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THE THERMAL EXPANSION OF FIVE TITANIUM CARBIDE CERMETS

By

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ABSTRACT

This paper presents measurements of the thermal expansion of five titanium carbide type cermets from 68 to 1800F. These cermets are designated by Kennametal, Inc., as K 138A, K 150A, K 151A, K 152B and K 162B. They contain from 64 to 80 weight percent titanium carbide, 10 to 30 weight percent metal binder and 6 to 10 weight percent other carbides. The metal binders are cobalt, nickel, or nickel and molybdenum.

An attempt was made to calculate the thermal expansion of each type cermet from the thermal expansions of the constituents. The expansion of the mixture was computed by weighting the expansions of the constituents according to (1) weight percent of the constituents, (2) volume percent of the constituents and (3) according to a value developed for mixtures by P. S. Turner.¹ It was found that expansions computed according to volume percent and by Turner's method agreed with measured values within ± 5 percent. The values calculated by weight percent were from 5 to 11 percent higher than the observed values.

The thermal expansions of these cermets are compared with the expansions of a group of metals and alloys.

¹P. S. Turner, "Thermal-Expansion Stresses in Reinforced Plastics," J. Research NBS, Vol. 37, July-December 1946, R. P. 1745.

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Introduction

The measurement of the thermal expansion of these titanium carbide cermets was undertaken to evaluate materials for values and bearings at elevated temperatures. An attempt was made to find a method for calculating the thermal expansion of such mixtures.

Specimens

The materials studied were Kennametal, Inc. catalogue mixtures K 138A, K 150A, K 151A, K 152B and K 162B. No chemical analysis of these cermets was made at Pratt & Whitney Aircraft. The nominal compositions shown in Table I are from Kennametal, Inc. advertising literature.

The dilatometer specimens were purchased from Kennametal, Inc. They were approximately 0.158 in. (4 mm) in diameter and 1.97 in. (50 mm) long. The measured bulk density of each material is shown in Table I, together with nominal and calculated density values.

Method

A. Tests

The bulk densities of these specimens were obtained by dividing the weight by the volume. The specimens were weighed on an analytical balance to the nearest 0.0001 g. The length and diameter were measured with micrometer calipers to the nearest 0.0001 in.

The thermal expansion measurements were made with the aid of the Leitz dilatometer shown in Figures 1 and 2. In this dilatometer the specimen expansion is magnified about 200 times and recorded on a photographic film by a beam of light. A change in length of 0.1 mm may be estimated on the photographic film. This represents a change in specimen length of 0.0005 mm (0.00002 in.). For a specimen 50 mm (1.97 in.) long this represents a change in length of 10⁻⁵ mm/mm or in./in. By repeating measurements of a standard material such as copper the reproducibility of the instrument, run at the heating rate we use, has been established as less than ± 3 percent.

The magnification factor of the instrument is redetermined once a year. It is determined by measuring the distance between dots made on the photographic film for each 0.2 mm change in specimen length. The magnification factor (196.2) has not changed more than 0.3 percent in the last five years.

For recording the expansion results a film was selected which shrinks less than 0.5 percent during processing. The films are stored, developed and measured at the same constant temperature and humidity at which they are exposed. This treatment assured a negligible change in dimensions of the film.

Heating of the specimen was accomplished by a program controlled electric resistance furnace. At the lower temperatures heating was supplied at the rate of 3 degrees F per minute and was gradually increased to 5 degrees F per minute. Heating from room temperature to 1800F was accomplished in 6 1/2 hours. When a constant heating rate of 4 deg/min was used the results showed a lag in specimen expansion at temperatures below 800F. When the heating rate from room temperature to 800F was reduced this lag was avoided.

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These specimens were heated in a vacuum at a pressure of from 0.1 to 0.4 microns. This vacuum was probably partially responsible for the poor heat transfer to the specimen at low temperatures.

The temperature of the specimen was assumed to be equal to the temperature measured with a chromel-alumel thermocouple, whose hot-junction was located in the groove in a standard chronin specimen. The test specimen and standard specimen were adjacent and symmetrically placed in the test zone. (See Figure 2). Temperatures were read on a Minneapolis Honeywell recorder, graduated in 10 deg F intervals. The test specimen and a standard specimen were placed horizontally in quartz troughs. The change in length of each is transmitted by a quartz rod to the head of the instrument. As these specimens expand, the quartz rods change the angle of the prism from which the beam of light is reflected to the photographic film. Thus, the change in specimen length causes a line to be drawn across the photographic film. The test specimen displaces the light vertically while the standard displaces it horizontally. If the test specimen and the standard specimen are of the same material the trace on the film would be a straight line at an angle of 45 degrees. During heating, the light is turned off at 100 degree F intervals. When analyzing the film, the vertical displacement of the light from the starting point is read at the breaks in the line. An expansion measurement is thus obtained for each 100 deg F. These readings are in millimeters on the film and must be divided by the magnification factor and the specimen length to bring the results to units of inches per inch of specimen length. A correction was made for the expansion of the 2 inch length of quartz trough under the specimen. No correction is necessary for the remaining length of trough and push rod since their expansions compensate for one another.

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A parabola, $y = A + BT + CT^2$ was fitted to the temperature-expansion data by the least squares method. For each material the curve fitted the experimental data within ± 3 percent for all temperatures above 800F for 2 or more runs. The mean coefficient, \measuredangle , from 68F to T is $\frac{y}{T-68}$. The instantaneous coefficient of expansion is the slope of the expansion-temperature curve, i.e., $\frac{dy}{dT} = B + 2CT$.

B. Calculations

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An attempt was made to find a method for calculating the thermal expansion of these cermets from the known expansions of the constituents. The method presented by Tarner (Ref 1) seemed the best approach for our material. He considers the coefficient of expansion of a mechanical mixture as the result of the stresses set up by the difference in coefficient of expansion of the constituents. The resulting coefficient of expansion would be:



in which & = mean coefficient of thermal expansion

P = weight fraction of constituent

- K = bulk modulus
- d = density

Subscripts i (1 to n) and r refer respectively to the ith component and to the resultant mixture. It can be seen that if the K's in the above formulas are nearly equal, the calculated \checkmark will be proportional to the volume fraction of the constituents. If the K/d ratios are nearly equal the resulting \checkmark will be nearly proportional to the weight fractions of the constituents. For these constituents the K's are about equal, with the exception of molybdenum. The K/d ratio for TiC is about twice that of nickel or cobalt.

In calculating the thermal expansions of these cermets, the mean coefficients of thermal expansion of the constituents used are shown in the curves of Figure 4. The values for the bulk modulus and density of nickel, cobalt, molybdenum, columbium-tantalum-titanium mixed carbides and titanium carbide used in our calculations are given in Table II. The thermal expansion of molybdenum was measured at Pratt & Whitney Aircraft-CANEL (Ref 2). The expansion data for nickel were from NBS Circular 592 (Ref 3) from room temperature to 1652F, and from WADC-TR-55-495 (Ref 4) from 1000F to 1800F. The thermal expansion data for cobalt were from measurements by Fine and Ellis (Ref 5). The expansion data for the carbides were from the data of Mauer and Bolz (Ref 6). These measurements were made by the X-ray diffraction method on a single crystal. Expansion measurements made on a specimen rather than a single crystal, would be more representative of the material with which we are dealing. Values for the thermal expansion of hot pressed titanium carbide reported in Ref 7 are higher than Mauer and Bolz, while those reported in Ref 8 are lower.

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C Results

The thermal expansion curves to 1800F of these five titanium carbide cermets are shown in Figure 3. Measurements after test showed no permanent change in length. Their mean coefficients of expansion, together with that of their constituents are shown in Figure 4. The mean coefficients of thermal expansion of these cermets are compared with those for a group of metals and alloys in Figure 5. The instantaneous coefficient of expansion for these cermets is given in Figure 6. The measured coefficients of expansion of these cermets are compared with the coefficients calculated from the expansions of the constituents by three different methods, in Figure 7. The calculated values were obtained by weighting the expansions of the constituents according to (1) the weight percent, (2) the volume percent and (3) Turner's stress method. The constants for the least squares parabola _ obtained for the thermal expansion curve of each cermet are given in Table III.

Our thermal expansion data for these cermets is compared with data reported by others in Table IV. The compositions of the cermets reported in Refs 9 and 10 differ slightly from our cermets. The difference between their measured expansions and ours is in the direction to be expected from the difference in composition.

Discussion

The thermal expansions of five titanium carbide cermets were measured and an attempt made to find a method for computing their thermal expansions from the expansions of the constituents. The values calculated by weight percent were from 5 to 11 percent higher than the observed values. The method presented by Turner (Ref 1) and the method based upon volume percent were found to yield values which agreed with measured values within ± 5 percent. This is considered a very good approximation since values reported for various specimens of TiC by one observer (Ref 8) differ by ± 4 percent and values reported by other observers

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differ more widely. The thermal expansion of hot pressed TiC bodies (Ref 7) showed a thermal expansion from room temperature to 1100F which was 2 percent higher than that measured by Mauer and Bolz (Ref 6). The thermal expansion of two hot pressed TiC bodies measured by Engberg and Zehms (Ref 8) showed expansions from RT to 1000C which were 9 to 11 percent below that of Mauer and Bolz. Using an expansion value for TiC 10 percent lower than that used in our calculations would lower the calculated Turner value approximately 6 to 8 percent. Using a value for \ll of TiC 4 percent lower than Mauer and Bolz would give calculated values for all these materials within ± 3 percent of the observed.

Knudsen, Moreland and Geller (Ref 9 and Ref 11) reported thermal expansion data for K 151A and K 138A. Their specimens apparently contained two percent less binder and 5 percent more mixed carbides than did ours. It is therefore to be expected that their expansions were less than ours. However, the difference in measured expansions was larger than was to be expected from this cause alone. The thermal expansions of K 162B and K 152B reported by Kennametal, Inc., (Ref 14, 15 and 16) are somewhat higher than our measurements. Expansions for 80 TiC, 20 Co measured by G. D. Deutsch, et al (Ref 12) are somewhat higher than our measured expansions for K 138A 20 Co, 10 Cb(TaTi)C, 70 TiC. A higher expansion is to be expected since the expansion of TiC is greater than Cb(TaTi)C.

Summary

The thermal expansions of four TiC cermets reported here have been compared with those reported in the literature and found to agree within a reasonable accuracy.

Turner's method for calculating thermal expansions of mixtures was found to yield results within the accuracy range of various observers of a single constituent such as TiC.

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The expansions of these five cermets have been compared (Fig 5) with the expansions of various metals and alloys. This group of cermets could probably be used with the metals such as chromium and columbium without difficulties due to differences in expansion.

Acknowledgement

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TABLE I

Chemical Composition and Density

	Noni	nal Ch	emical (Composition, we	ight %		Density, g	/cc
	Ni	Co	Mo	Cb (TaTi)C*	TIC	Measured	Calculated	(Kennametal)
K 150A	10			10	80	5.82	5.34	5.8
K 151A	20			10	70	5.64	5.61	5.8
K 138A		20		10	70	5.75	5.61	5.8
K 162B	25		5	6	64	6.07	5.84	6.0
K 152B	30			6	64	5.93	5.81	6.0

"Added as a solid solution of CbC, TaC and TiC, containing about 90 per cent CbC.

TABLE II

Values for the Bulk Modulus and Density Used in These Calculations

	N1	Co	Mo	Cb(TaTi)C	TIC
K, 10 ⁶ psi	26.3	25	43.6	24	25
d, g/cc lb/cu in.	8.90	8.90	10.2	7.72	4.90
K/d, 10 ⁶ in.	81.7	77.6	118.2	86.0	141.2

TABLE III

Constants for the Thermal Expansion versus Temperature

Equation Y = A + BT + CT²

	▲	B	c		
K 150A	247 x 10-3	3.62 x 10-6	.500 x 10-9		
K 151A	282	3.80	-527		
K 138A	275	3.88	.520		
к 1628	294	4.16	.547		
K 152B	302	4.14	.608		

"Y in inches/inch T in degrees F

TABLE IV

THEFTAL ENGLISH DATA FROM OTHER OBSERVED.S Thermal Expension, inches/inch.

	1		K 151	4		I constant		K	574			K 1628	_
Temperature dec F	Speci	imen Fl	ef 9 Speci	men #2 Cooling	Falla Seating	Spect Heating	men fl	ef 9 Spect Heating	ner #2 Conling	Fisha Neating	Temperature deg F	Ref. 15	758
79 to 572 79 to 752 79 to 752 79 to 112 79 to 1292 79 to 1292 79 to 1652 79 to 1652 75 to 1632 75 to 2201 Nominal Composition	.0019 .0027 .0035 .0013 .0051 .0055 .0066 .0077 .0093	-0013 -0027 -0027 -0035 -0014 -0051 -0051 -0070 -0091	.0018 .0027 .0035 .0044 .0053 .0053 .0052 .0071 .0080 .0096	.0020 .0029 .0037 .0016 .0055 .0063 .0073 .0093	.00206 .00786 .00370 .00550 .00550 .00550 .00645 .00713 .00814	.0019 .0027 .0036 .0053 .0053 .0053 .0053 .0051 .0051 .0103	.0018 .0026 .0035 .0044 .0053 .0062 .0073 .0080 .0100	.0018 .0096 .0035 .0063 .0052 .0052 .0070 .0070 .0090 .0102	.0017 .0025 .0035 .0051 .0051 .0050 .0050 .0050 .2100	.00211 .00298 .00379 .00468 .00956 .00956 .00956 .00955 .00858	HT to 1800	.00594 .00507	.005
Co Ni			BI					18 1		1 - 20%			

tean Coefficient of Thermal Expansion, microinch/inch-deg /

Temperature	K 2	623	K 1529		Temperature			K 335A
deg F	Sef 16	PENA	Ref 1L	FEWA	deg 7	Pef	10	TEMA
200 to 1200 5-1	1 4.94	5.3	4.99	58 to 1100	4.6	3.9	6.10	
70 to 1200	1.95	1.05	5.0	1.90	Composition Co TiC Cb(Ta,Ti)C	208 80	30% 70	20% 70 10



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Thermal expansion apparatus.



Thermal expansion apparatus showing specimen in place.

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Thermal expansion of five titanium carbide cermets.



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Mean coefficients of thermal expansion of five titanium carbide cermets and of their constituents. (room temperature to T deg F)



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Comparison of the thermal expansions of some metals and alloys with the expansions of these five titanium carbide cermets. (mean coefficients room temperature to T deg F)



Instantaneous coefficients of thermal expansion of five titanium carbide cermets.



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Comparison of the measured thermal expansion of five titanium carbide cermets with that calculated from their compositions by three different methods. (mean coefficients of thermal expansion 68 to T deg F)



