

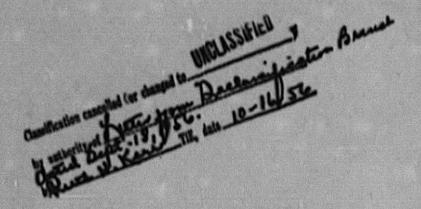
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SURVEY OF REFRACTORY URANIUM COMPOUNDS

by

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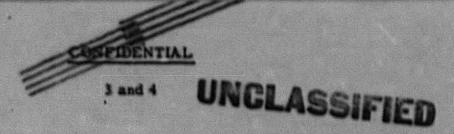
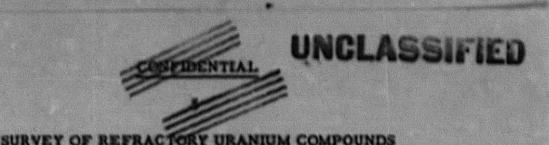


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Chemical and physical data on twenty binary uranium compounds that may prove suitable for refractory nuclear fuels were assembled. The compounds were those with aluminum, borom, carbon, iron, nickel, nitrogen, silicon, or sulfur.

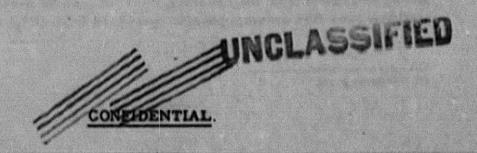
Too little is known at this time about the compounds to evaluate any of them for fuel. The program is being extended in an effort to provide the needed data.

INTRODUCTION

A need exists in our nuclear-energy program for fuel compounds that are stable at high temperatures. Considerable research and development is being done on one such compound, UO₂. However, a number of other refractory uranium compounds are receiving little attention. The first phase of a research effort to supply needed data on them is summarized in this report.

As indicated, the present interest is in compounds that are resistant to melting or decomposition at high temperatures. Also, in general, a nuclear fuel should contain a high concentration of fissionable material and should have a low absorption cross section for slow neutrons. Other properties of particular interest in this program include vapor pressure, free energy of formation, thermal conductivity, thermal expansion, Young's modulus, and corrosion resistance.

The results of a literature survey are given in this report together with data obtained to date in a concurrent laboratory effort. Further reports will be issued as particular compounds are prepared and measurements of their properties are completed.



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SELECTION OF COMPOUNDS

Twenty binary uranium compounds, with aluminum, boron, carbon, iron, nickel, nitrogen, silicon, or sulfur, appeared interesting on the basis of uranium content and refractoriness. Compounds melting much below 800 C were arbitrarily excluded. Melting points and uranium contents of the twenty compounds are shown in Table 1. Two nitrides, U2N3 and UN2, and one sulfide, US2, were excluded from Table 1 because of reported low stabilities at high temperatures (1, 2), although they had the desired high melting points and uranium contents.

Table 2 lists the absorption cross sections and comparative induced radioactivities for each of the eight combining elements in Table 1. The induced activities were computed for 1 g of combining element irradiated for 1 yr in a thermal-neutron flux of 10¹⁴ neutrons/(cm²)(sec). In each case, in computing the activity, the decay chains were continued until a stable isotope was reached. In the case of boron, calculations were made for the low-absorption natural isotope, boron-11.

For comparison with Table 2, the activity of the fission products of uranium-235 was estimated to be 1.85 x 10^{11} curies per g, 1 hr after removal from a reactor operating at a thermal flux of 10^{14} nv. Therefore, in each compound considered, the uranium will introduce a much larger amount of activity than is associated with the combining element.

URANIUM CARBIDES, NITRIDES, AND SULFIDES

Preparation

Uranium carbides, nitrides, or sulfides are prepared by a variety of reactions. Equations for these are given in Table 3.

The carbides usually are prepared by arc melting stoichiometric mixtures of the elements or by reaction of uranium oxides with carbon. The reaction of monatomic carbon and UO₂ is carried out in a graphite crucible at about 1800 C. The reaction is essentially complete when the evolution of CO markedly decreases and should be stopped at this point to avoid further pickup of carbon from the crucible. (6, 8) UC can be prepared by passing methane over fine uranium powder (prepared from UH₁) at 650-700 C. (7)

⁽¹⁾ References at end.

TABLE 1. URANIUM CONTENTS AND MELTING POINTS OF URANIUM COMPOUNDS

	Urar	ium Content		
Compound	w/o	G per Cm ³ (a)	Melting Point, C	Reference
uc	95, 19	12.97	2350-2400	(3)
U ₂ C ₃	92, 97	11.97	1775 (decomposition)	(3)
uc _z	90, 83	10.61	2450-2500	(3)
UN	94, 44	13.52	2650 ± 100	(4)
US	88, 12	9,58	>2000	(1)
v ₂ s ₃	83, 40	7.32		
U ₃ Si	96, 21	14, 99	930(b)	(5)
U3Si2	92.70	11.31	~1650	(5)
UŚi "	89.44	9.30	~1600	(5)
Alpha USi2	80. 91	7.27	~1600	(5)
Beta USi2	80. 91	7.48	~1600	(5)
USI3	73. 86	6. 02	~1500 ⋅ ,	(5)
UB ₂	91.66	11.75	>1500	(6)
UB ₄	84. 61	7.94	>1500	(6)
U ₆ Ni	96. 05	16.90	- 790(c)	(5)
UNi ₂	66. 98	9. 02	985(c)	(5)
UNI5	44, 77		1300	(5)
UAI2 .	81.52	6.64	~1590	(5)
U6Fe ·	96. 20	17.00	815(c) ·	(5)
UFez	68.00	8, 98	1235	(5)

⁽a) Based on product of X-ray density of compound and w/o uranium.

(b) Peritectoid temperature.

(c) Peritectic temperature.

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TABLE 2. ABSORPTION CROSS SECTIONS AND INDUCED RADIOACTIVITIES OF COMBINING ELEMENTS IN REFRACTORY URANIUM COMPOUNDS

Thermal-Neutron-Absorption Cross Section, millibarns per atom	Induced Activity(a), curies per g of element
3,2	0.0000282
1880	0,025
490	2, 25
130	0.2
50	0, 738(b)
4600	1,16
230	14.0
2530	3,77
	Cross Section, millibarns per atom 3, 2 1880 490 130 50 4600 230

(a) Activated for 1 yr in a thermal-neutron flux of 1014 neutrons/(cm2)(sec).

⁽b) Three seconds after removal from the neutron flux this activity will be attenuated by a factor of 10-8.

Compound	Physical Appearance	Liquid-Salid Reactions	Solid-Solid Reactions	Gas-Solid Reactions
UC	Bright hard crystals	U + C UC (2100 C)(7/a) U + C UC (1600-1700 C/B)	UO2 + 3C UC + 2CO (1750-1800 C) ⁽⁶⁾	U + C2H2 UC + C + H2(7) U + CH4 UC + 2H2(7)
U ₂ C ₃	Bright hard crystals	20 + 3C UC + UC2(2000 C)(9)	uc + uc ₂ u ₂ c ₃ (1600 c)(9)	
uc ₂	Gray metallic crystals	U + 2C UC2 ⁽¹⁰⁾	$UO_2 + 4C \longrightarrow UC_2 + 2CO (1750-1800 C)^{(8)}$ $U_3O_8 + 14C \longrightarrow 3UC_2 + 8CO (2400 C)^{(10)}$	
ÜN	Gray metallic crystals		<u>Nitrides</u>	$\begin{array}{c} \vdots \\ \text{U} + \text{N}_2 (<1000 \text{ C}) \longrightarrow \text{UN}_3(x>1); \text{UN}_3 \longrightarrow \text{UN} (1300-1600 \text{ C})^{(4)} \\ \text{UCI}_4 + \text{NH}_3 (\text{red heat}) \longrightarrow \text{UN}_3 (x>1); \text{UN}_3 \longrightarrow \text{UN} (1300-1600 \text{ C})^{(1)} \\ \text{U} + \text{NH}_3 (<1000 \text{ C}) \longrightarrow \text{UN}_3 (x>1); \text{UN}_3 \longrightarrow \text{UN} (1300-1600 \text{ C})^{(4)} \end{array}$
			Sulfides	
US	Gray metallic crystals		uos + c → us + co (1900 c) ⁽³⁾	UO ₂ + H ₂ S UOS + H ₂ O (<1200 C in graphite crucible); UOS + CS US ₂ + CO (>1200 C); US ₂ US + S (in vacuo at 1600 C) ⁽¹⁾
				2UH3 + 2H25 2US + 5H2 (500-600 C)(12)
U ₂ S ₃	Gray metallic	: U + 3US2 2U2S3 (1800 C) ⁽¹²⁾		2UH ₃ + 3H ₂ S - U ₂ S ₃ + 6H ₂ (500-600 C) ⁽¹⁰⁾

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Uranium sesquicarbide (U2C3) can be formed by heating a stoichiometric mixture (7.03 w/o carbon) at 2000 C in vacuo and then reheating at 1600 C while shaking the material about in the crucible. Samples containing up to 90 w/o U2C3 phase have been obtained in this manner. (9)

UN is prepared by reducing higher nitrides in vacuo at 1300-1650 C.

The higher nitrides can be prepared by several methods, as shown in Table
3. Chiotti⁽⁴⁾ reported sintering UN crucibles at temperatures between 2000 and 2100 C to approximately 85 per cent of theoretical density. These crucibles contained about 0.43 w/o carbon.

Brewer (12) reported that US can be prepared by reacting UH₃ with H₂S in a vacuum tube at 400-550 C. Hydrogen evolved from the reaction is removed by evacuation through a liquid-air cold trap. The reaction product is crushed and heated at 500-600 C to decompose any remaining hydride, then reheated at 1800-1900 C in a molybdenum crucible to obtain a uniform product.

Another method of making US is to form US₂ by one of the methods described by Brewer⁽¹²⁾ and to react the US₂ stoichiometrically with UH₃. The mixture of US₂ and UH₃ is heated to 400-600 C to decompose the hydride and then to 2000-2200 C to homogenize the product.

US also can be formed by reacting UO₂ and H₂S in the presence of carbon to form US₂ and subsequent reduction of US₂ to US in vacuo at 1600 C. (1) UO₂ was heated in H₂S in a graphite crucible below 1200 C and UOS was formed; the temperature was then raised above 1200 C and CS was formed by the reaction of H₂S and the graphite crucible. CS reacted with UOS to form US₂.

Crystallography

Crystallographic data for the carbides, nitrides, and sulfides are given in Table 4.

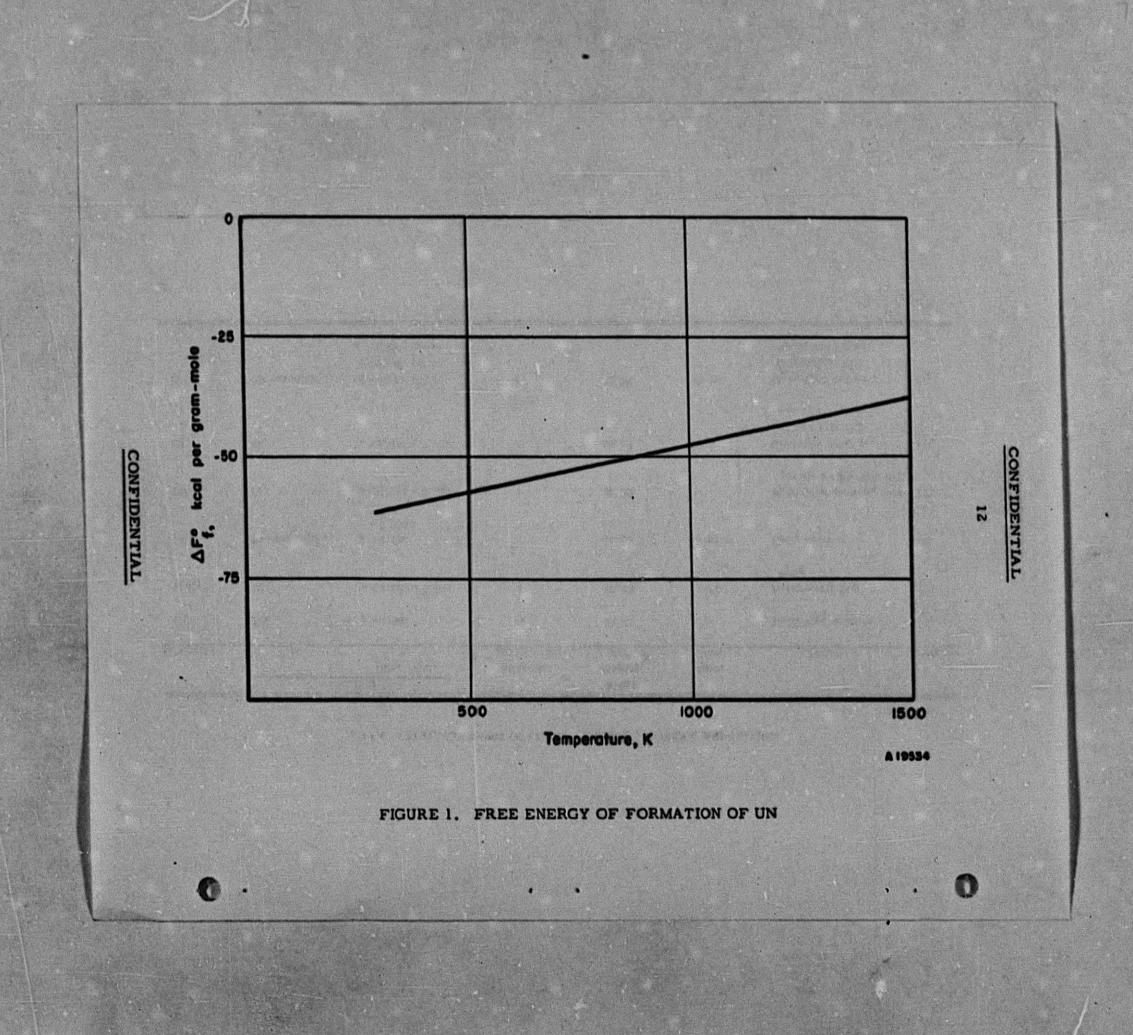
Thermochemical Aspects

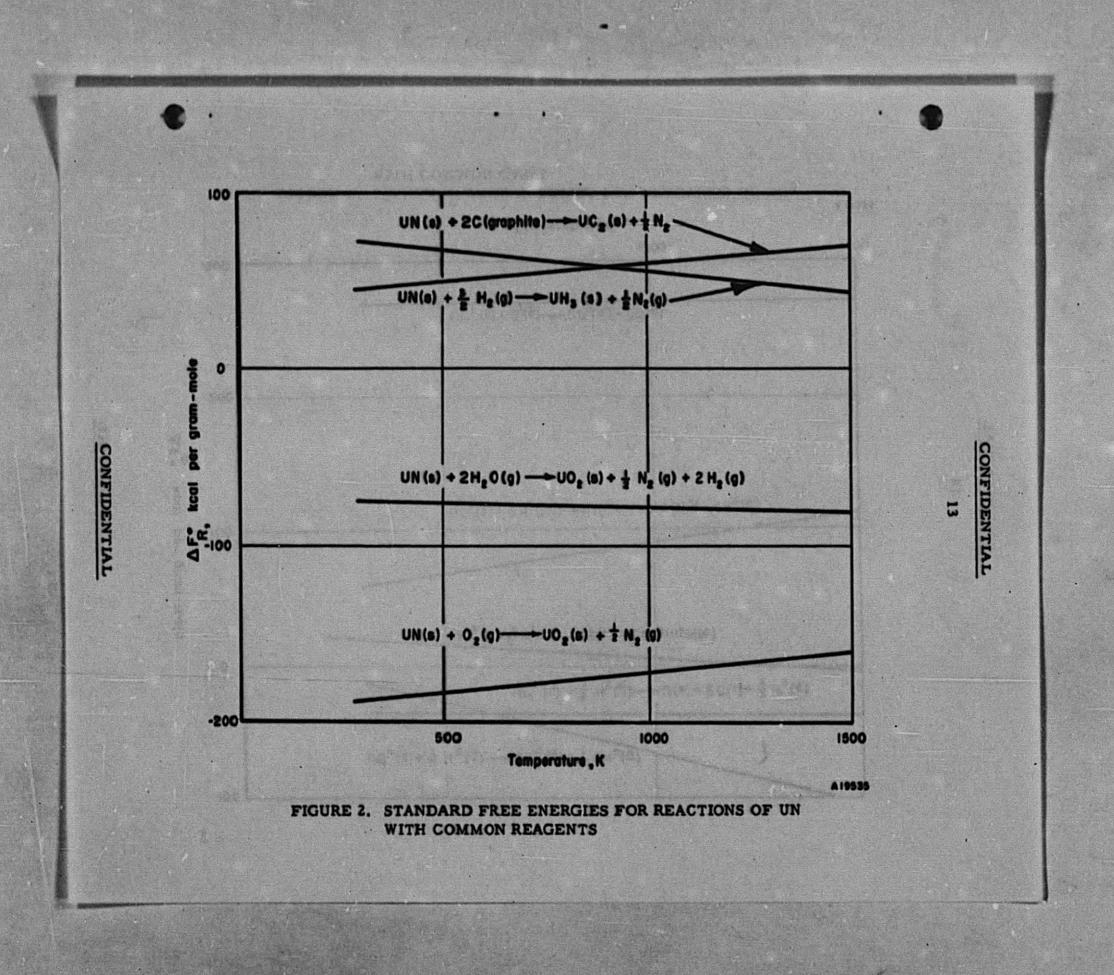
The standard free energy of formation of UC₂ at 298 K was calculated to be -42,000 calories, based on Rossini's (13) values for $\triangle H$ and $\triangle S$ at 298 K. $\triangle F_T$ up to 1500 K was computed from Kubaschewski's rule (14) by assuming $\triangle C_p = 0$.

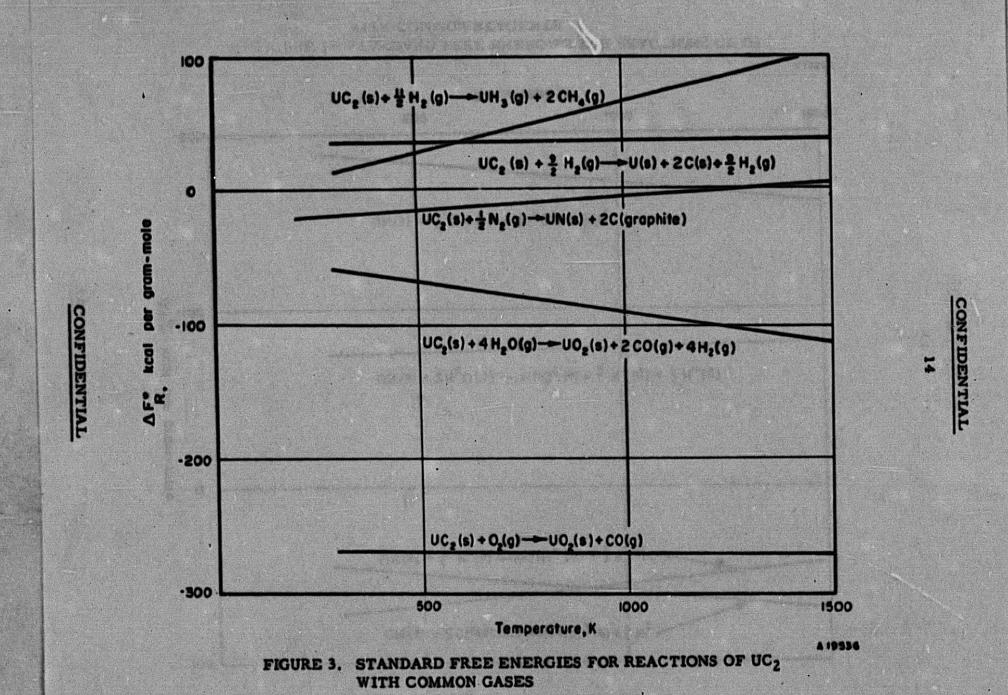
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	Uni	t Cell		X-Ray			
Compound	Туре	Dimensions,	Molecules per Unit Cell	Density, g per cm ³	Space Group	Remarks	Reference
UC	Fcc	a = 4, 961	. •	13, 63		NaCI-type structure	(10,2)
U ₂ C ₃	Bec	a = 8,088 ± 0,001		12, 88	1434	Isostructural with P ₂ C ₃	(9)
UC ₂	Bc tetragonal	a = 3,524 c = 5,999	2 .	11.68	. 14/mmm .	CaC ₂ structure	(10, 2)
UN .	Fcc	a = 4,880 ± 0.001	•	14, 32	(· · · ·	NaCl-type structure, com- pletely soluble with UC	(2)
US	Bcc	a = 5.484 ± 0.002	•	10, 87	••••••	NaCI-type structure, soluble with ThS and CeS	(I)
U ₂ S ₃	Orthorhombic	a = 10,41 ± 0,02 b = 10,65 ± 0,02 c = 3,89 ± 0,01	•	8,78	Pbnm	Sb ₂ S ₃ -type structure, isomorphous with Np ₂ S ₃ and Th ₂ S ₃	(1)

TABLE 4. CRYSTALLOGRAPHY OF URANIUM CARBIDES, NITRIDES, AND SULFIDES







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Standard free energy of formation values for UN were calculated up to 1500 K and are presented in Figure 1. These calculations were based on experimental data of Kubaschewski and Evans(14).

Standard free-energy changes of several reactions of UN and of UC₂ with some common gases and with carbon are presented in Figures 2 and 3.

The large negative values of standard free-energy change indicate that UN and UC₂ should react readily with oxygen, water, or steam. They are shown to be thermodynamically unreactive with hydrogen.

Standard free-energy changes for the reaction of US with oxygen, based on an estimated value for the heat of formation of US, are shown in Figure 4. Because of the large negative values for the calculated ΔF_R^0 , it is highly improbable that the estimated heat of formation is sufficiently in error that the actual ΔF_R^0 would be positive. Thus, it is fairly certain that, thermodynamically, US will have poor resistance to oxidation. A similar calculation showed that US is not thermodynamically resistant to steam.

Chemical Properties and Corrosion Data

Reactions of UC₂ with some common elements or compounds are given in Table 5.

UC₂ reacts with water at 82 C to give hydrogen, CH₄, paraffins, and traces of C₂H₂, CO, and CO₂. (15) At 248 C, approximately 96 per cent of the gaseous reaction product is hydrogen.

Moissan⁽¹⁶⁾ reported UC₂ to be decomposed by dilute HCl, HNO₃, and H₂SO₄, giving yellow uranyl salt solutions. Concentrated acids, except HNO₃, react only slowly with UC₂ at room temperature but very rigorously when heated. Daane⁽¹⁷⁾ reported a slow reaction with H₃PO₄ at room temperature and a vigorous reaction when heated.

UC2 was reported by Rideal(18) to be readily decomposed by alkalis.

UC decomposes in water at 83 C with the evolution of a gaseous mixture composed of approximately 78 per cent CH₄ and 12 per cent hydrogen. As the temperature is increased, the ratio of hydrogen increases until at 400 C the mixture contains 99.2 per cent hydrogen. (15)

US has been found to be stable in boiling water if it is well sintered. (1)

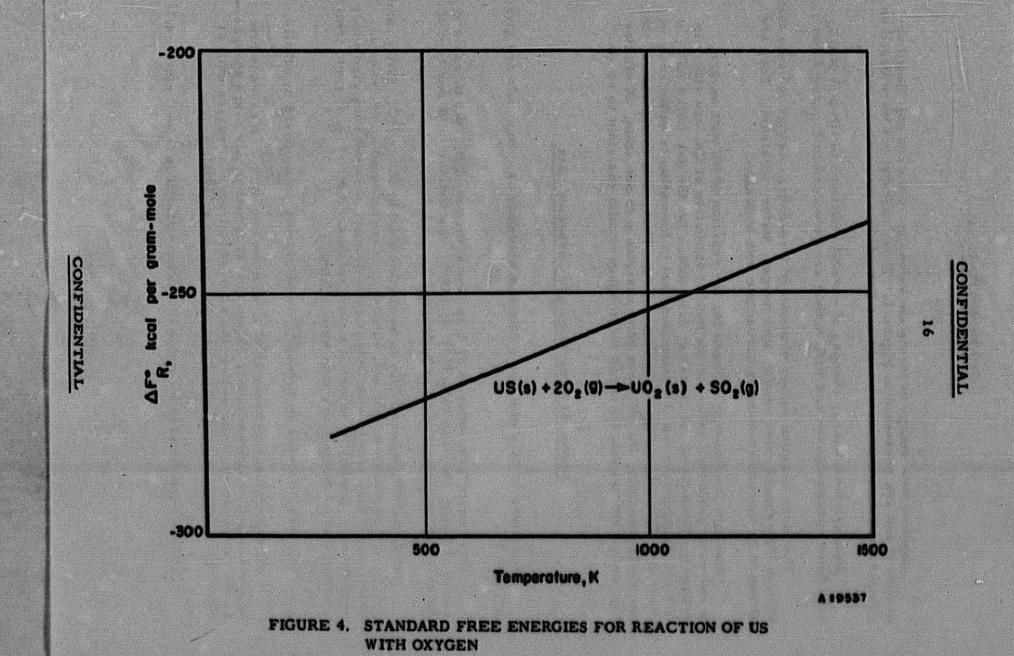


TABLE 5. CHEMICAL REACTIVITY OF UC_2

Reactant	Reaction Temperature, C	Products of Reaction	Remarks	Reference
02	970	U ₃ O ₈ and CO ₂		(16)
02	400-500	•	Oxidized completely within 4 hr in an air stream	(22)
N ₂	1100	••		(16)
N ₂	1180	Uranium nitride	After 12 hr all carbide is converted to nitride	(23)
Cl2	350	Volatile chloride		(16)
Cl2	. 600	uci ₄ .		(24)
F2	. 30	No reaction		(16)
F2	Slightly above 30	Explosive reaction		(16)
Br ₂	390		Carbide ignites in bromine vapor	(16)
Dr ₂	800-900	UBr ₄		(25, 26)
l ₂	500	UI4		(27)
NH3	Red heat		Partial decomposition of UC ₂	(16)
H ₂ S	600	A sulfide	UC2 ignited in hydrogen sulfide	(16)
5	•	Uranium sulfide and carbon disulfide		(16)
HCI	600	A uranium chloride		(16)
H ₂ O	•	Hydrocarbons	Decomposes slowly at room temperature, decomposes rapidly when heated	(16)

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Physical Properties

The boiling point of UC₂ was estimated by Mott(19) to be 4370 C under 760 mm of mercury.

Two values for the thermal conductivity of UC have been reported: 0.078(20) and 0.082(21) cal/(sec)(cm²)(C/cm). The former value is for a temperature of 44 C while the temperature of the latter measurement was not specified.

The average coefficient of thermal expansion of UC₂ for the temperature range 20-235 C was reported to be 12.5 x 10-6 per deg C. (28)

URANIUM SILICIDES AND BORIDES

Preparation

Uranium silicides (U3Si, U3Si2, USi, alpha USi2, beta USi2, and USi3) were prepared as part of the present research by heating stoichiometric mixtures of the elements in an electric arc furnace. Fabrication of shapes from the arc melts was accomplished by ceramic techniques.

Some of the as-cast uranium-silicon compounds crystallized as multiphase materials and had to be annealed to remove extraneous phases. This generally was done by extended heating of the ingot in an inert atmosphere near the melting or decomposition temperature. Annealing also was affected during sintering processes,

U3Si is more ductile than the other uranium silicides. It has been coextruded with other metals at 750-800 C. (29)

Brewer⁽⁶⁾ and Zalkin⁽³⁰⁾ prepared gray metallic crystals of UB₂ and UB₄ by heating stoichiometric mixtures of uranium and boron powder in an inert atmosphere in molybdenum crucibles at 1500 C for about 1 hr.

Crystallography

Crystallographic data on the uranium silicides and borides are listed in Table 6.

TABLE 6. CRYSTALLOGRAPHY OF URANIUM SILICIDES AND URANIUM BORIDES

	Unit Cell						
Compound	Туре	Dimensions,	Molecules per Unit Cell	Density, g per cm ³	Space Group	Remarks	Reference
UgSI .	Be tetragonal	a = 6.029 ± 0.002 c = 8.697 ± 0.003		15.58	14 mcm		(31) (32)
UgSI2	Tettagonal	a = 7.3298 ± 0.0004 c = 3.9003 ± 0.0005	2	12.20	P ₄ /mbm		(33)
USI	Orthorhombic	a = 5,66 ± 0,01 b = 7,66 ± 0,01		10,40	Ponm	Isomorphous with FeB	(31,32)
Alpha USI2	Bc tetragonal	a = 3.98 ± 0.03 c = 13.74 ± 0.08	•	8,98	14/amd	Isomorphous with ThSi ₂ , PuSi ₂ , CeSi ₂ , and NpSi ₂	(31,32)
Beta USi ₂	Hexa gonal	a = 3,86 ± 0,01 c = 4,07 ± 0,01	1	9.25	C6/mmm	Isomorphous with AIB ₂ and TiB ₂	(83)
USI ₂	Cubic	a = 4,035	•	7,80			(34)
USIS	Cubic	a = 4,03	1	8,15	Pm3m	L12-type AuCu ₃ ordered structure	(31, 32)
UB ₂	Hexagonal	a = 3,12 c = 3,96	: 1	19,82	-	May be isomorphous with AlB2	(29)
UB4	Tetragonal	a = 7.075 ± 0.004 c = 3.979 ± 0.002	•	9,38	F4/mbm	Isomorphous with ThB ₄ and CeB ₄	(30)

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The constitutional diagram of the uranium-silicon system, as determined by Kaufmann^(31, 32) indicates the existence of the intermediate phases, U₃Si, U₅Si₃, USi, U₂Si₃, USi₂, and USi₃. However, there is some doubt whether this diagram is accurate in the region 5 to 30 w/o silicon. In later work, Zachariasen⁽³³⁾ rejected the formulas U₅Si₃ and U₂Si₃. His interpretation of the X-ray diffraction data was that U₅Si₃ should be replaced by U₃Si₂, and that the phase identified by Kaufmann as U₂Si₃ is an allotrope of USi₂ with a hexagonal structure. Zachariasen called this compound beta USi₂. It is not quite clear whether Zachariasen's samples were of the same composition as those of Kaufmann as no chemical analyses were given.

Brauer and Haig (34) identified a cubic phase with the USi2 composition. Their samples were prepared in a molten aluminum bath and analyzed 19 w/o silicon and 80,7 w/o uranium.

Thermochemical Aspects

No experimental thermochemical data were found on the uranium silicides or borides. Estimates of the standard free energies of reaction of USi2 and UB2 with oxygen and water are shown in Figure 5. The predictions are based on Battelle estimates of heats of formation. The large negative values of the standard free-energy changes indicate that neither USi2 nor UB2 is thermodynamically resistant to oxidation. As USi2 probably is the most stable of the uranium silicides, none of the other silicides would be expected to have thermodynamic resistance to oxygen.

The silicides and borides appear to be thermodynamically resistant to hydrogen but not to water vapor, on the basis of Battelle estimates.

Chemical Properties and Corrosion Data

Data obtained at Battelle on the chemical stability of U₃Si and alpha USi₂ in acidic and basic solutions and in hydrogen gas are shown in Table 7. The results of corrosion tests in water of U₃Si and alpha USi₂ are given in Table 8.

U3Si is reported to resist exidation in air at 100 and 200 C, but not at 300 C. (35) It corrodes only slightly faster in steam at 125 psi than in boiling water. (35) The surface was roughened and a slight weight gain was observed after 5 hr in a lead-bismuth eutectic mixture at 425 C. (35)

Measurements made at Battelle on air oxidation of uranium silicides at 400 C are given in Table 9. In general, oxidation resistance in air

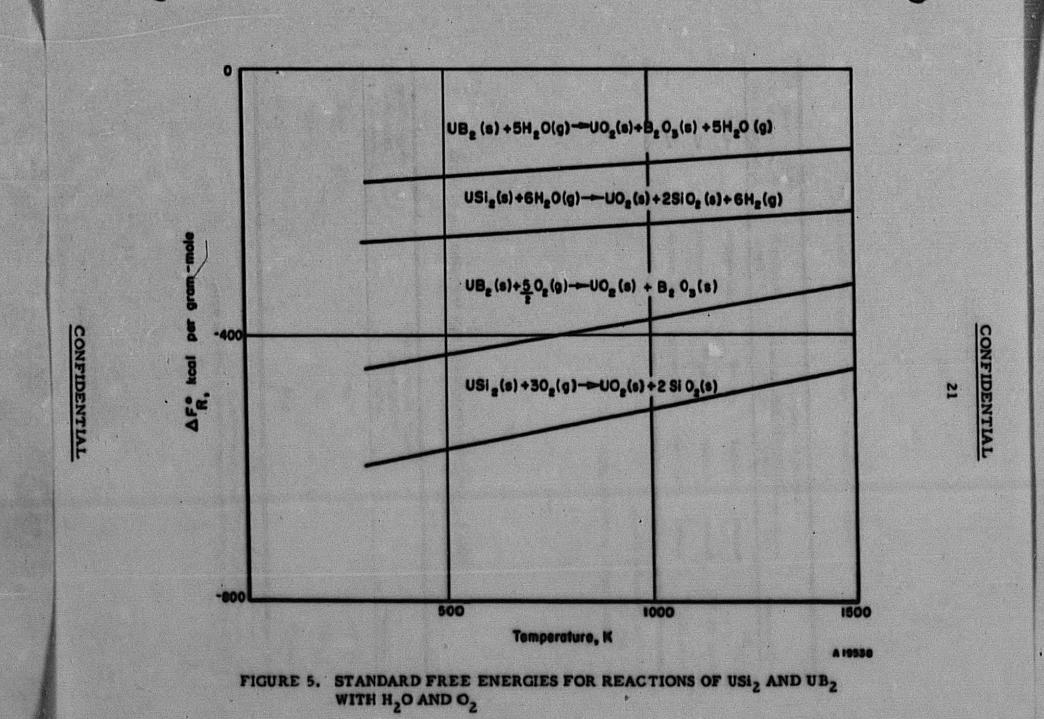


TABLE 1. CHEMICAL STABILITY OF Up is AND AUTHA $\mathrm{Usi}_2^{(a)}$

		tability is loc Concentrated		Stability in	Weight Change After 1-th Exposure in 500 C	
Compound	liQ	temp	11-204	IN NACH	Hydrogeo, per cent	
U38(6)	Reacted	Reacted	Rescord	No teaction	-4.0	
Alpha USig ^(b)	Bracked	Rescried	Rescred	No reaction	*4.00	
Alpha USig ^(C)	Reacted	Rescued	Slow reaction	No reaction	•••	

⁽a) Data obtained at Sattelle,

TABLE 4. COMPOSION RESISTANCE OF USSI AND ALPHA USIN IN WATER

سلم	Weight Loss After I III in Boiling Water, per cent	Weight Change in 650 F Water, mg/(cm ²)(br)
U ₃ s/(1)		-1.00
U _S S(N)		•1.00
ner ^t (c)		-1,00

⁽b) Arc melt.
(c) Compact stonered at \$400 C in argon.

⁽a) Data obtained at Sattelle on epsilonized arc melt,
(b) Data obtained by WAPO⁽²⁰⁾ on extraded bar,
(c) Data obtained at Sattelle on compact sintered at 1400 C in argon,

TABLE 9. OXIDATION IN AIR OF URANIUM SILICIDES

Compound	Silicon Content,	Weight Gain in 7-1/2 Hr at 400 C, per cent
U ₃ Si	3.6	19, 6 (disintegrated)
UjSiį	7,25	18.5 (disintegrated 1.4(a)
USI	10.5	16, 5 (disintegrated)
Alpha USi ₂	19.0	0.19(b) 0.14(a)
USI3	26.0	o. 07(e)

⁽a) Contained appentimately 3 to 5 w/o iron.
(b) Contained appentimately 1 w/o tungsten.
(c) Contained appentimately 4 w/o tungsten.

Note: Data obtained at Sattelle. Specimens weighed approximately 2.5 g, and were 1/2 in. in dia by 1/4 is, long.

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increased with increasing silicon content, although the results on USi2 and USi3 may have been influenced by impurities. The presence of 3 to 5 w/o iron in U3Si2 greatly improved its oxidation resistance.

Physical Properties

The thermal conductivity of U₃Si was measured by Foote⁽³⁶⁾, with the following results:

Temperature, C	Thermal Conductivity, cal/(sec)(cm ²)(C/cm)
	0.036 .
50	0,041

Coefficients of linear thermal expansion for U3Si, U3Si2, and USi3 are given in Table 10,

The data of Kaufmann (32) on the strength of U₃Si in compression and tension are given in Table 11.

COMPOUNDS WITH METALS

Preparation

Compounds of uranium with aluminum, iron, or nickel were prepared at Battelle by meiting stoichiometric mixtures of the elements in an electric arc furnace. The products were brittle, and fabrication of shapes required ceramic techniques.

Crystallography

Crystallographic data for compounds of uranium with aluminum, iron, or nickel are given in Table 12.

'The compound UAl2 was tentatively identified by Gordon and Kaufmann(37) on the basis of microscopic examination of alloys and X-ray determinations of crystal structure. Because of difficulties with chemical

TABLE 10. THERMAL EXPANSIONS OF URANIUM SILICIDES (A)

		Mean Coefficient of Linear Thermal Expansion, 10 ⁻⁶ per deg C			
Temperature Range, C	U ₃ Si(b)	U3Si2(c)	USi3 (d)		
20-200	13.0	15,5	13.4		
20-300	13.4	15.3	13,6		
20-400	14.2	15.2	14, 3		
20-500	14.9	15.3	14.6		
20-600	15.8	15,2	14.9		
20-700	16.8	15, 1	15.4		
20-750	17,5		-		
20-800		15,0	15.7		
20-900		14.7	16.1		
20-950		14.6	16,3		

⁽a) Data obtained at flattelle.
(b) As-cast inper epsilonized at 800 C for 168 hs.
(c) U₃Si₂ powder sintered at 1400 C in argon for 15 hs.
(d) USi₃ powder sintered at 2250 C in argon for 15 hs. Contained approximately 4 w/o tungston.

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TABLE 11. STRENGTH OF U3SI(a)

Temperature, C	Yield	Strength,	Ultimate Strength,		
	Tension	Compression	Tension	Compression	
8	37,000	115,000	37,000	280,000	
600		55, 000			
700		18,000			
750		10,000			
850		4,000			

(a) From Kaufmann, (32)

TABLE 12, CRYSTALLOGRAPHY OF URANIUM COMPOUNDS WITH METALS

	Unit Cell			X-Ray			
Compound	Type	Dimensions,	Number of Molecules	g per cm ³	Space Group	Pemarka	Reference
UAI2	Poc .	4 * 7.011	•	8.14	F4 3m	C15-type Cu ₂ Mg structure	(37)
UeNE	Sc tetragonal	a = 10,37 a 0,04 c = 5,31 a 0,02		1	14/mcm, 142 or 14 c 2	Isomorphous with UgFe. UgCo, and UgMn	(40)
UN12	Hexa gonal	a • 4, 968 c • 8, 252		19,46	C6/mmc	C14-type MgZn ₂ structure	(40)
UNIS	rœ	a * 6. 7830 a 0. 0005			F43m ot F20	Similar to MgCu ₂ structure, but of lower symmetry, isomorphous with UCu ₅ , PdBe ₅ , and AuBe ₅	(40)
Ugfe	Oc tetragonal	A = 10.31 a 0.04 c = 5.24 a 0.02		17.7	14/mcm, 142 or 14 c 2	Isomorphous with UgMn, UgCo, and UgNi	(41)
UPeg	Too	4 = 7.012	•	13,21	Fd Sm	G15-type MgCu ₂ structure; isomorphous with UAL ₂	(41)

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analyses and with inhomogeneity in their alloys, the composition could not be fixed to better than \$2 a/o. However, the measured density (8.21 g per cm³) of their sample corresponded closely with the calculated X-ray density (8.38 g per cm³).

The uranium-nickel system has been investigated by Grogan and Pleasance (38), Foote (39), and Baenziger (40). The data in Table 12 are from Baenziger. He also attempted to analyze the pattern of an alloy with the composition UNi but was unable to resolve its complex structure.

Properties

No data on the physical or chemical properties of these compounds were found. Also, there are no reliable methods of estimating their thermodynamic properties.

REFERENCES

Note: Reference numbers marked with an asterisk were taken from "The Chemistry of Uranium", Part I, J. J. Katz and E. Rabinowitch, National Nuclear Energy Series, VIII-5 (1951).

- Eastman, E. D., Brewer, L., Bromley, L. A., Gilles, P. W., and Lofgren, N. L., "Preparation and Properties of the Sulfides of Thorium and Uranium", J. Am. Chem. Soc., 72, 4019-4023 (1950).
- (2) Rundle, R. E., Baenziger, N. C., Wilson, H. S., and McDonald, R. A., "The Structures of the Carbides, Nitrides, and Oxides of Uranium", J. Am. Chem. Soc., 70, 99-105 (1948).
- (3) Mallett, M. W., Gerds, A. F., and Nelson, H. R., "The Uranium Carbon System", J. Electrochem. Soc., 99, 197-204 (1952).
- (4) Chiotti, P., "Experimental Refractory Bodies of High-Melting Nitrides, Carbides, and Uranium Dioxide", J. Am. Ceram. Soc., 35, 123-130 (1952).
- (5) Saller, H. A., and Rough, F. A., "Compilation of U. S. and U. K. Uranium and Thorium Constitutional Diagrams", BMI-1000 (June, 1955).
- (6) Brewer, L., Sawyer, D. L., Templeton, D. H., and Dauben, C. H., "A Study of the Refractory Borides", J. Am. Ceram. Soc., 34, 173-179 (1951).



- (7) "Metallurgy of Tuballoy", BMI-HWR-26 (February 1, 1945).
- (8) Daane, A. H., Spedding, F. H., and Wilhelm, H. A., "The Preparation and Fabrication of the Carbides of Uranium", ISC-11 (December 22, 1947).
- (9) Mallett, M. W., Gerds, A. F., and Vaughan, D. A., "Uranium Sesquicarbide", J. Electrochem, Soc., 98, 505-509 (1951).
- (10) Litz, L., Garrett, A. B., and Croxton, F. C., "Preparation and Structure of the Carbides of Uranium", J. Am. Chem. Soc., 70, 1718 (1948).
- *(11) Rammelsberg, C., Poggendorff's Annalen der Physik, 53, 323 (1642).
- (12) Brewer, L., Bromley, L., Gilles, P., and Lodgren, N., "The Preparation and Properties of Refractory Sulfides", AECD-2242 (September 27, 1951).
- (13) Rossini, F. D., et al., "Selected Values of Chemical Thermodynamic Properties", NBS Circular-500, U. S. Government Printing Office, Washington, D. C. (1952).
- (14) Kubaschewski, O., and Evans, E. D., "Metaliurgical Thermochemistry", Academic Press, Inc., New York, N. Y. (1951).
- (15) Litz, L. M., "Uranium Carbides Their Preparation, Structure, and Hydrolysis", Dissertation, The Ohio State University (1948).
- *(16) Moissan, H., Bull. soc. chim. France, Paris, 17, 14 (1897).
- *(17) Daane, A. H., Snow, A. I., CT-751 A (June 2, 1943).
- *(18) Rideal, E. K., Dissertation, University of Bonn, Germany.
- *(19) Mott, W. R., Trans. Electrochem. Soc., 34, 279 (1918).
- (20) NEPA Technical Report No. 5, "Special Materials", 59 (September 15, 1947).
- *(21) Carter, J. H., CT-609 (April 24, 1943).
- *(22) Spedding, F. H., CP-42 (April 25, 1942).
- *(23) Heusler, O., Anorg. V. Chem., 154, 366 (1926).
- *(24) Almann, D. H., CT-393, Sec. L. Part B3 (December 15, 1943).

UNCLASSIFIED



- *(25) Spedding, F. H., CC-298 (October 16, 1942).
- *(26) Powell, T., CC-1778 (August 18, 1944).
- *(27) MacWood, G. E., and Altman, D., UCRL-1, RL 4.7.600 (October 24, 1944).
- (28) Unpublished Battelle data (1948).
- (29) Losco, E. F., and Shipiro, C. M., "Résumé of Uranium Alloy Data, IX", WAPD-PMM-262 (September 1, 1955).
- (30) Zaikin, A., Templeton, D. H., "The Crystal Structures of CeB4, ThB4, and UB4", Acta Cryst., 6, 269-272 (1953).
- (31) Kaufmann, A. R., and Cullity, B. D., "Alloys of Uranium and Silicon, I, The Uranium-Silicon Phase Diagram", CT-3310 (June, 1945).
- (32) Kaufmann, A. R., and Bitelanes, Guet, "Alloys of Uranium and Silicon, II, The Epsilon Phase", CT-3309 (June, 1945).
- (33) Zachariasen, W. H., "Crystal Chemistry Studies of the 5f-Series of Elements, VIII, Crystal Structure Studies of Uranium Silicides and CeSi₂, NpSi₂, and PuSi₂", Acta Cryst., 2, 94 (1949).
- (34) Brauer, V. G., and Haig, H., Z. anorg. Chem., 259 (1949).
- (35) Losco, E. F., WAPD Technical Progress Report for Period January 13 to February 23, 1956", "Pressurised Water Reactor Program", WAPD-MRP-60.
- (36) Foote, F., CT-2668 (January 23, 1945).
- (37) Gordon, P., and Kaufmann, A. R., "Uranium-Aluminum and Uranium-Iron", J. Metals, 182 (1950).
- (35) Grogan, J. D., and Pleasance, R. J., "A Survey of the Uranium-Nickel System", J. Inst. Metals, 82, 141-147 (1953-54).
- (39) Foote, F., Clark, I. R., Cieslicki, M., Nelson, B. J., and Lane, T. R., Unpublished (1945); given in BMf-1000, see Reference (5).
- (40) Baenziger, N. C., Rundle, R. E., Snow, A. I., and Wilson, A. S., "Compounds of Uranium with the Transition Metals of the First Long Period", Acta Cryst., 3, 34-40 (1969).

UNGLAS ... CONGIDERYTIAL



UNCLASSIFIED

(41) Grogen, J. D., "The Uranium-Iron System", J. Inst. Metals, 77, 571-580 (1950).

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