GEMDLGDKSP	ARXLONSP	GESMDASPPT?
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TABLE A-3

CDA 101/102 DATA BASE (cont.)

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TABLE A-3

CDA 101/102 DATA BASE (cont.)

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0/4			1.81	1.86	1.71	1.58	1.80	1.93	2.17			2.18		1.66	0.62		1.37	16.0		1.13		1.70	1.61	1.38	1.56	1.83	1.56	1.70	2.27	1.43	2.18	2.14	0.65	0.87	2.69	1.17	0.87	1.73	2.13		1.42	0.84	4.15	1.76	D: UD:		2.35
Cor At	(Alliny)		0.0241	0.0170	0.0180	0.1080	0.0320	0.0350	0.0320			0.0310		0.2100	0.080.0	•	0.1610	0.0840		1.7605		0.0550	0.0500	0.0255	0.0520	0.0449	0.0410	0.0190	0.0550	0.0239	0.0300	0.0394	0.0411	0.0635	0.0373	0.1224	0.0636	0.0422	0.0483		0.6002	0.0613	0.0383	0.0720	0.0211		0.032s
3	(6 e)		9	50.01	11.7	5.1	p. 4.	5.2	4 .			1.8		31-2	32.3		54.7	13.9		448.2		3.1	9 .	2.6	10.1	11.6	7.9	3.1	m	4	2.9	3.8	20.B	11.6	3.5	31	11.6		12.3		257.2	11.3	3.5	4	4.5		5, 1
	1 D 6 1		15.8	4.00	10.2	4.4	4.3	41.4	P - 4			7.8		27.1	23	,	46.2	10.9		366.4		2.7	. 3	2.2	8.7	10.2	6.8	2.7			7.b					25.5		9.6			218.5	. d	3.3	3.5	М		ri m
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e:	76081	0::	10.1	14.0	14.0	1.0	M	3.2	3.2	3.2	3.2	6.1	80	3.2	7.0	100	7.0	3.2	3.2	5. 1	3.5	1.2	4.1	2.1	1.1	5.6	4.1	3.5	1.2	3.5	2.1	2.1	6.9	3.5	2.1	5.1	3.5	5.6	9.5	3.5	8.9	3.5	2.1	1.2	3.5	3.5	2.1
1 36	Ē	720	11,593	10,078	10,076	1:0	2,334	2,534	2,334	2,280	2,280	4,416	5,40B	2,280	5, 616	6.40B	5,016	2,280	2,280	3,672	2,499	860	2,927	1,517	2,927	4,007	2,927	2,499	860	2,499	1,517	1,515	6,408	2,493	1,515	3,672	2,499	4,007	4,007	2,499	6,408	2,499	1,517	996	2.499	2.499	1.515
er (ĵ	95	.5€	1.50 0.00	150	95	150	. 150	150	ው ያ	D.	150	45	9	9	υ· Γ	9.5	9	D.	95	95	95	9.5	9.5	95	150	50	φ. (7)	9 5	95	95	150	9.5	56	150	٠. د	Q.	150	150	ç,	95	9.5	Ω. NJ	φ.	56	416	150
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"A" data on gas part "A" data on gas part

ARDKRSLC

FWDKSMSP

GEPTLC/ARDKSP

ARSMEKSP

RU/ARDKRS FWDKSRSP

CL/FWLC

Allay Type Ser No.

Visual Examination

LCIM ARDKRS/NFCL ARSMPILCIM

1.82 4.00 1.96 1.65 Cor Rt 0.0036 0.0310 0.1090 0.0700 0.0218 0.1620 0.0243 0.0680 0.0310 0.2600 CDA 101/102 DATA BASE (Cont.) Wt Loss Film Wt (mg) 15.9 3.2 7.8 10.5 F1ux (XE-5) 4.1 0.67 1.5 4.1 0.033 0.67 4.1 2.5 2.5 0.031 0.037 4.6 0.98 4.6 1.5 Time (mon) 6.1 1.2.1 1.2.1 1.0 1.0 1.0 1.2 1.2 1.2 1.0 1.0 1.0 836 11,593 4,416 11,593 1720 5,736 5,736 4,416 720 10,078 2,334 Inme (n) 2,280 2,280 2,280 5,304 4,007 3,672 2,499 5,304 Phase Temp (C) CDA vessel Cage Runs

ARVSSPPT?/ARLCIM?

ARLCIM ARLCIM

ARLCIM ARLCIM? ARLCIM

ARLCIM

ARLCIM ARLCIM

TABLE A-3

CDA 101/102 DATA BASE (Cont.)

Non-data; Spacer; GELCP17/GEDKSP ARSMSP/P1? Non-data; Spacer; GEDk SPFT PP Spacer; GELCPT/GEDNSP PP Spacer;6ESMPI/ARDKSP GEMPLISP/ARDASPPIT Visual Examination GESML I BASP / ARVSPT GELTHP/FWMDDKSP FWMDDKLTSP/PT? GEMPORL 15P / P17 FMDKSPPT/FWLC CL/FWDASPPT7 ARSMPT/FWPT **BESMOKSPPT?** GEXLBKSP/PI **GENDBKSPPT? GESMXLDKSP** GEMDL GDKSP CL/ISMDSP CL/ARBKSP CLIFWDKSP CL /FWRDND FULGDKSP GEXLDKSP **GEVSDKSP** ARBKSPPT ARDKSPPT FWLGDKSP GENDOKSP LL/FWSP ISP1/CL **ISBKSP** GESMP1 1591 CL NF Anaiy ADP a. c a. (2) (L 9/2 Lor At 'Bg) (B11.yr Wit Loss Fiim Wit inste Flux (son) (xE-5) 1.3 3.1 2.6 2.6 0.027 0.019 0.027 0.027 5,016 2,499 2,499 1,515 5,304 4,416 2,334 2,280 2,334 2,334 2,280 5,489 3,672 6,408 4,007 6,408 4,007 5,304 6,408 3,672 6,40B 2,499 2,1B3 B. 423 7,499 10,078 95 95 150 150 150 Frase Temp 95 95 150 150 95 Vessel Cage Runs 12345 1234 . 101 100 101 101 101 101 101 101 101 Allay Type Ser No.

TABLE A-4

CDA 613/614 DATA BASE

in Viscai Examination		GEDRING INCHARDED			ROPICOLNA	ARD*FM/ARVSSPT				6EVSL 1SP	רוכיבניקר	GREATEDFOR	ARCUCOFFRPT			GEVSPT/ARCUCO	GEDAFM/CL	ROPT GEDNER/155P	GEDAF#/CL	GEFL/ARP1/ROP1	GEFL	ARCUCO/65VSSPPT	ARCUCO/6ESMP1/80P1	BESHP1: ISLC		GEDS-FR - ARVINGP	CUCOSPAD/ARCUCO/6EVSSP	SEFL/SEL15P	ISP1/CUC0:01	10年10日日	DRIGHT / GEDAFF	GESASPOTA	GESMSPP1/FWLGP1	ARDAFMUSELTDASP	GEVSPT/FWXLFL		GELGPT	# Non-data;PP Spacer: 6ExtFill	PP Spacer; EEX.FLLC	PP Spacer; GEALFLLC		CL ROPIMD - ARDNEM FM / FMF1
Analy				=1		u.		a. G	O.						ADF	Œ.		CI		ű	CL.		<u>ن</u> ا												a.							
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TABLE A-4

CDA 613/614 DATA BASE (Cont.)

Visuai Examination		CLAF	MF-JUSHM	112		سا ند آما ل نات :	4. 1	L iš		GEPT/ARLE	N.N.	ISLEIM	ARLCIM'15ft7/COF#	ARLCIM?/CUFM	u.		ARLCIM?/NF/CUFM	ARLC	SEL CFL	ARLCIN/CUFM		ARPTLCIM/CUFM	GELCPTFL	GELCATFL			151.			LL13 / LUTA		GEDNEM/GEVSDNPTSP	おいまりはい、いいないにはつ 日日	TAT data on gas part	GEDNFM: ARCULE	GEDW.FM.ARCULC	GEDAF#/GEVSPTSP	GEBNFM/CL	BENEFI FREE FAR		Telegion and the second and the seco	ಿ. ಇ. ತೂಕತ ವಾ ಸುವರ್ಭದ ವರ್ಣ	GERU/Furci		
Anal		u	u	u.	ir.	d.	1 . (ı) .	Œ	a.	í).	u. T	a.	ιτ	A L	a.	AR	A K	n.	α.	a.	a.	a.	Œ	a.	aL (n. (1 . (a. (1 . (ı.	a.	a.	Œ.	Œ.	n.	a.	u.	GL.	ų.	G.	G.	ď.	a.	n.
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i/i	: († E																																tr.		12	α. ~	, v.	2.6	म			1	0 7 7 7		2.7
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		-	(D)	5.5	: 4 - 4	1.0		ö	1.2	C)	14.0	J - 1	Û - T	3.2	3.2	3.2	3.2	3.2	7.4	3.5	5.1	٠, د	7.4	6.9	9, 9	2.1	5.0	3.5	3.5	آ	5.6			0	0 4	4.	1.41	4	, ,	o-		(-	13.1	9.5	1.0
Q 1	È	4	5,736	11.593	e B	720	11.593	4.416	B 3c	5.736	10,078	720	720	2,280	2,334	2,334	2,280	2,280	5,304	2,499	3,672	2,499	5,304	6,408	6,408	1,515	4,007	2,499	2,499	3,672	4.007	320	11.593	022	10 07B	0.00	11.593	4 4	4:4	A CA	9 7 8	730	9.4°5	5.40B	720
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TABLE A-4

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	i.			c 6	4	1.17	0.67	1.07	0 74	1	6		0 -	11.1	74-1	25.0	1 (0 * 0			;	1.41			0.81	0.87	10-1	1.01	0.84	0.80		0.58	0.95	1.08	- m	86 °0	12.1	5.27	3. T	1.15	0.47	0.78	2.01	0.78	0.93	15.0		55.0	0.87	0.95	06-0
·	ب ن ن ن ن ن ن			u : 10	95000	9.2170	0.0580	0.0400	6.0071	•	4010 0	90100	0.010	0 0140	0 0000	0.0710		244010			0	05/0-0			0.0420	0.0710	0.0366	0.0569	0:1572	0.1659		0.0612	0.0505	0.0148	0740-0	U. 0153	0.010.0	0/10/0	4. U.D.	0-1655	0.1311	0.0710	0.0209	0.1621	0.0482	0428		0.1107	0.0570	(Paca-	0.000.0
(Cont.)	## *\$ El	è		ď	1	9.	4.5	2.1	6.		7.9	a c		u i	1 00	23.9	17.0				C.	7-7			7-4	et :			25.9			27.3														6.8			11-1		
A BASE	Mt (.055 F	e di		3.4	7. 4	7.9	3.2	1.1	9.5		2,3	2.3	, , , ,	3.1	0.	17.9		:			q	r d			M 1	64 4	2.8	7-2	٠ - ك 1	21		()- ()-	0	√η «	t ·	- · ·	7-1	7 - 7	, t	> - 10	7.7	١٠٠	i - 6	20-5	7.7	4.		30.6	an Tr	9 · 6	ir ib
14 DATA		(XE-5)	-	0.00	4	4	4.5	4.6	6.42	4	Ģ.	4	4	1.5	4	4	4	. 4	ч	- «1	r 0		T 6	- n	9 .	1 - 1	0	0.041	0.04	0.04	0.04	0° .	0.041	u	p u	7 6	7 .	- v	7 0	r ·	• .	1.6	U. 51	ر. د. د.	9-1	0.041	0_04	u r	(4	r4	ιı
613/614	1.00	00 8	(·	16.1	13.1	1.0	1.6	1.4	14.0	1.0	3.2	3.2	7.7	6.1	3.2	7.0	7.0	2.2	3.2	1 2	1 7	4 6	r u	n .	7-7	7 .			ر. د د	ν,	, ,	or i		י ה ה	 	,		4 Y	i v	- c	י ה נו	7-1	7.7	7	7 - 7		٠, د	 		4.1	;! \$T
CDA 6	1 3 6	ē	ü	11,593	9,408	720	720	835	10,078	720	2,334	2,334	2,334	4.416	2,280	5,016	5.016	2,280	2,280	2,280	2,280		, ,		/ 170.7	200	1,317	7,44,7	7.474	7,479	444	6,408	2,475	, ,	700	700,1) ir	700	7 477	7,0,7	0 0	9 40	1,113	****7	1,31/	7,445	2,499	3,672	17.07.1	2.527	2.927
	e di	ن	ý	150	9	5	47	150	150	9.	150	150	150	150	45	20	رم دريا	ę.	55	5	47.		יי ני	7 u	ים מים	1 V	7 w	T 0	מי נ	بر ان ان	ጠ i	pr c	T 4	1 0	٠ ا د	9 5	. r.	051	50	י ני	1 y 10	7 (200	ין ע ר מ	ָר נ <u>י</u>	r) (n De l	vn i Or i	ς, (Σ.	ę, L	ር ኒ
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ABLE A-5

CDA 715 DATA BASE

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÷ E e c c c	ti.	u,						.21	(2)					ADP				C)	L		(L.																			a.	
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4 CC	27	10.1	1.0	15.1	्र । इ.स.	- C		<u>د</u> • .	[1]	· (*)	11.7		1.2	1.0	13.1	3.2	3.2	3.2	3.2	7.0	3.2	3.5	7.0	7.5	5.6	5.6	89, 9	3.5	6-8	4.1	2.1	3.0	ri Fi	5.1	3.0	5	7.4	9.8	5.5	(I)	ירו ירו
1 mg	9,40B	11.593	720	11.593	10,07E	4.415	(1) (1)	2) /34 /5	40 (CI)	## 	E 4 €	4,410	836	720	9.408	2,334	2,334	2,334	2,280	5,015	2,280	2,499	5,015	2,499	4,007	4,007	5,405	2,499	6,405	2,927	1,515	2,183	2,499	3,672		3,572	5,304	6,408	2,499	6,408	1,450
e ()	un a-	150			150	150	±5.	Çt.	150	u 1 (r	ुद	150	150	75	9.5	150	150	150	ς. 21	α· ις	95	95	9.5	45	150	150	95	95	gr.	95	150	ር የን	IJ.	95		95	φ. V1	ፈግ (ታ	55	9.5	in Gr
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ó	'n	۳1	٠,	۲,	rı	1-1	,,	~ 7	m	• 1	m	ĦΊ	n	1-1	1/1	۲٦	ω,	۲۱	۱	11	М	n	m	11	M	r1	۲۱	۲1	r:	M	n	**)	۱,	11	111	Fil	M	М	MI	P 1	11

TABLE A-5

CDA 715 DATA BASE (Cont.)

																																					٠										
Visual Examination		GEVSPIIM/CLNF	GEVSP11M/CLNF	GESHPTLC2M	- CLNF/DRFM	CLNF/DAFM	CL/DKFMISLCIM	1 SLC / CLNF	CLNF	CLNF	ARVSP11M/CLNF	FWLGDKSP/GEDKFM	FWLGLC/GEDKFM	GEDKF1:/CLNF	CLNF	GESMMDLTSP/ARVSPTIM	CLNF	GEVSPIIM	ARLCFL5M/GEDKFM	GEVSPT2M/ARSMPTGE	ARPILC/ARDKFM	ARLCFL	ISI C/F! NF	CLNF			GEVSPTIN	GEVSPTIM	CLNF	ARVSPIIM	FWXLLC1M/GEDKFM	GELC/RUAR/SMPT/ARRU	CL/6ERU	CL/ISVSPI/RU		GEDKFM	ARBREM/ARLISP	GESALTSP		13/3/51	"A" data on gas part	GESALTSP	CL/GERU			"A" data on liquid part	CL/ISVSPT
Analy		AAP	O.	o_	AKP	O.	OL.	α.	۵	α.,	O.	O.	AXP	AXP	Q.	AXP	AXP	a.	a.	α.	a	a	a.	. съ.	α.	. α.	Q.	α.	O.	o_	a.	۵.	a.	هـ	A P	a.	ΑP	OL.	Q.	α.	O.	Q.	o_	AXP	a.	ΑP	α.
u, 0																																0.64	0.97	1.20	:	0.88	1.02	1.47		0.78		1.73	1.03				0.93
Cor At	(M11/47)																															0.4700	0.0480	0.2110		0.0164	0.0420	0.0310		0.0830		0.1180	0.0410				o.1480
	E. (5E)																															26.7	34.8	12.1	1	13.9	13.1	8.9		4.5		5.5	58				8.8
																																19.2	27.6	10	,	10.8	10.5	7.6		3.4		4.8	23.3				۲.
33 3. U 13. U 14. A	. FE-37	7.	ÿ.75	4.1	1.6	0.42	2.5	4.6	2.5	1.5	4.9	2.5	2.5	1.6	4.6	4.1	2.5	4.1	2.5	4.1	0.67	0.67	0.033	0.75	2.5	1.5	4	0.037	0.31	0.037	0.033	4	0.42	4.6	4	9.78	1.5	5	٠.	4.9	~	4 .	0.42	9.		母	. q.
e :		0.1	6.1	1. Ů	1.2	14.0	В.0	1.2	10.1	6.1	8.0	16.1	1.6	3.2	3.2	3.2	3.2	3.2	8.9	8.9	7.4	7.4	3,5	5.6	5.1	5.6	5.1	3.5	2.1	3.5	3.5	1.0	14.0	1.2	1.0	16.1	9.1	6.1	6.0	1.0	1.0	1.0	14.0	1.2	0.1	0.1	1.2
E 4		720	4,416	720	636	10,078	5,736	B36	11,593	4,416	5,736	11,593	720	2,334	2,334	2,280	2,280	2,280	6,408	6,408	5,304	5,304	2.499	4,007	3.672	4,007	3,672	2,499	1,515	2,499	2,499	720	10,078	836	720	11,593	4,416	4,416	6,408	720	720	720	10,078	836	720	720	836
а Ст.	ر	9.5	150	9.5	150	150	95	150	150	150	95	150	95	150	150	95	95	95	95	95	95	95	95	150	95	150	9.5	95	150	95	9.5	95	150	150	95	150	150	150	9	95	95	95	150	150	9	9	150
Phase		_	9		Ö	9	9	9	9	G	_i	9	C)	9	S		ഗ		S	ب	9	9	ഗ	ß	9	(D	ب	_	9	_	ເລ	9	9	G	9 (ء.	9	_O				ب	£.	G			
Runs			~		_	1234	123	-	12345	2	123	12345	-	2	2	. 2	2	2	34	34	1234	1234	_	45	4	4.5	4	_	57	_	_	-	1234			17545	~	m	4	_	-	-	1234			_	
Cage		a	I	a	z	99	W	۵.	C	a	~	ນ	2	I	a.	_	0	0	0	<u> </u>	-	=	w	T	0	Q.	۵	X.	ш	E	ш	ب.	99	α.	ً بـ	I Z	உ	a.			1		ອີວິ	OL.	ي	ابد	D.
Vessel		-	7			7														-		m		2		2	-	47	2	4	4	-	7	C4	 (7	24	7		-		-	(.4	C4 ·	-	1	i 4
CDA		715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	1.15	715	715	715	715	715	715	715	715	715	715	715	7.15	01/	01/	Z .	Λ. I	715	715	5	1.5	715	7	71.7	717
S S	:	~	4	S	40	7	œ :	0.1	=	12	13	4	2	1.6	1 7	9	21	23	24	25	26	23	28	58	30	31	32	33	4	35	3.7	-	N	n	₹ ,	n	a 1	٠. (י מ	۰ :	ري 		1.2		4	uo 0	*
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CDA 715 DATA BASE (Cont.)

APPENDIX B

SUBGROUPS OF SPECIMENS SORTED BY ALLOY AND SPECIFIC CORROSION FEATURES/CONDITIONS

Note: Explanation of columns and codes are in Tables A1 and A2 of Appendix A.

ABLE B-1A

SPECIMENS SHOWING EVIDENCE OF PITS (Crevice and Interface Specimens Not Included)

Visuai Examination			GEMDDK SPP17	GEDKSPLCP1 .	ARDKSPLCP17/CL/FWSP	CL/ARVSSPPT?	GESMP17D+SP	GESMDKSPPT?	ISP1/CL	FWR00KLTSP/PT?	GEDKRSSPLCP:	GEDKRSSPLCPT?	GED#SPLCP17	GEMPDKLTSP/PT?	GEMBBk SPPT?	PP Spacer;GELCP1/GEBKSP	PP Spacer; GESMPT/ARDKSP	ARDKRS/GEXLLCPT?	GEXLBNSP/PT	GESMPI	CL/FWD+SPPT?	ARSHSP/PT?	GERUZARVSPT	GEMPLISP/ARDKSPPI7	GESML TOKSP. ARVSPT	1981	CL/FWSP/1SPT	GEPTLC/ARDKSP	GEPTLC/ARDNSP	FWDKSPP1/FWLC	ARPILC/ARDKSP	ARBKSPPT	ARDKSFPI
a.c.		g u							a												C					Ċ							
æ Æ			1.79	1.76	2.27	4.15	2.18				1.56	1.56	1.61					1.42					0.62				1.93	1.65	1.71		1.86		
Cor. Ht	emali vro		0.055	6.072	0.055	0.0383	0.03				6.052	0.041	0.05					6.6002					û. ûB				0.035	0.038	0.018		0.017		
	(6w)		3.1	47	۱۳۱	3.5	2.9				10.1	7.9	9.6					257.2					32.3				5.2	24.9	11.7		10.9		
Wt LOSS Flim Wt	(ĎE)		2.1	3.5	7.7	3.3	2.b				8.7	B. 4	8.3					218.5					7.3				9.	21.6	10.2		9.6		
	(#00) ()E-2)	1.3	1.0	1.6	1.6	1.6	1.6	4.6	1.3	0.019	7	2	7	1.3	1.3	4.6	4.6	च	1.3	2.6	7.6	0.027	4.4	2.6	2.6	3.1	4.6	0.42	0.42	7.1	0.42	64	2
 e	1000	1.0	1.2	1.2	1.2	2.1	2.1	3.0	3.2	3.5	4.1	4	4.1	5.1	1.0	7.4	7.4	8.9	13.1	1.0	1.0	3.5	7.0	6.8	13.1	3.2	3.2	14.0	14.0	14.0	14.0	1.91	16.1
1	(H)	720	098	098	960	1,517	1,517	2.183	2,280	2.499	2,927	2,927	2.927	3,672	5.016	5,304	5,304	6.408	9,408	720	720	2,499	5,016	6,408	9.408	2,334	2,334	10,078	10.078	10.018	10.078	11,593	11,593
- - -	()	9.5	9.5	5	i,	95	95	2,6	9.5	95	9.5	95	95	95	6.2	45	6.5	95	95	95	95	95	9.5	95	9.5	150	150	150	150	150	150	150	150
9 2 B C 7		ıo	ťΩ	9	د،	Ġ	9	9	9	ıσ	ß	ψ	9	Ġ	9	9	9	ō	9	ب.		ب	ب.	ا.		Ġ	9	g	U)	9	و ر	ß	co
⊕ un ¥				۳		•	4	64	2		13	1.2	1.2	4	23	1234	1234	3.4	1234		_		23	34	1234	. 7	7	1234	1234	1234	1234	12345	12345
ا باق	S	u.	S	S	S	Ŋ	S		u.	>	88	5.5	5.5	LL.	LL,		_	_	ц.	Œ	₫	×		a	Œ	7	o.	66	99	88	99	9	33 63
Vessei Lage	NO.	1	ריז	~	~		~	m	-	4	~	~	r			۰,٦	~	-	. •		-	4				2	7	r4	2	۲4	7	. (*4	1.4
400		101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101
u C	No.	Ξ	4	65	5 5	6.4	25	M1	20	26	4	40	2.4	40	<u>ن</u>	38	58	67	7	S	Гd	32	3.4	39	0	91	27	r.a	4	4	101	M	47
- c		æ	*	3	3	3	×	œ	œ	œ	3	.3	3	æ	8	αħ	CE)	3	æ	Œ	E	æ	3	Œ	æ	æ	*	3	3	æ	3	æ	œ
Hilo, Type ser		-		-		-	_			****	-		1	~	-			1				••			-								

TABLE 8-1B

SPECIMENS SHOWING LOCAL CORROSION (Crevice and Interface Specimens Not Included)

Analy Visuai Examination		ARDKRSLC	GEDKSPLCP:	ARDKSPLCPT?/CL/FWSP	GEDKRSSPLCPT	GEDKRSSPLCP19	GFDKSPI CPT?	PP Sparp: GFI CPI / GFDKSP	ARDKRS/GEXLLCP1?	CL/FHLC	GEXLDKSPLC	SEPTLC/ARDKSP	6FPT1 / ABB/KSP		ARPILCIARDKSP
0 <u>r</u>	ı	1.65	1.76	2.27	1.56	1.56	1.61		1.42	1.80	0.65	1.65	1.71		1.86
Cor At (mil/yr)		0.26	0.072	0.055	0.052	0.041	0.05		0.6002	0.162	0.0411	0.038	0.018		0.017
13 m 12 t		12.1	4	m	10.1	7.9	9.6		257.2	7.5	20.8	24.9	11.7		10.9
Wit Loss Film Wit (mg) (mg)		10.5	3.5	2.7	B. 7	B.9	8.3		218.5	9-9	15	21.6	10.2		9.6
Flux h (xE-5)		4	1.0	1.6	7	2	7	4.6	4	4.9	4.9	0.42	0.42	2.1	0.42
Time (mon)		1.0	1.2	1.2	4 . 1	4.1	4.	7.4	8.9	1.0	8.9	14.0	14.0	14.0	14.0
Time (h)		720	998	9.60	2,927	2,927	2,927	5,304	6,408	720	6,408	10,078	10,078	10,078	10,078
Tesp (C)		9.5	95	95	95	95	95	95	95	95	9.5	150	150	150	150
ص بو بو بو		9	D	9	9	9	ഗ	Ġ	Ð	_	٦.	9	9	9	9
Runs		-	2	~	12	12	1.2	1234	34	-	34	1234	1234	1234	1234
Cage No.		-4	S		25						-			89	99
Vessel No.			Μ)	~	3	m	m	רו	-		-	7	2	7	L1
CDA		101	101	101	101	101	101	101	101	101	101	101	101	101	101
Ser No.		1.7	65	4	4	40	42	3.6	9	12	Ω 4	7	24	4	23
l y p e		x	3	3	3	æ.	3	Œ	3	3	x	38	3	œ	л
Alloy Type			-	-	-					-	-		-	-	

TABLE B-1C

SPECIMENS HAVING A CLEAR SURFACE OR NO CORROSION FEATURES (Crevice and Interface Specimens Not Included)

Stantagen (France)			CL /FWSP	ARDKSPLCPTZ/CL/FWSP	CL/ARVSSPPT?	ISPI/CL	כר	נר	כר	RU/15DFSP/CL	CL/FWLC	CLIFWDKSPPTP	CL/APBKSP	נר	j.	CL/GELISP	כר	CL/FWDESP	CL/FWRGWE	כר	CL	CLNF	ָּ	נר	CL/FWSP/ISPI	נר	כר	נו	כר	CLNF	- LT	CL	CLNF	CLNF	CLNF	CL/1SMBSP	CLNF	CLNF	כר	נר	נר
्त ट ा						â						0	DP					a																							
c Y) :			2.27	4.15		1.89	1.43	1.70	1.58	1.80			0.91	0.84	0.87	0.87				2.14		2.35	2.69	1.93	2.17			2.13		1.83	1.73		2.18	1.95		1.90		1.82	1.90	1.81
() ()	mil/yr)			0.055	0.0383		0.0211	0.0239	0.019	0.108	0.162			0.084	0.0613	0.0636	0.0636				0.0394		0.0326	0.0373	0.035	0.032			0.0483		0.0449	0.0422		0.031	0.031		0.031		0.0243	0.0218	0.0241
3 6	, toe,	,		m	3.5		3.4	4	3.1		7.5			13.9	11.3	11.6	11.6				3.8		3.1	3.5	5.2	4.8			12.3		11.6	=		6.7	8.8		B		18.1	16.2	18
33 6 7 7 7 7 7 7 7 7 7 7	. (EG)	,		2.7	M . 1		۳1	3.4	2.1	4.4	9.9			10.9	8.7	۰	6				H.		2.8	3.2	4.6	4.3			1.1		10.2	9.6		7.8	7.8		7.B		15.9	14.3	15.8
,				1.6	1.6	1.3	0.041	0.041	0.041	9	4.9	2.b	2.6	4.9	0.04	0.04	0.04	3.1	3.1	3.1	0.31	1.5	0.31	0.31	4.0	4.6	3.1	3.1	1.5	1.2	1.5	1.5	1.2	.5	1.5	1.2	1.5	1.2	0.98	96.0	96.0
•	(000)		1.0	1.2	2.1	3.2	3.5	5,5	3.5	1.0	1.0	1.0	3.2	3.2	3.5	3.5	3.5	1.2	1.2	1.2	2.1	2.1	2.1	2.1	3.2	3.2	3.2	3.2	5.6	5.6	5.6	5.6	5.6	6.1	6.1	6.1	6.1	11.7	16.1	16.1	16.1
	(P)		720	୍ବର	1,517	2,280	2,499	2,499	2,499	720	120	720	2,280	2,280	2,499	2,499	2,499	836	636	836	1,515	1,515	1,515	1,515	2,334	2,334	2,334	2,334	4,007	4,007	4,007	4,007	4,007	4,416	4.416	4.416	4,416	B, 423	11,593	11,593	11,593
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, ,	11 10 10 11 11		(9	9	ເລ	9	9	وي	ى		ب	ر.	_	-4			ب.	ß	9	9	9	ß	മ	ß	9	ō	വ	ß	ى	9	9	9	ß	(4)	9	ິນ	IJ	Ð	9	ß	9
	rage ners			<u>د</u>	4 8	e1			-				2 4	1 2		۲ ا				1 1	E 5	5 8	E S	5	F 2	۲۷ دء	3 2	2 5	P 45	3.45	F 45	P 45	A. A.	15	m a	5	10	3.45		NHH 12345	_
				m	M		4	đ	4	•~•			-		4	4	4	2	7	7	C4	2	7	7	2	7	2	7	2	7	2	7	£4	71	7	7	2	7	2	1.4	Z
000			101	101	101	101	101	101	101	161	101	101	101	101	101	101	101	101	101	101	101	101	101	101	161	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101
			13	4	40	20	99	51	4	25	1.2	7	22	۳	604	55	56	່ວ	80	Or-	5.5	28	9	53	7.	C1	17	12	9	m		ń.) Ģ÷	92	31	J	5	9	-	4	1 1	22
	 			3	3	H .	3	3	3	3	3	CECI	a	3	2	38	3	cca	in	œ	3	œ.	3	3	3	3	aı	caca	3	Œ	3	3	co	. 2	3	æ	3	υn	**	3	3
	ī					-	-	-	-					-							_	-	+	-	7					•			***			1					

TABLE B-1D

SPECIMENS SHOWING FILIFORM-LIKE LOCAL CORROSION (There were no pure copper specimens showing these features.)

ABLE B-IE

LOW FLUX SPECIMENS

ty Visual Examination	# Non-data;PP Spacer;6EDKSPPI	NFLC.	ָּר	ಪ	긥	ARXLOKSP	FWMDDKLTSP/PT?	NFCL	FWLGDKSP				נו	GELTMP/FWMDDKSP	CL / GEL 15P	ARSMSP/PT?	ARLCIM	FWLGDKSP	ARLCIM	נ
Yibaiy			•	m									•		7					_
ů E			1.8	1.43	1.7(0.84		0.87		.1 -		i	0.87
Cor Rt (mil/yr)			0.0211	0.0239	6.019					,			0.0613		0.0636					0.0536
ilm st mg/			3.4	4	3.1								11.3		11.6					11.6
Wit Loss Film Wit Cor Ri .mg· (mg/ (mil/yr			Š	3.4	2.7								В.7		٥					o•
Flux (4E-5)		0.033	0.041	0.041	0.041	0.019	0.019	0.033	0.019	0.04	0.04	0.04	0.04	0.027	0.04	0.027	0.037	0.027	0.037	0.04
11 me	0.0	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5
1 ae		2, 99	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499
Temp (C)	95	95	95	95	95	95	95	95	95	95	95	95	95	95	95	95	95	95	9.5	95
Phase Temp		ເລ	ιa	Ö	Ģ	9	ഥ	ß	9		-	-	٠.,	1	_1			_		ب
Runs		-	1								_	_		_	_	_	-4	-		
Cage No.	*	w	3	3	3	>	>	ш	>	> -	>	>	>	×	>	×	K:	×	E.	>
Vesse!	n	4	4	•	4	4	4	4	4	•	4	4	4	4	4	~	4	**	4	•
CDA	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101	101
Sier No.	30	4	99	5	7 4	42	7.0	17	27	67	6 1	4	63	24	55	32	4	18	0 \$	58
lype	ú	u	3	3	2	cata	ac	٠.	o co	3	3	3	38	œ	3	œ	· (4	œ	Ų	3
Allov			-	***					•				-			-				

TABLE B-2A

SPECIMENS SHOWING EVIDENCE OF PITS (Crevice and Interface Specimens Not Included)

Visual Examination			FWFL/FWXLLC/ARSMPT	ARCUEO/FWPT	AHEULG/15PI	אאגורותות מו	OFFITTI/ARTOLE	GEDKSFURF	BEFL/RRFI/ROFI	FWLC/6ESMP1	BEFM/GESMSPP1/FWLC		DESTANTIVENCE/BEFF	GEFLPILC/ARCUCO	פבורוור	GEFLPILC/ARCUCO	ISLE/GESMPT	GECULC/FWPT	ARLC/6EVSPTSP	GELGPT	GEFLP:/ROPTWD	GEVSSPPT	GEXLLTSPVSDKP T	GEVSSPPT/ARDKF#	GELTMDSPSMPT/AOPTWD	GESMLCPT/ARCUCO	ARCUCO/SESHPI/ROPI	GESMSPPT?	BESMP 1LCSP7/ARCUCD	ARCUCO/6ESM/DPPTSP?	GESMPTLC/ARCUCO	GESMSPP1/FMLGP1	ISPI/CUCO/CL	ARCUCO/GEVSSPP1	TRICICULT/FERFOSF17	1000 4000		GEVUTI FRANCE	DEVSF1/ARCUCE	したいのうまし		FWSSPPI/ROCUCO	ARSHP1/GED*F#	CL / FASTSPFT	ROPT/GEDKFM/19SP	CL/F#VSDKSPFT
Anair			C)		r				a								•	Œ							A D		a														í	a		•	c)	
0/1					1.1/	0 6	1 0	70.1	•	m 4	7.0	10.1		0.4.0	70.0	U. 40	24.0	0.75	0.47		1.07	0.62		0.62		1.41			0.84	0.78	0.80			•) d	,			0	7 .	1.07		,	2.21		BO - 1
Cor Rt	(m 1 1 / yr)			6 213	0.071	0 142	0.0356	מים מים	F600 0	0.0803	9050.0	0.000	7 0	0.00	30.0	0.000	0.110)	0.071	0.1311		C.0.0	0.068		0.074		0.0/3			0.1572	0.1621	0.1659			•	0.0412				420 0	070.0	•			0.017	. 0 4 1 0	D + 10 .0
	. i			0	. 4	. 4		1	0	o ~	, o	· a				4.71	7.07	1.5.7	2.0	•	7.4	n •	•	ر. و	c	4.4			7.67	1.77	9.17			0 7 1	77 1	1			7		7 - 7		•	* .	r-	, ,
) 			7 9	-	7.2	2.3)	C-	1 4			or or	່ແ	9 01	300	0 7 1	7.7	F - 7 F	4	י ר י	7.5	,	7.7	0			9	* · · · ·	20.3	17			-	0						•		-	7.7	.,	ר
F) ux	1	-		. 4	9.1	1.6	0.1	P7	4	0.041	0.0	0.041	2	2	2	1 4			-	7 -	. 4		9.4	r -	0.4		6.0	70.0	5 0	50.0	40.0°	170.5	7 , 6	9	6	2.6	2.6	7.4	4.	4	,		7 -	7	, ,	:
Time		1.0	0 1	1.0	1.2	2.1	2.1	3, 2	3.2	3.5	3,5	3.5	7	4	4		7 0	o o	i a		-				7.0	7	7 15	, ,	י ר	, h	י ר י ר	· .	7	7.0	ο· Θ	65	6.9	13.1	1.2	1.7	1.7		, ,	• c		,
Time (h)		720	720	720	966	1,517	1,517	2,286	2,280	2,499	2,499	2,499	2.927	2,927	2.927	3.672	5.016	A 408	408	9.408	720	720	720	720	7.7BG	2 280	7 499	7 499	2 499	7 400	7,47	1 672	5.016	5.016	904'9	408	6,408	9.40B	B36	836	836	515	515	2,334	4.007	
Temp (C)		9.5	9.5	95	95	95	9.5	95	95	95	9.5	95	95	95	95	95	9.5	95	9.5	9.5	95	. 0	י ני	. G	. 6	. 6	5.	٠ د د	. 0	, o	י סי	2	9.5	95	95	95	93	5	150	150	150	150	150	150	150	
Phase		G	æ	២	Ŋ	ឲ	co	ന	ເອ	មោ	O	ល	9	ß	9	9	ശ	œ	G	G	_	۰		۱			ا	۰ ــ	J	٠ _		٠ ـــ	ر ــ ا	1		_			ല	ß	ഥ	(2	ഥ	G	دا	
Runs				_	m	.,	•	C.1	۲.,	_			2	(4	7		۳	34	34	1234													1	•~		_		1234								
Eage No.		u.	LL.		S	s	ر. س	LL.	٠.	38	3	3	55	55	55	7	7				1	4		A	1 2	A 2	-	۲ ۲		, ,	, ,	đ			I 34				۵.	 a.			m N		4.5	
Vessel (₫								-		-		-	-	-	*	47	•	4	4		-													
CDA		613	613	613	613	613	615	613	613	613	613	613	613	013	613	613	613	613	613	613	613	613	613	613	613	613	613	613	613	613	613	613	613	613	613	613	613	613	£13	613	613	613	613	613	F7 40	
8 US		7	12	20	cu in i	5	-	-	20	41 (42	9	ģ	5	9	49	31	57	38	œ	21	49	16	4	30	23	۲3 ۲3	4	09	4 5	3.4	23	22	32	7 4	4 7	36	C.		77	~	32	54	r ·	()·	
Type		CE)	CC(3	3	* :	2 (an :	3	38	3	3.	X	3	2	×	3	æ	æ	CC)	3	æ	3	Œ	3	æ	œ	38	3	æ	æ	Œ	œ	38	3 (י נו	י מב	I)	æ	x	ដា	Œ	: 3	æ	4	
A:10v	ı	N	[H	M E	`1 (4 0	4 F	• •	4 (4	*1 1	. J (પ (na f	7	۲۷	7	7	C4	M	7	7	7	C1	L4	7	2	N	7	M	ru	ra.	r-4	r4 i	14 (1 (۱ ۱۰	., ,	e (4	(4)	(s)	ru	:4	14	

TABLE B-ZA (Cont'd)

Visual Eramination	CL/FOFIWD/ARDAFH/FWPT CL/ISSMSPPI GEVSPI/GEFM GEVSPI/AFDFFM RDF/CLNF GEDKFH/GEVSPISP ARDAFH/GESPPISP ARDAFH/ARVSSPPI GEVSPIDKSP/ARDKFM	
ni c T		
0 / E	0.98 1.12 1.67 0.94 0.97	:
Cor At	0.015 0.015 0.015 0.0089 0.0116	
HE ECT	אין	
Mat 2008 tile bt Cor Rt (80) (80)	W W W W W W W W W W W W W W W W W W W	0.1
Time Fig. 41 Gon, (AE-B)	2.1 2.1 2.1 2.0 2.0 2.0 2.0 2.0 2.0 2.0 2.0 2.0 2.0	
0 c c	0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0	10.
₩ .	4, 416 4, 416 4, 416 111, 593	
Phase Temp	150 150 150 150 150 150	150
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on Ca Ca Ca	d d in air i a	1 12345
00 0 01 03 10 2€		Ï
Vessel No.	иниииии	6.1
400	6 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1 1	613
Ser No.	4 N C4	7
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o c c	HEREAL CALLA CA	2

TABLE B-2B

SPECIMENS SHOWING LOCAL CORROSION (Crevice and Interface Specimens Not Included)

	Visual Examination	FWFL/FWXLLC/ARSMPT	ARPILCCUCO	GEFM/GESMLC	GEDKSPP1/ARLGLC	FWLC/6ESMP1	GEFM/GESMSPPI/FWLC	ARLCPTSP	GESMSPPT/FWLC/GEFM	GEFLPILC/ARCUCO	GEFLPTLC	GEF1PTLC/ARCUCO	ISLC/6ESMP1	GECULC/FWP1	PP Spacer; GEALFLLC	PP Spacer; SEXLFLLC	ARLE/GEVSPISP	GERU/FWXLLC	GESMLCP1/ARCUCO	GESMPTLCSP?/ARCUCO	GESMPTLC/ARCUCO	FWCUCOLC/FWARVSPT?	1SLC/ARVSPT	GESMPI/ISLC	ISLC/GELIFM/GESMLISP	GEDKF#/FWLCLTSP	GEDKF#/ARCULC	GEDKFM/ARCULC
	Analy	Q A												₫				Œ										
	Ð/E		0.7B	0.93	0.81	1.45	0.95	1.01	0.97	0.90	0.82	0.95	0.93	0.75			0.47	1.18	1.41	0.84	0.80	0.40	0.68		0.14	0.76	0.78	0.74
	Cor Rt (mil/yr)		0.071	0.0482	0.042	0.0803	0.0506	0.0569	0.0428	6.05	0.057	0.056	0.1107	0.071			9.1311	0.0951	0.073	0.1572	0.1659	0.044	0.0612		6600-0	0.0071	0.01	0.0075
			4.1	4.7	4.2	10.8	8.1	6	6.9	11.4	11.1	12.4	26.2	23.9			65.3	54.5	ъ. ъ	25.9	27.6	17.9	27.3		13	₩.	7	5.1
	Wt Loss Film Wt (mg) (mg)		5.1	3.7	3.2	9.2	4.9	7.2	5.4	o . 69	Б	9.8	20.6	17.9			42.5	44.9	8.4	19.9	21	1.1	19.9		4.7	3.6	5.3	3.8
•	F1 ux (XE-5)	1.3	1.6	1.6	1.6	₩	0.041	0.041	0.041	2	7	2	4	4	4.6	4.6	d	4	6.4	0.04	0.04	4.9	4.9	2.6	4.9	0.42	0.42	0.42
	11 me (mon)	1.0	1.2	2.1	2.1	3.2	3.5	3.5	3.5	4	4.1	4. 1	5.1	7.0	7.4	7.4	8.9	13.1	3.2	3.5	3.5	7.0	8.9	8.9	13.1	14.0	14.0	14.0
	71 me (h)	720	860	1,517	1,517	2,28∪	2,499	2,499	2,499	2,927	2,927	2,927	3,672	5,016	5,304	5,304	6,40B	9,408	2,280	2,499	2,499	5,016	6,408	6,408	9,408	10,078	10,07B	10,078
	Tesp (C)	95	95	95	45	95	95	95	9.5	٠. د	95	9.5	95	95	95	9.5	95	95	95	95	95	95	9.5	95	95	150	150	150
	Phase	Ŋ	ല	G	æ	ເລ	Ġ	Ġ	Ċ	භ	æ	ى	ເມ	IJ	æ	œ.	ŋ	ເລ	ب.		_1	ب	_	ب		9	ເລ	ם
	Runs	_	~	4	4	2	_	-	_	1.2	13	12	4	23	1234	1234	34	1234	2		-	23	34	34	1234	1234	1234	1234
	Cage No.	u	m	Ŋ	Ŋ	_	38	3	*	55	S S	52	۱	_	-	1		_	-	>-	;= -			Œ	7	99	99	99
	Vessel No.	***	M	₽	~	-	4	♥	4	~	m	m		-	m	n	-1			**	•				-	2	2	2
	CDA	613	613	613	613	613	613	613	613	613	613	613	613	617	613	613	613	613	513	613	613	613	613	613	613	613	613	613
	Ser No.	7	S	61	39	30	4	42	29	29	65	99	40	31	4 1	Ů	5,	13	9	4 4	4.5	32	7.4	24	16	24	4	'n
	er er	ω	: #	3	×	*	3	38	3	:#	×	3	-	3	œ	Œ	3	*	3	3	35	3	**	æ	ж	38	:	3
	Alloy	2	2	2	2	2	2	r4	2	17	2	2	7	2	2	2	(~)	2	N	7	7	7	7	7	2	2	2	7

TABLE B-2C

SPECIMENS HAVING A CLEAR SURFACE OR NO CORROSION FEATURES (Crevice and Interface Specimens Not Included)

Visual Examination	15P1/1360/Cz	CL/FMSF	CL/FWSMSPP:	GEDFFM/CL	GEDNFM/CL	CL/FWL1SP	CL/FWVSDkSPP1	CL/ROPIWD/ARDNEW/FWP1	CL/ISSMSPP1	GEDKFM/CL	ROP1/CLNF	LTCUCO/CL	GEDKFM/FWLCL1SP
Arania Y					,								
0/4		1.51	3.27			0.9B	1.08		0.98	0.94			0.76
ior At mil/yr			9.017			0.0153	0.0148		0.0153	0.012			0.0071
1 (6 m)		1.4	4.1			5.9	7.5		3.9	3.3			4.8
Witces Film Witcer At (mg) (mg)/yr/		1.2	1.3			3.1	М		3.1	2.6			3.6
Flux (XE-5)	2.6	0.31	9.31	3.1	3.1	1.5	1.5	1.2	1.5	1.5	1.2	1.2	0.42
T = = (= 0 =)	5.1	2.1	7.1	3.2	3.2	5.6	5.6	5.6	5.6	6.1	6.1	11.7	14.0
ine (n)	3,672	1,515	1.515	2,334	2,534	4,007	4.007	4,007	4,007	4.416	4,416	8,423	10,078
lemp (C)	95	150	150	150	150	150	150	150	150	150	150	150	150
Phase Temp		ഥ	פי	מו	ø	G	g	Ġ	ß	Ф	Ġ	ى	9
म एव स	•#	2	5	14	7	54.5	ان ا	٠. د	4 7	~ 1	n	345	1234
Ca⊊e No.	Œ	ш	ш	-,	~	a.	a	~	۵.	a.	~	~	99
S S S S S S S S S S S S S S S S S S S	`	7	2	2	2	2	2	7	€.4	2	(4	2	2
CLA	613	613	613	613	613	613	613	613	613	613	613	613	613
Ser No.	52	53	5.4	16	9	5.5	4	4.	5.1	7	4	10	24
e d	Œ	3	3	æ	00	*	3	(12)	3	3	600	œ	3
<u>a</u>	L1	2	2	7	i 64	2	۲-	1 64	104	(*-	f (4	t (N	01

TABLE 8-20

SPECIMENS SHOWING FILIFORM-LIKE LOCAL CORROSION (Crevice and Interface Specimens Not Included)

Visual Examination	FWEL/FWXLLC/ARSHP1 GEFL/ARP1/ROP1 GEFLPTC/ARCUCO GEFLPTC/ARCUCO GEFLPTC/ARCUCO GEFLPTC/ARCUCO PROUCO/FWFL FP SpacerickxLFLC PP SpacerickxLFLC	GEFLPT/ROPTWD GEVSPI/FWXLFL
Anair	Q Q	
0 / E	0.90 0.82 0.95	1.07
Cor At #11/yr)	0.06 0.057 0.066	0.015
1 (o e)	4	4.2
Wit Loss film wit Cor Rt (mg) (me) (mill/yr)	ထား တာ ဇာ ဟာ ထာ	5.4
line Flux (mon) (XE-5)		1.3
1 .me	3.2 3.2 4.1 5.1 5.1 6.9 6.9	8.9 8.9
1 a a a a a a a a a a a a a a a a a a a	720 2,927 2,927 2,927 3,672 5,304 5,304	9,40B
lemp (C)	\$ \$ \$ \$ \$ \$ \$ \$ \$ \$ \$ \$ \$ \$ \$ \$ \$ \$ \$	45 24
Phase Temp 11	9900000000	ר פ
Runs	1	34
Caye No.		- ⊄
Vessel Cage No. No.		
CDA	613 613 613 613 613 613	613
Ser No.	20 65 65 37 20 20 20 20 28	36
l ype) (D
Alioy lype Ser No.	иизичиини	1 74

TABLE B-2E

LOW FLUX SPECIMENS

Visuai Eramination	(Mildy) (SIVI) Labers Ad	TOUR SPACE STATE OF THE PERSON AND T	ARDKFM/GEL 10KSP	661.61.91	SEDAFA	GEFM/GESMSPPI/FWLC	GEL CP 7	GEDKFM/ARVSSP	ARLCPISP	GESMSPP1/FWLL-GEFM				GESMSPPT?	ARLE INVENER	ARPILCIM/CUFM	CUCOSP40/ARCUCO/GEVSSP	GESMPT.CSP2/ARCUCO	ARCUCO/GESM/DPPTSP?	GESMPTLC/ARCUCO	1d5 (M371/3d5W535)
An																					
0 / 12		٠				0.95			1.01	16.0								0.84	0.78	0.80	
Cor Rt Imiliyri						9.0296			0.0564	6.0428								0.1572	0.1621	0.1659	
Film wit Cor Ri imgf (mil/yr)						. . .			6	9.9								25.9	27.1	27.6	
						4.9			7.2	5.4								19.9	20.5	21	
linare Flux Wit Loss (non) (XE-3) (mg)			0.019	6.033	6.019	0.041	0.033	0.019	0.041	0.041	0.04	0.04	0.04	0.027	0.037	0.037	0.027	0.04	0.04	0.04	0.027
1 a a c)	0.0	0.0	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5	3.5
T 3 B C			2,499	2,499	2,494	2,449	2,499	5.400	5,499	2,499	2,499	2.499	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499	2,499
Temp (C)			9.5	45	45	95	95	95	9.5	95	95	9.5	95	9.5	95	9.5	95	95	9.5	95	6.5
Phase Temp (C)			9	و	ŋ	ග	9	9	9	g		_	-	_	ب		ب	ر		ر	_
หือกร			_	_		_	_		_	_	-	_			_		_			_	
N.C.		=	>	ų	>	3	u)	>	3	3	-	۔	-	×	r	E	×	-	>	>	×
vessel Cage No. No.	m	m	∢	•	4	4	4	4	4	₹	4	4	4	4	4	•	4	4	4	•	4
0.00	513	613	613	614	613	613	614	613	613	è I 3	613	613	613	613	4 9	4 19	613	613	613	613	613
Ser No.	4	39	35	3.5	30	9	36	20	4.5	62	38	63	9	33	25	27	27	4 4	9	45	74
1456	œ	a	20	ں	œ	3	Ļ	œ	3	3	3	3	3	æ	u	Ü	æ	3	3	3	æ
Alloy	r a	2	C4	*4	2	ř.	N	f-4	~•	r-1	£ 4	. 4	7	~	(1	7	C-4	۲4	7	r4	r u

TABLE B-3A

(Crevice and Interface Specimens Not Included)

Visual Examination		GESTRO / ISLU		GE SMP :	ARSHP17CL	AKOSP (/ DKFR		ISLELL/ARMUPI	מטויים אוארוי או			# Non-data; PP Spacer; GELCP1/ARCUCG/Ru	PP Spacer; BELCPT/ARCUCO/RU		LL / 188371 / #U	CL/15VSP1	GEVSL TSP / FWPT / ARRU	GEVSLISP/FWPI/ARRU	GEVSLISP/FWPT/ARRU	GEMP/GEMDPTDD/FWLC	GEMP/GEP100/FWLC	GELTSP/FWSMPT	GEDKFM/GEVSLTSPPT	GEDKFM/GEvslisPPT	GESPMPLCPI	GESPLCP1	GEXLSPLCCUPT	GEVSP11 CDD/GEDKFM	GELCPT
Analy						ć) (2												₫	Œ								
0 / £													94		7.60	0.43	0.76	78.0	1.13	69.0	1.08	1.08	0.87	0.84	0.84	0.81	0.70	0.82	0.85
Cor Rt (mil/yr)													740		111.0	0.148	0.000	9.0.0	0.0876	0.224	0.54	0.036	0.0586	0.0562	0.082	0.066	0.057	0.055	0.031
													76.7	12.	. 0		7 - 5 1	0 :	14.5	87.6	8 /	11.1	10.7	10.4	5.2	4.2	12.8	10.2	5.7
Wt Loss film Wt (mg) (mg)													19.3	9.7	. ~	•		* ·	11.7	64.5	9.0/	6	8.3	œ	4	3.2	4.6	7.8	4.4
flux (XE-5)	-		-						0.016	1.2		4		4.6	4	. 4	4		•	.	d .	.5	0.04	0.04	1.6	1.6	7	0.04	0.041
7 t 9 (C C E)	13.1	16.1	-	4	0	1.0	3.2	7.0		5.6			1.0	1.2	1 2	1.1		, ,	7.5) ·	7.5	9	3,5	3.5	1.2	1.2	4.1	3.5	3.5
[1me	9.408	11.593	720	10.078	720	720	2,280	5.016	2.499	4,007		5.304	720	836	83.6	7, 114	7, 334	2001	+ C C ' 4	מוחיר	7,200	4,416	2,499	2,499	960	860	2,927	2,499	5,499
lesp (C)	9.5	150	9.5	150	95	9.5	95	9.5	95	150		95	9.5	15.	150	150	150	2	000	י ט ט	7 9	00.	ה ה	٠ د ا	9.5	٠ د	95	5	95
Phase Temp (C)	9	9	9	9		9	9		و	(J)		(1	(2)	(3	15	"		. ,,		ט ני		•				_			
Runs	234	2345		234				3		5 4		1234								, ,			.	_ (. د			,	Ð
Cage No.	u.	88	F.	88	•	-	F 2	£ 3	>	4	*	_	-	 α	 a	с,	P 2	٥		1 -) - L)		-	n 1	Λ : Π :		 - :	
Vessel C	-	2		7	-	-	-		4	(4	٣	3	-	(4	17	2	2	2	. –	•	٠.	7 •		* ~	n r	٠ ٠	,, ,	٠,	•
CDA	715	715	715	715	715	715	715	715	7.15	51,	715	715	512	715	715	715	715	7.15	715	715	715		517	7	7.5	71.	61/	21,5	71
Ser No.	-	2	~	S	10	1.4	1.7	23	52	26	27	39	_	1	<u>0</u> -	2 b	27	82	52	33	2	, 4	0	·	י ר		י י	0 4	2
Type	20	20	Œ		20	œ	œ	æ	а	a	œ	ഥ		3	3	3	3	3	3	*	3	3	: 3	: 3	: 3	. 3	. 3	: 3	
Alloy Type	M	m	M	~	~	~	m	m	٢	~,	m	M	·^	~	~ 1	~	'n	٢٦	m	2	~	· ~	, ~	, ~	·	۰ ۳	n ~	· ~	1

TABLE B-3B

SPECIMENS SHOWING LOCAL CORROSION (Crevice and Interface Specimens Not Included)

Hnaly Visual Examination		GESHPT/15LC	D ARLISP/15LE	GEDKFM:FWLC/ARRU	D GESMLISP/ISLC		GEARRU, FWLC	D ISLGLC/ARMOPI	GELCRU	GEXLRUSP/FWLC/DKFM	. סבירות המינים	bealth * Mon-data DD Coarda (F) (PI/QR(UF))/RU		PP Space: 6FLCPT/ARCUCO/RU		65 61 7 0KD	GEL CARRON AMPTABBIL		ומרר/ רר סבו טופיונטאווע				_	A GEOKFA/FULL	_	GEKLLCLISPUKFA	THE TOTAL CONTROL TO THE TOTAL	FWXLLL()bEUXFB/bEMPLIDF	AMLL/ DEALL IOTLILL AMLL / DEALL / DEALL IOTLILL AMLL / DEALL / DEAL	FWXLLL/GEFMXSGEDUSF	GESTATILET I		6EL15P/15*LLL	GEXLSPLCLUP	GEFLLGLC	GEVSPTLCDD/GEDKF#	GERUMPLC	GERUMPLC	GELCPT	GEXLCULTSP	GEXLLISPLLLU	GEMPLE/15%LLL
I																											_						_	_	_	٠.		٥.		.		
۵/ ی																	•	6 6	B	9.0	0.51	0.69	0.70	1.14	1.08	0.87	0.83	0.83	9. 9.4	1.03	0.84	0.81	0.86	0.70	1.06	0.82	0.64	0.62	0.85	0.68	0.76	2.37
רסי אֹנ	(611.vr)																	/ * . 0 .	0.083	0.1033	0.0049	0.224	0.09	0.14	0.54	0.0684	0.0162	0.0803	0.062	0.0735	0.082	0.066	6.074	0.057	0.0267	0.055	0.2078	0.3322	0.031	0.058	0.059	0.0991
) (6E)																	1.97	4.5	76.4	9.6	87.6	35	21.6	83	7.6	m	0		33.2	5.5	4.2	æ.	12.8	4.7	10.2	105.3	97.1	5.7	13.3	13.2	22.9
s	. be ;																(14.5	3.4	34·B	5.3	64.3	7.52	17.7	70.6	5.9	2.3	6.9	M	26.7	4	3.2	6 .8	4.4	3.8	7.8	75.5	1.69	4.4	4.7	6.6	20.7
	(36-5)	-	, ,	2.5	3.1	2.6	2.6	1.3	1.3	2.6	4.6	3.1			•	۲.۷	2.6	-	4.0	∢	4.9	•	4.9	6.4	4	1.6	0.041	1.6	1.6	4.9	1.6	1.6	1.6	7	0.041	0.04	4	4	0.041	7	7	4.9
	(nom;		: -			3.2	7.0	3.2	8.9	8.9	4.1	3.0	0.0		. ,	00	60	0.1	1.0	13.1	13.1	7.0	7.0	3.2	3.2	2.1	3.5	2.1	1.2	8.9	1.2	1.2	2.1	4.	3.5	3.5	8.9	5.1	3.5	7.7	4.1	5.1
a.	Ē	9097	720	9 4 P	7 114	2.280	5.016	2, 280	6,408	6,408	2,927	2,183		3,672	3,504	6,4 08	6,408	120	720	9,408	9,408	5,016	5,016	2,280	2,280	1,517	2,499	1,517	999	6,408	098	986	1,517	2,927	2,499	2,499	6,408	3,672	2,499	2,927	2,927	3,672
jeap	(C)	á					, 0 ,	9.5	9.5	9.5	95	95		95	5	95	95	95	95	95	9.5	95	9.5	95	95	9.5	95	9.5	95	9.5	95	95	9.5	9.5	9.5	9.5	9.5	9.5	20	95	95	95
Phase		Ų	۰.	.ı –	ی ر	·		ی ر	ധ	ب.	ŋ	9		១	co	(J		ى	٠.	9	ب.	9	ر		9	وی	ເອ	(9	ıρ	ب.	s	9	ය	' 5	(9	·	(1)	ور	و.	9	נט	1
Runs		1274		1734		1 C	23	2 2	34	34	12	2		4	1234	34	34		_	1234	1234	23	23	2	2	4	-	4	2	34	m	2	4	27	·		**	4		1.2	<u>e</u>	47
Cage	NO.	u	٠ <	z d	·	, 4	: c	ш.	ı	₫	p	œ	=	اعنا	_	u.	Œ	_	-	د	-				ب	s	*	S	s	-	S	S	ı,n	SS	3	*	٠	ا	*	55	5.5	
ā	No.					٠ -					2		2	-	m		-	-	-		-	-	_	-	-	n	₩.	~	₩	-	~	3	m	~	đ	4	-		4	М	2	
CDA		7:5	717	7.17	512	1	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	715	7.15	715	715	7.15	7 1 2	715	715	715	715	715	715	715
	S.	-	- u	. d	7 0	0 0	2.0		56	3.1	32	34	3.7	\$8 8	34	0	42	-	6	20	22	29	30	32	33	38	43	4	4	4 7	50	51	52	5.3	י גרו	. 49	· (1)	, v	6.9	62	5.5	0
Type		۵		ם מ	aca	a C	ı cc	1 00	o co	Ľ.	G	æ	m	œ	æ	œ	œ	3	¥	3	3	3	*	*	3	3	3	3	3	3	3	3	¥	3	3	3		3	: :	3	3	3
Alloy Type			^1 p.	γ ~) r	7 ~	, ~	, ~	· ~	, pr.	M	m	~	·)	~ 1	m	~	~	m	~	~	~	۳	~	~	17	~	₽	~	~	~	~	~	•	. 1	~1		. ~	~1	۳٦	2	M

TABLE B-3C

SPECIMENS HAVING A CLEAR SURFACE OR NO CORROSION FEATURES (Crevice and Interface Specimens Not Included)

Visual tramination	ARSMP1/CL CL/ARLISP CLNF CLNF CLNF CL/GEDACUFM ARDKFM/CL CL/FWMDLTSP GEDKFM/CL CL/FWMDLTSP GEDKFM/CL CL/FWMDLTSP GEDKFM/CL CL/GERU CL/GERU CL/GERU CL/GERU CL/GERU CL/GERU
Analy	a
0/#	0.97 1.20 0.78 1.03 0.93 0.99 1.11
t Eor Rt (mil/yr)	0.048 0.211 0.083 0.041 0.048 0.0897 0.0897
130 E	34 4.2 1.2.1 1.2.2 2.2.0 9.4.4.0 8.3
Wt Luss film Wt (mg) mg!	27.6 10 23.4 23.3 77.7 18 7.7.7
lime Flux (mon) (XE-5)	2.1 3.1 3.1 3.1 3.1 3.1 3.1 1.2 0.027 0.019 0.027 0.027 0.42 4.6 4.6 0.42 0.42
Time (mon)	1.2.0 1.2.2.0 1.2.2.0 1.2.2.0 1.2.0 1.2.0 1.2.0 1.2.0 1.2.0 1.2.0 1.3.0 1.0
Time (n)	10.078 836 836 2,334 2,499 2,499 1,515 2,499 10,078 836 10,078 1,515
Temp (C)	150 150 150 150 95 95 95 150 95 150 150 150 150
Pnase Temp (C)	
Runs	1 1 2 3 4 4 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5 5
Cage No.	80 - 0 - 0 - 0 - 0 - 0 - 0 - 0 - 0 - 0 -
Vessel i	иииии д аидидаии
CDA	715 715 715 715 715 715 715 715 715 715
Ser No.	2 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7 7
ارا درا م	
A 1 0 4	м м м м м м м м м м м м м м м м м м м

TABLE 8-30

SPECIMENS SHOWING FILIFORM-LIKE LOCAL CORROSION

	Analy Visual Evamination	FWXLLi/SEFLLC GEFLLG:C
	Analy	
(pa	0.4	0.83
(Crevice and Interface Specimens Not Included)	Vessel Cage Runs Phase lemp Time Flux Mt Loss Film Wt Lor Ht No. No. (E) (h) (mon) (AE-5) (mg) (milyr)	4 1 6 95 2,499 3.5 0.041 2.5 3 0.0162 0.83
Not	ilm wit (mg)	Α. V. L
cimens	lemp Inne Inne Flu. Witcossijn wit Corfit (C) (h) imonitaE-5) (mg. (mg) (mil'yr)	en at
e Spec	Flu: W (xE-5)	0.041
erfac	1.0001	χ. 2. ε
nd Int	11 me	2,499
e ar	једр (С)	و. دي و
revi	Prang	9
; <u> </u>	8. G 0	
	age No.	32 3
	Vessel Cage No. No.	स र
	CBA	43 715
	Ser Re.	44 q 60 q
	v ivpe Ser Ne.	* 3
	>	mr

ABLE B-3E

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LOW FLUX SPECIMENS

Visual Examination	13 / E	1851713	G71 (G234 / 13	ARLICO	* Non-data:PP Spacer: GELEPI/AREUCG-RU	CLNF	CLNF	ISLEVEINE	ET LASO BE	ARVSPIIM	FWXLLCIM/GEDNFM		GEDKFM/GEVSLTSPFT		FWXLLC/GEFLLC	GEDKFM/GEVSLTSPPT	GEFLLGLC	GEVSP1LCDD/GEDKFM	GELCPT	
Analv																				
9/4													0.87		0.83	0.84	1.06	0.83	0.85	
Cor Rt (mil/yr)													0.0586		0.0162	0.0562	0.0267	0.055	0.031	
Film Wt Cor Rt (mg) (mil/yr)													10.7		۳,	10.4	4.7	10.2	5.7	
Wt Loss Film Wt (mg) (mg) (2.3	9	3.8	7.8	7.4	
Fier XE · Si	0.027	0.019	610.0	0.027		0.019	0.027	0.033	0.037	0.037	0.033	0.04	0.04	0.04	0.041	0.04	0.041	0.04	0.041	20.0
lime (mon)	3.5	3.5	3.5	3.5	0.0	5.5				3.5		3.5		3.5				3.5		ر د
11 me	2.499	2,499	2,494	2,499		2,499	2,499	2,499	2,499	2.499	2,499	2,499	5,499	2,499	2,499	2,499	2.499	2.499	2,499	2 499
(C)	9.	5.5	٠ ار	9.5		95	95	45	56	٠. ئ	9.5	95	9.5	9.5	95	9.5	65	9.2	95	50
Phase		ເລ	و	ند.		9	Ų.	Ç	٦.	_	وا			1	S)	-1	ග	-4	9	
Auns																				
a ð e	_	>	>	~		>		ш	T .	æ	u)		 	, I			3	-	*	- -
vessel Cage No. No.	4	u	47	-3	M	•	₹	• 7	4	4	~	t	4	4	*	4	+	4	4	4
V AGO	715	715	715	715	315	715	315	715	715	715	715	715	715	715	715	715	715	715	715	715
Ser No.	22	25	30	3.5	23		r 4	9.5	33	3.5	37	3.6	7	7	¥	4 9	55	26	0.9	5
i y p e	Œ	8	Œ	Œ	œ	æ	æ	ري	ر	ر	ب	3	3	3	3	. x	38	2	3	3
Ailoy	~,	۲,	~,	···	M	M	m	2	М	rı	M	3	n	n	М		М	n	n	•

DATE FILMED 1108/190

1)